



# Spacegroup P4<sub>2</sub>/mnm

**Structure given by:**  
 spacegroup  
 lattice parameter  
 positions of atoms  
 (basis)

**Rutile TiO<sub>2</sub>:**  
 P4<sub>2</sub>/mnm (136)  
 $a=8.68$ ,  $c=5.59$  bohr

Ti: (0,0,0)

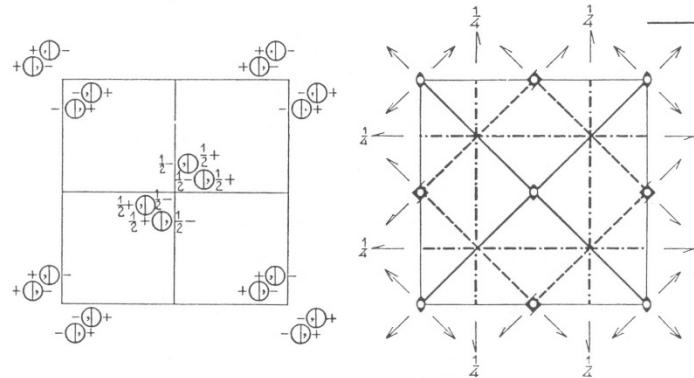
O: (0.304,0.304,0)

P4<sub>2</sub>/mnm  
 $D_{4h}^{14}$

No. 136

P 4<sub>2</sub>/m 2<sub>1</sub>/n 2/m

4/m m m Tetragonal



Origin at centre (mmm)

Number of positions,  
 Wyckoff notation,  
 and point symmetry

Co-ordinates of equivalent positions

Conditions limiting  
 possible reflections

16	<i>k</i>	1	$x,y,z$ ; $\bar{x},\bar{y},z$ ; $\frac{1}{2}+x,\frac{1}{2}-y,\frac{1}{2}+z$ ; $\frac{1}{2}-x,\frac{1}{2}+y,\frac{1}{2}+z$ ; $x,y,\bar{z}$ ; $\bar{x},\bar{y},\bar{z}$ ; $\frac{1}{2}+x,\frac{1}{2}-y,\frac{1}{2}-z$ ; $\frac{1}{2}-x,\frac{1}{2}+y,\frac{1}{2}-z$ ; $y,x,z$ ; $\bar{y},\bar{x},z$ ; $\frac{1}{2}+y,\frac{1}{2}-x,\frac{1}{2}+z$ ; $\frac{1}{2}-y,\frac{1}{2}+x,\frac{1}{2}+z$ ; $y,x,\bar{z}$ ; $\bar{y},\bar{x},\bar{z}$ ; $\frac{1}{2}+y,\frac{1}{2}-x,\frac{1}{2}-z$ ; $\frac{1}{2}-y,\frac{1}{2}+x,\frac{1}{2}-z$ .	General: $hk\ell$ : No conditions $hk0$ : No conditions $0kl$ : $k+l=2n$ $hh\ell$ : No conditions
----	----------	---	--	---

8	<i>j</i>	<i>m</i>	$x,x,z$ ; $\bar{x},\bar{x},z$ ; $\frac{1}{2}+x,\frac{1}{2}-x,\frac{1}{2}+z$ ; $\frac{1}{2}-x,\frac{1}{2}+x,\frac{1}{2}+z$ ; $x,x,\bar{z}$ ; $\bar{x},\bar{x},\bar{z}$ ; $\frac{1}{2}+x,\frac{1}{2}-x,\frac{1}{2}-z$ ; $\frac{1}{2}-x,\frac{1}{2}+x,\frac{1}{2}-z$ .	Special: as above, plus } no extra conditions
---	----------	----------	--	--

8	<i>i</i>	<i>m</i>	$x,y,0$ ; $\bar{x},\bar{y},0$ ; $\frac{1}{2}+x,\frac{1}{2}-y,\frac{1}{2}$ ; $\frac{1}{2}-x,\frac{1}{2}+y,\frac{1}{2}$ ; $y,x,0$ ; $\bar{y},\bar{x},0$ ; $\frac{1}{2}+y,\frac{1}{2}-x,\frac{1}{2}$ ; $\frac{1}{2}-y,\frac{1}{2}+x,\frac{1}{2}$ .	$hkl$ : $h+k=2n$ ; $l=2n$
---	----------	----------	--	---------------------------

8	<i>h</i>	2	$0,\frac{1}{2},z$ ; $0,\frac{1}{2},\bar{z}$ ; $0,\frac{1}{2},\frac{1}{2}+z$ ; $0,\frac{1}{2},\frac{1}{2}-z$ ; $\frac{1}{2},0,z$ ; $\frac{1}{2},0,\bar{z}$ ; $\frac{1}{2},0,\frac{1}{2}+z$ ; $\frac{1}{2},0,\frac{1}{2}-z$ .	
---	----------	---	--	--

4	<i>g</i>	<i>mm</i>	$x,\bar{x},0$ ; $\bar{x},x,0$ ; $\frac{1}{2}+x,\frac{1}{2}+x,\frac{1}{2}$ ; $\frac{1}{2}-x,\frac{1}{2}-x,\frac{1}{2}$ .	
---	----------	-----------	---	--

4	<i>f</i>	<i>mm</i>	$x,x,0$ ; $\bar{x},\bar{x},0$ ; $\frac{1}{2}+x,\frac{1}{2}-x,\frac{1}{2}$ ; $\frac{1}{2}-x,\frac{1}{2}+x,\frac{1}{2}$ .	
---	----------	-----------	---	--

