

Defect thermochemistry: Spinney

18-04-24. 27th Wien2k Workshop. Trieste.

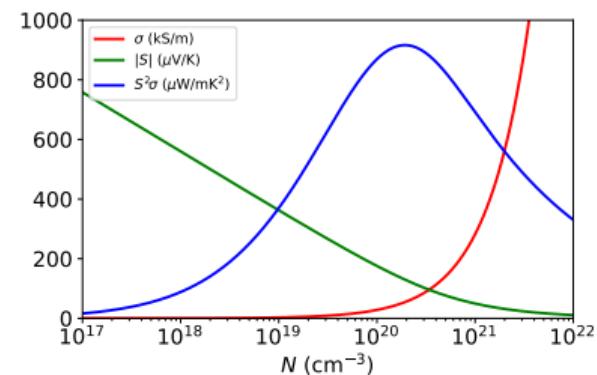
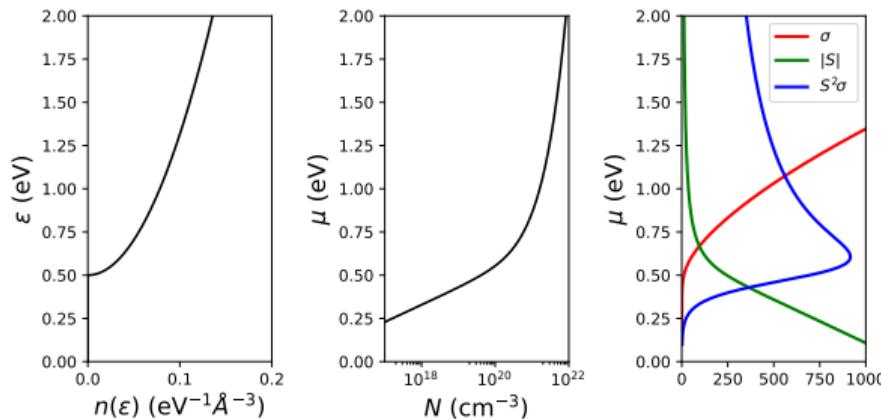


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1. Defects
2. Defect thermochemistry
3. Examples

Parabolic band. DOS, transport and doping

$$\varepsilon(k) = \varepsilon_{\text{CBM}} + \frac{\hbar^2 k^2}{2m^*} , \quad n(\varepsilon) = \frac{m^{3/2}}{\pi^2 \hbar^3} \sqrt{\frac{\varepsilon}{2}} , \quad \sigma(\varepsilon) = \frac{\sqrt{8m^*}}{\pi^2 \hbar^3} T \varepsilon^{\frac{3}{2}}$$



Defects according to their dimensionality

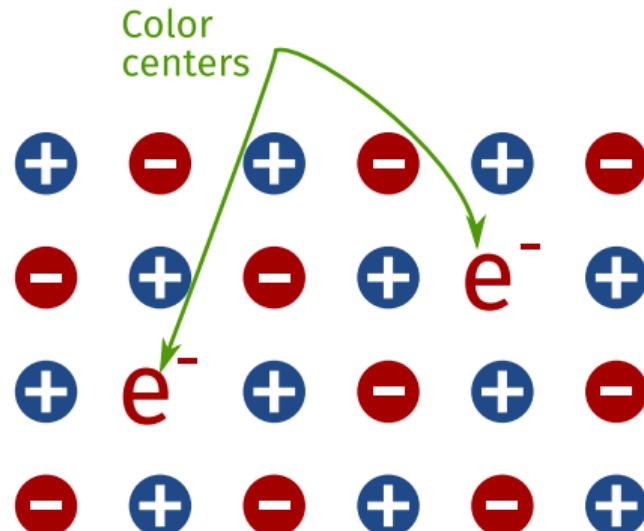