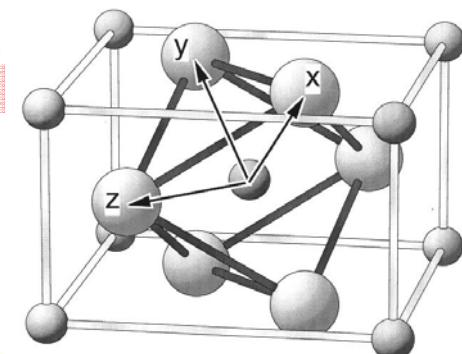
4	<i>e</i>	<i>mm</i>	$0,0,z$ ; $0,0,\bar{z}$ ; $\frac{1}{2},\frac{1}{2},\frac{1}{2}+z$ ; $\frac{1}{2},\frac{1}{2},\frac{1}{2}-z$ .	
---	----------	-----------	---	--

4	<i>d</i>	4	$0,\frac{1}{2},\frac{1}{4}$ ; $\frac{1}{2},0,\frac{1}{4}$ ; $0,\frac{1}{2},\frac{3}{4}$ ; $\frac{1}{2},0,\frac{3}{4}$ .	
---	----------	---	---	--

4	<i>c</i>	2/m	$0,\frac{1}{2},0$ ; $\frac{1}{2},0,0$ ; $0,\frac{1}{2},\frac{1}{2}$ ; $\frac{1}{2},0,\frac{1}{2}$ .	
---	----------	-----	---	--

2	<i>b</i>	mmm	$0,0,\frac{1}{2}$ ; $\frac{1}{2},\frac{1}{2},0$ .	
---	----------	-----	---	--

2	<i>a</i>	mmm	$0,0,0$ ; $\frac{1}{2},\frac{1}{2},\frac{1}{2}$ .	
---	----------	-----	---	--



- **Specify:**

- *Number of nonequivalent atoms*
- *lattice type (P, F, B, H, CXY, CXZ, CYZ) or spacegroup symbol*
  - if existing, you must use a **SG-setting** with inversion symmetry:
    - Si:  $\pm(1/8,1/8,1/8)$ , not  $(0,0,0)+(1/4,1/4,1/4)$ !
- *lattice parameters  $a,b,c$  (in Å or bohr)*
- *name of atoms (Si) and fractional coordinates (position)*
  - as numbers (0.123); fractions (1/3); simple expressions ( $x-1/2, \dots$ )
  - in fcc (bcc) specify just one atom, not the others in  $(1/2,1/2,0; \dots)$

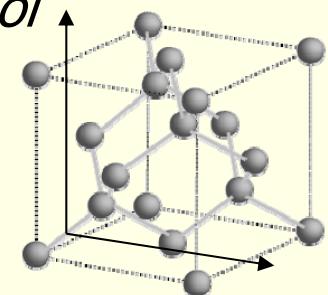
- „**save structure**“

- *updates automatically Z, r0, equivalent positions*

- „**set RMT and continue**“: (specify proper “reduction” of NN-distances)

- *non-overlapping „as large as possible“ (saves time), but not larger than 2.5 bohr*
- *RMT for sp (d) - elements 10-20 % smaller than for d (f) elements*
- *largest spheres not more than 50 % larger than smallest sphere*
- *Exception: H in C-H or O-H bonds: RMT~0.6 bohr (RKMAX~3-4)*
- *Do not change RMT in a „series“ of calculations, RMT equal for same atoms*

- „**save structure – save+cleanup**“



## ■ init\_lapw

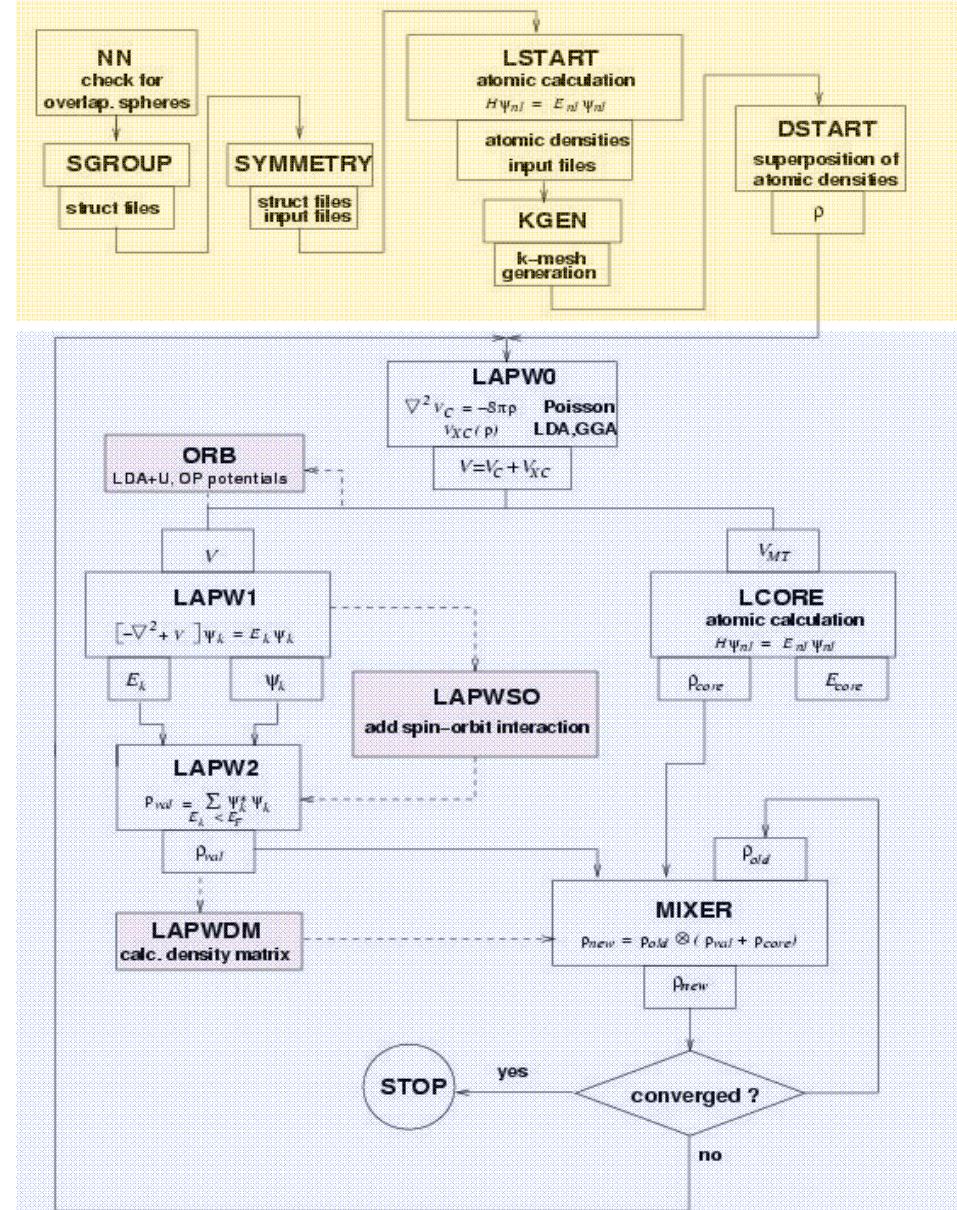
- *step-by-step or batch initialization*
- *symmetry detection ( $F$ ,  $I$ ,  $C$ -centering, inversion)*
- *input generation with recommended defaults*
- *quality (and computing time) depends on **k-mesh** and **R.Kmax** (determines #PW)*

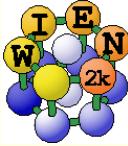
## ■ run\_lapw

- *scf-cycle*
- *optional with SO and/or LDA+U*
- *different convergence criteria (energy, charge, forces)*

## ■ save\_lapw tic\_gga\_100k\_rk7\_volo

- *cp case.struct and clmsum files,*
- *mv case.scf file*
- *rm case.broyd\* files*





- The convergence criterion in APW is the product of  $R_{MT} \cdot K_{max}$

$$\Psi = \sum_{K_n}^{K_{MAX}} c_{K_n} e^{iK_n r}$$

- [http://www.wien2k.at/reg\\_user/faq/rkmax.html](http://www.wien2k.at/reg_user/faq/rkmax.html)
- medium quality convergence for **smallest** atom:

- basis set scales with  $RKmax^3$
- cputime scales with  $N_{PW}^3$
- increasing  $Rkmax$  by 10 %  
→ doubles cputime

Rkmax	Element
3.0	H
4.5	Li
5.0	Be, B, Si
5.5	C, P
6.0	N, S
6.5	O, Cl, Na, K, Rb, Cs, Mg, Ca, Sr, Ba, Al
7.0	F
7.5	Sc-Cr, Ga-Br, Y-Mo
8.0	Mn-Zn, Ru-Cd, In-I, La, Ce, Hf-Re
8.5	Os-At, Pr-Lu, Ac-Lr

**START** with **SMALL**  $Rkmax$  (relaxation), **increase/test** later

- Replace the “integral” of the BZ by a finite summation on a mesh of “k-points”

$$\rho(r) = \sum_n^{E_n < E_F} \int \psi_{k,n}^* \psi_{k,n} d^3k = \sum_{k,n} w_{k,n} \psi_k^* \psi_k$$

- weights  $w_{k,n}$  depend on k and bandindex n (occupation)

- for full “bands” the weight is given by “symmetry”

- $w(\Gamma)=1, w(x)=2, w(\Delta)=4, w(k)=8$

- → shifted “Monkhorst-Pack” mesh

- for partially filled bands (metals) one must find the Fermi-energy (integration up to NE) and determine the weights for each state  $E_{k,n}$

- linear tetrahedron method (TETRA, eval=999)

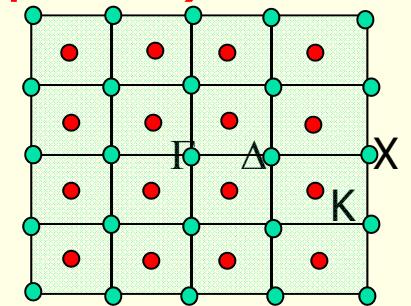
- linear tetrahedron method + “Bloechl” corrections (TETRA)

- “broadening methods”

- gauss-broadening (GAUSS 0.005)

- temperature broadening (TEMP/TEMPS 0.005)

- broadening useful to damp scf oszillations, but dangerous (magnetic moment)





# k-mesh generation



- **x kgen** (generates k-mesh and reduces to irreducible wedge using symmetry)
  - *automatically "adds inversion"*
    - time inversion holds and  $E(k) = E(-k)$
    - except in magnetic spin-orbit calculations (**x -so kgen**; uses `case.ksym` file)
    - **x -fbz kgen** (generates „full mesh“ in BZ)
  - *always "shift" the mesh for scf-cycle*
    - gaps often at  $\Gamma$  ! (might not be in your mesh)
  - *small unit cells and metals require large k-mesh (1000-100000)*
  - *large unit cells and insulators need only 1-10 k-points*
  - *use at first a fairly **coarse** mesh for scf/relaxations*
  - *continue **later** with **finer** mesh*
    - mesh was good if nothing changes and scf terminates after few (3) iterations
  - *use even finer meshes for DOS, spectra, optics,...*

- All programs are executed via the „master“ shell-script `x_lapw`

`x lapw2 -up -orb`

- This generates a „def“ file: `lapw2.def`

```
5,'tin.in2c',      'old',     'formatted'  
6,'tin.output2up', 'unknown', 'formatted'  
8,'tin.clmvalup', 'unknown', 'formatted'  
10,'./tin.vectorup','unknown', 'unformatted'
```

- and executes: `lapw2c lapw2.def`

- All WIEN2k-shell scripts have long and short names:

- `x_lapw; runsp_lapw, runfsm_lapw` → `x; runsp; runfsm`

- All scripts have a „help“ switch „-h“, which explains flags and options (without actually execution)

`x -h`      `x lapw1 -h`



# scf-cycle

## ■ run\_lapw [options]

(for nonmagnetic cases)

■ -ec 0.0001	<i>convergence of total energy (Ry)</i>
■ -cc 0.0001	<i>convergence of charge distance (e<sup>-</sup>)</i>
■ -fc 1.0	<i>convergence of forces (mRy/bohr)</i>
■ -it (-it1,-it2 , -noHinv)	<i>iterative diagonalization (large speedup)</i>
■ -p	<i>parallel calculation (needs .machines file)</i>
■ -SO	<i>add spin-orbit (only after „init_so“)</i>
■ <i>Spacegroups without inversion use automatically lapw1c, lapw2c (case.in1c,in2c)</i>	

## ■ case.scf: master output file, contains history of the scf-cycle

■ *most information is stored with some „labels“ (grep :label case.scf)*

■ :ENE	:DIS	:FER	:GAP	:CTO001	:NTO001	:QTL001
■ :FOR002:	2.ATOM		19.470	0.000	0.000	19.470
■ :FGL002:	2.ATOM		13.767	13.767	0.000	total forces
■ :LAT	:VOL		:POSxxx			



# Getting help



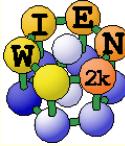
- **\*\_lapw -h**                    „help switch“ of all WIEN2k-scripts
- **help\_lapw:**
  - opens *usersguide.pdf*; Use *^f keyword* to search for an item („index“)
- **html-version of the UG:** (\$WIENROOT/SRC\_usersguide/usersguide.html)
- **[http://www.wien2k.at/reg\\_user](http://www.wien2k.at/reg_user)**
  - *FAQ page with answers to common questions*
  - *Update information: When you think the program has an error, please check newest version*
  - *Textbook section: DFT and the family of LAPW methods by S.Cottenier*
  - **Mailing-list:**
    - **subscribe** to the list (always use the same email)
    - **full text search** of the „digest“ (your questions may have been answered before)
    - **posting questions:** Provide sufficient information, locate your problem (case.dayfile, \*.error, case.scf, case.outputX).
    - „**My calculation crashed. Please help.**“ This will most likely not be answered.



# most common problems



- „QTL-B“ value too large - STOP (or :WARN): “ghostbands”
  - identify for which **eigenvalue**, **atom** and  $\ell$  it happens, check  $E_F$  (`case.scf2`, `case.output2`)
  - identify the corresponding linearization energies in `case.scf1`
  - change the corresponding linearization energy in `case.in1`
    - compare and check with :EPL and :EPH lines in `case.scf2`
    - default E-parameters are adapted automatically but may need changes for
      - surfaces, molecules (negative EF) or heavy elements (EF often larger than 1.0)
      - add a local orbital (or adjust its energy)
  - if QTL-B occurs for an atom with large RMT, reduce RMT
    - this may happen for larger RKMAX („numerical linear dependency“)
- scf-cycle diverges (grep :DIS `case.scf`):
  - check structure (most likely a wrong structure caused divergence);
  - reduce mixing in `case.inm` slightly; `rm *.broyd* case.scf; x dstart`
  - check E-parameters (see above), check :NEC01 (correct number of e-)



case.in1



set  $E$  to  $E_F - 0.2$  Ry

■ WFFIL

**EF=0.634**

■ 7.00

10

4

(WFPRI, SUPWF)

(R-MT\*K-MAX; MAX L IN WF, V-NMT)

■ 0.30

5 0

global E-param with N other, napw

■ 0 0.30

0.000 CONT 1

Es

■ 0 -3.72

0.005 STOP 1

Es-LO with search

■ 1 -2.07

0.010 CONT 1

Ep with search

■ 1 0.30

0.000 CONT 1

Ep-LO

■ 2 0.30

0.010 CONT 1

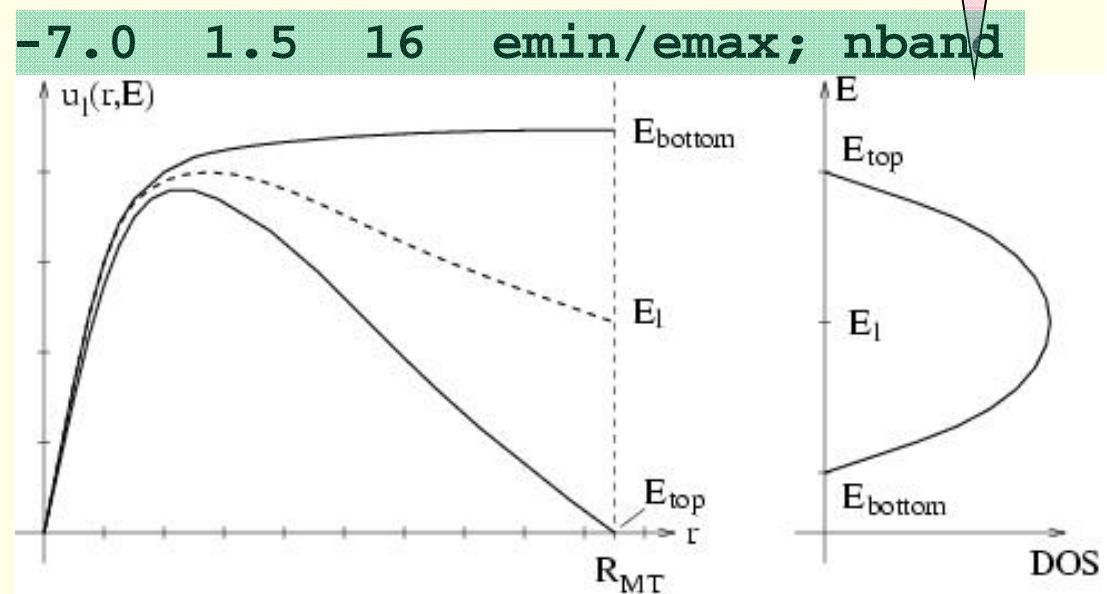
0/1...LAPW/APW+lo

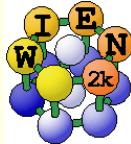
■ K-VECTORS FROM UNIT:4

$$\Psi = \sum_{K_n}^{KMAX} c_{K_n} e^{iK_n r}$$

$$\Phi_{K_n} = \sum_l^{l \text{ max}} A_{lm} u_l(E_l, r) Y_{lm}$$

$$H_{n,m}^{NS} = \langle \Phi_l | V_{LM}^{NS} | \Phi_{l'} \rangle$$





# case.klist, case.in2

- ```
■ GAMMA          0      0      0     40   1.0   IX, IY, IZ, IDIV, WEIGHT  
■                 1      0      0     40   6.0  
■ ...  
■ X              40      0      0     40   3.0  
■ END
```

## case.in2:

- TOT (TOT, FOR, QTL, EFG, FERMI)
  - -9.0 16.0 0.50 0.05 EMIN, NE, ESEPARMIN, ESEPARO
  - TETRA 0.000 (GAUSS, ROOT, TEMP, TETRA, ALL eval)
  - 0 0 4 0 4 4 6 0 6 4
  - 0 0 4 0 4 4 6 0 6 4
  - 14. GMAX(for small H set it to 20-24)
  - FILE FILE/NOFILE write reciprlist

$$\rho(r) = \sum_{LM} \rho_{LM}(r) Y_{LM}(\hat{r}) \quad \rho(r) = \sum_G \rho_G e^{iG r}$$



# Properties with WIEN2k - I



## ■ Energy bands

- *classification of irreducible representations*
- *'character-plot' (emphasize a certain band-character)*

## ■ Density of states

- *including partial DOS with l and m- character (eg.  $p_x, p_y, p_z$ )*

## ■ Electron density, potential

- *total-, valence-, difference-, spin-densities,  $\rho$  of selected states*
- *1-D, 2D- and 3D-plots (Xcrysden)*
- *X-ray structure factors*
- *Bader's atom-in-molecule analysis, critical-points, atomic basins and charges*  
 $(\nabla \rho \cdot \vec{n} = 0)$
- *spin+orbital magnetic moments (spin-orbit / LDA+U)*

## ■ Hyperfine parameters

- *hyperfine fields (contact + dipolar + orbital contribution)*
- *Isomer shift*
- *Electric field gradients*



# partial charges “qtl” + DOS



- be sure to have `case.vector` on a dense tetrahedral mesh after a scf calculation

- eventually:

- `x kgen`
  - edit `case.in1` (larger Emax)
  - `x lapw1`

- `x lapw2 -qtl`

$$\Psi_n * \Psi_n = 1 = q_{out} + \sum_t^{at} \sum_l q_{t,l}$$

- `case.output`

- integrated DOS

- `case.dos1ev (3ev)`

- text-file for plotting
  - $E$ -zero at  $E_F$

Session: TiC  
/susi/pbla/lapw/TiC

## Density of states

- `x lapw2 -qtl` Calculate partial charges  interactively
- `edit TiC.int` Edit input-file for TETRA
- `x tetra` Calculate partial DOS  interactively
- `edit TiC.outputt` Check output of TETRA
- `dosplot` Plot DOS

Session: TiC  
/susi/pbla/lapw/TiC/TiC.int

## File:

/susi/pbla/lapw/TiC/TiC.int

continue with DOS

Save

Download this file:

## Header from TiC.qtl:

```
ATOM 1 tot,0,1,2,3,xdos(i,j),j=1,i),i=1,1xdos2)
ATOM 2 tot,0,1,2,D-eg,D-t2g,3
```

## Title

-0.50 0.002 1.500 0.003

3

0  
1  
2

1  
2  
5

total  
Atom1-s  
Atom2-eg

EMIN, DE, EMAX, Gauss-broadening(>;de)

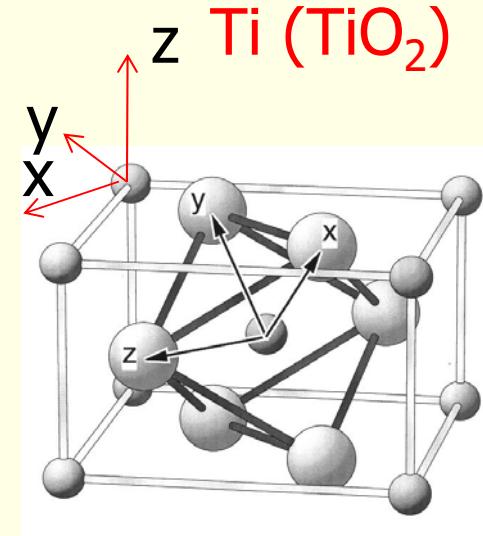
NUMBER OF DOS-CASES specified below

atom, case=column in qtl-header, label

## ■ local rotation matrix:

- transfers  $z$  ( $y$ ) into highest symmetry
- reduces terms in  $LM$  series
- "chemical" interpretation
  - $p_x$  is different from  $p_y$

$$\begin{pmatrix} 1/\sqrt{2} & 1/\sqrt{2} & 0 \\ -1/\sqrt{2} & 1/\sqrt{2} & 0 \\ 0 & 0 & 1 \end{pmatrix}$$



- see *case.struct* and *case.outputs*

## ■ x qtl (instead of x lapw2 -qtl)

- **f-orbitals**
- **qls for different coordinate system** (eg. "octahedral" in  $TiO_2$ )
- **relativistic basis** ( $p_{1/2}$ - $p_{3/2}$  or  $d_{3/2}$ - $d_{5/2}$  splitting in so calculation)
- **for angular dependend TELNES (ISPLIT 88, 99)**



# Properties with WIEN2k - I



## ■ Energy bands

- *classification of irreducible representations*
- *'character-plot' (emphasize a certain band-character)*

## ■ Density of states

- *including partial DOS with l and m- character (eg.  $p_x, p_y, p_z$ )*

## ■ Electron density, potential

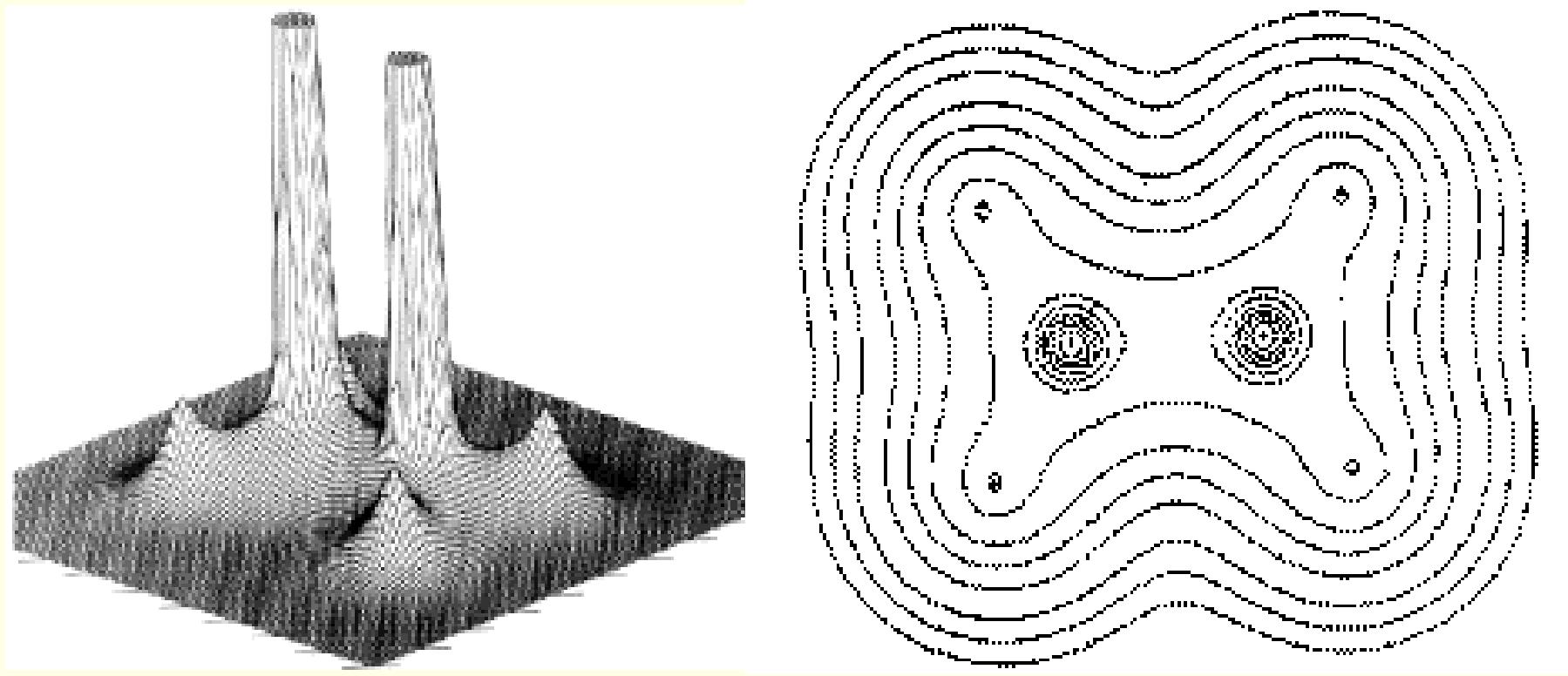
- *total-, valence-, difference-, spin-densities,  $\rho$  of selected states*
- *1-D, 2D- and 3D-plots (Xcrysden)*
- *X-ray structure factors*
- *Bader's atom-in-molecule analysis, critical-points, atomic basins and charges*  
 $(\nabla \rho \cdot \vec{n} = 0)$
- *spin+orbital magnetic moments (spin-orbit / LDA+U)*

## ■ Hyperfine parameters

- *hyperfine fields (contact + dipolar + orbital contribution)*
- *Isomer shift*
- *Electric field gradients*
- *NMR chemical shifts*

- Theory to characterize atoms and chemical bonds from the topology of the electron density, by R.F.Bader  
([http://www.chemistry.mcmaster.ca/faculty/bader/aim/aim\\_0.html](http://www.chemistry.mcmaster.ca/faculty/bader/aim/aim_0.html))

Electron density of  $\text{C}_2\text{H}_4$

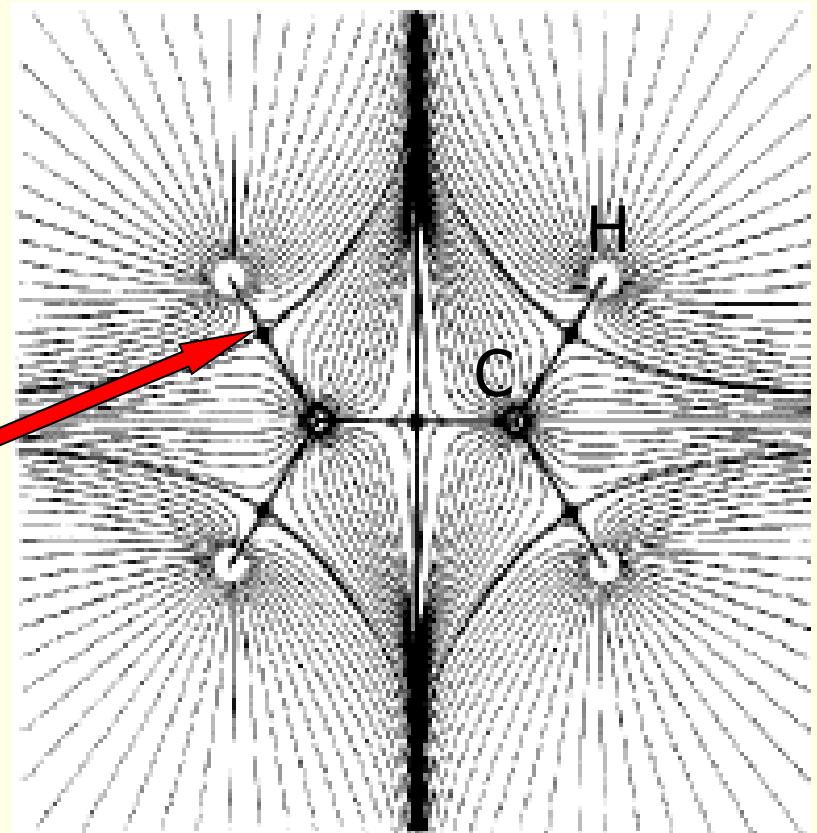


- Bonds are characterized by „critical points“, where  $\nabla\rho = 0$

- density maximum: (3,-3); 3 negative curvatures  $\lambda$ , (at nucleus or non-NM)
- bond CP: (3,-1): 2 negative, 1 positive  $\lambda$  (saddle point)
  - positive (and large) Laplacian: ionic bond
  - negative Laplacian: covalent bond
- bridge CP: (3,1)
- cage CP: (3,3) (minimum)

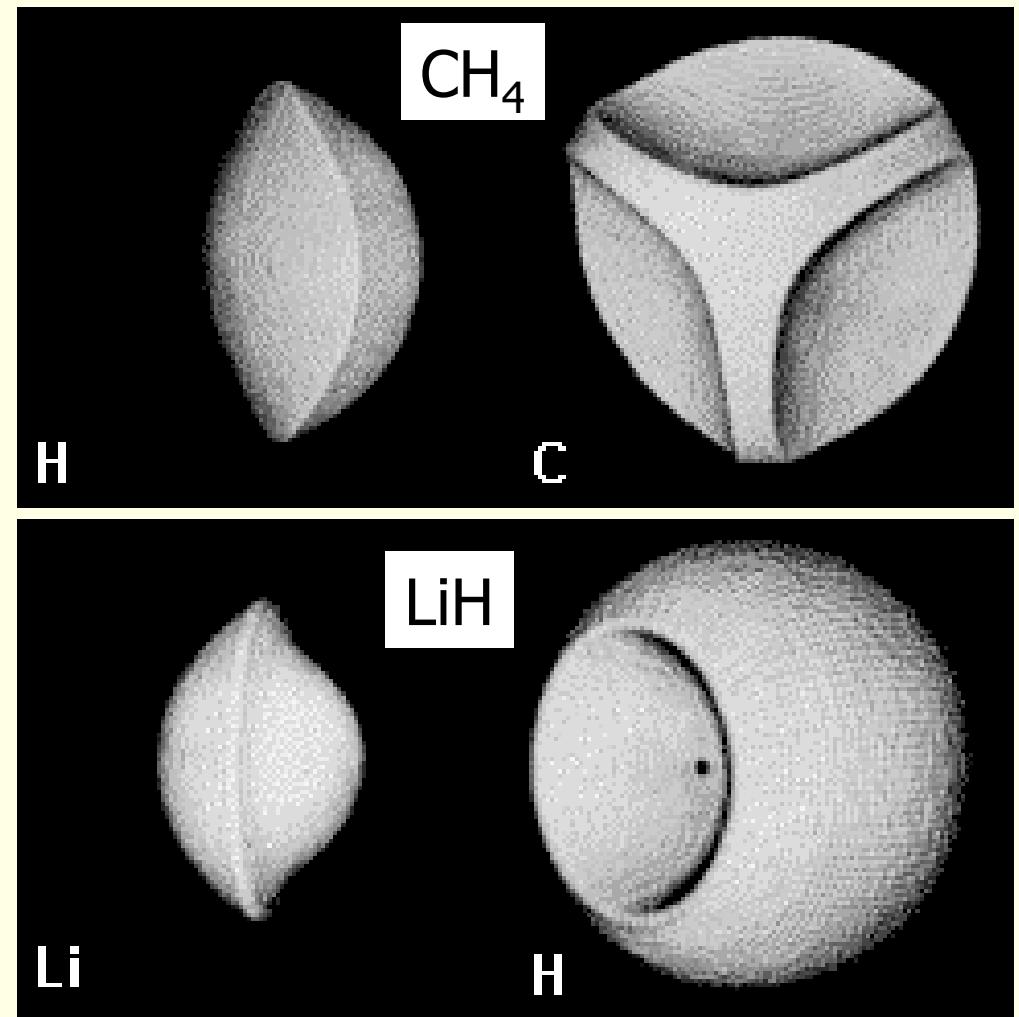
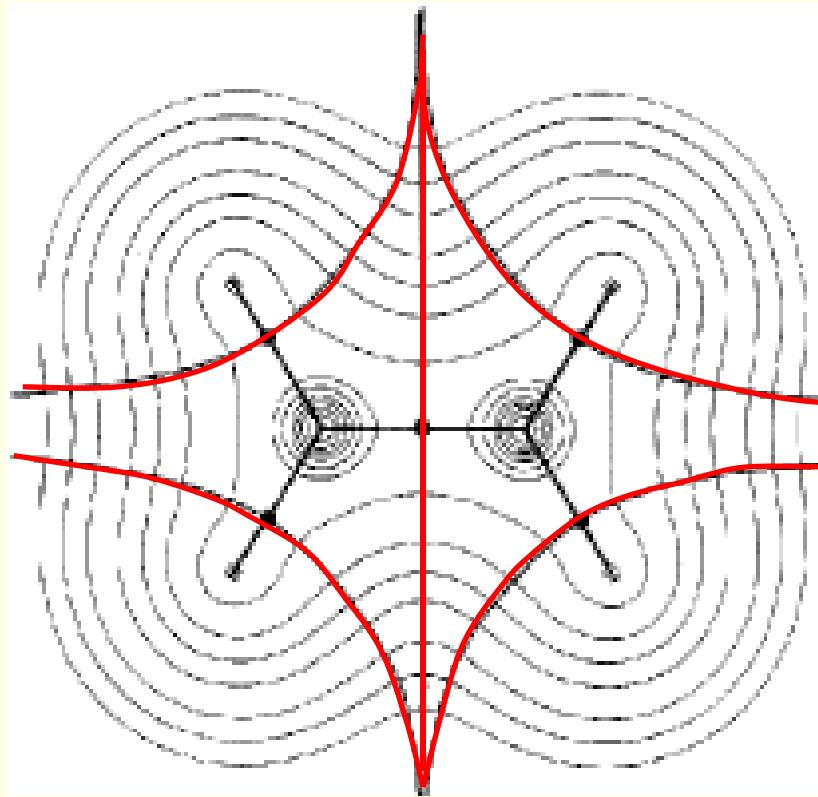
(3,-1) BCP

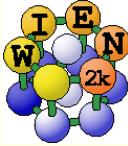
trajectories of constant  $\nabla\rho$   
originating at CPs in  $C_2H_4$



- “Atoms” are regions within a zero-flux surface  $\vec{\nabla}\rho \cdot \vec{n} = 0$

$\rho$  of  $\text{C}_2\text{H}_4$  with zero-flux lines defining atomic basins





- example of BN/Ni with “difference” to free atoms,
- workfunction shift
- Bader analysis of some inorganic compounds:

|                 | $\rho(\text{e}/\text{A}^3)$ | $\Delta\rho(\text{e}/\text{A}^5)$ | Q (e) |
|-----------------|-----------------------------|-----------------------------------|-------|
| Cl <sub>2</sub> | 1.12                        | -6.1                              | -     |
| I <sub>2</sub>  | 0.48                        | -0.9                              | -     |
| TiC             | 0.51                        | 1.8                               | 1.7   |
| TiN             | 0.47                        | 3.9                               | 1.7   |
| TiO             | 0.43                        | 5.8                               | 1.5   |
| KCl             | 0.08                        | 1.2                               | 0.6   |

Cl<sub>2</sub> more covalent  
then I<sub>2</sub>

more ionic, but less charge?  
less ionic then TiC ?



- You must have a “good” scf-density (case.clmsum)
  - no core leakage, LMs up to  $L=8-10$  in case.in2

## SURF

```
1          atom in center of surface (including MULT)
20 0.0 1.570796327 theta, 20 points, from zero to pi/2
20 0.0 0.785398163 phi, from 0 to pi/4 (depends on symmetry!!)
0.07 1.0 4      step along gradient line, rmin (has reached an atom)
1.65 0.1        initial R for search, step (a.u)
3 3 3           nshell
IRHO          "INTEGRATE" rho
WEIT          WEIT (surface weights are available in case.surf)
30            30 radial points outside min(RMIN,RMT)
END
```

---

## CRIT

```
1          atom around you search for critical points
ALL        two, three, four, all (dimers, trimers, .... all=2+3)
3 3 3        nshell
END
```

extractaim\_lapw: → critical\_points\_ang (converted units)  
:PC x, y, z,  $\lambda_1$ ,  $\lambda_2$ ,  $\lambda_3$ , ch, laplacian, rho



## ■ Total energy and forces

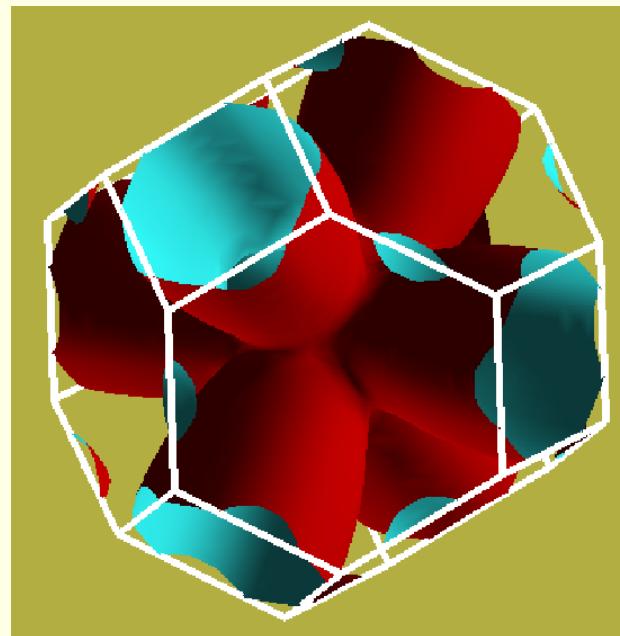
- *optimization of internal coordinates, (MD, BROYDEN)*
- *cell parameter only via  $E_{tot}$  (no stress tensor)*
- *elastic constants for cubic, hexagonal, and tetragonal cells*
- *Phonons via supercells*
  - interface to PHONON (K.Parlinski) – bands, DOS, thermodynamics, neutrons
  - interface to PHONOPY (A. Togo)
    - [http://www.wien2k.at/reg\\_user/unsupported](http://www.wien2k.at/reg_user/unsupported)

## ■ Spectroscopy

- *core level shifts*
- *X-ray emission, absorption, electron-energy-loss (with core holes)*
  - core-valence/conduction bands including matrix elements and angular dep.
- *optical properties (dielectric function in RPA approximation, JDOS including momentum matrix elements and Kramers-Kronig)*
- **fermi surface: 2D, 3D (using XcrysDen)**

- **`xcrysden --wien_fermisurface tin.struct`**

- choose a good k-mesh (eg. 10000 points)
  - plot the FS for all bands which cross  $E_F$  and compare to band structure



- for 2D plots there is also a WIEN2k-tool „`fsgen`“ (see UG)
  - SKEAF ([www.wien2k.at/reg\\_users/unsupported](http://www.wien2k.at/reg_users/unsupported)): quantum oszillations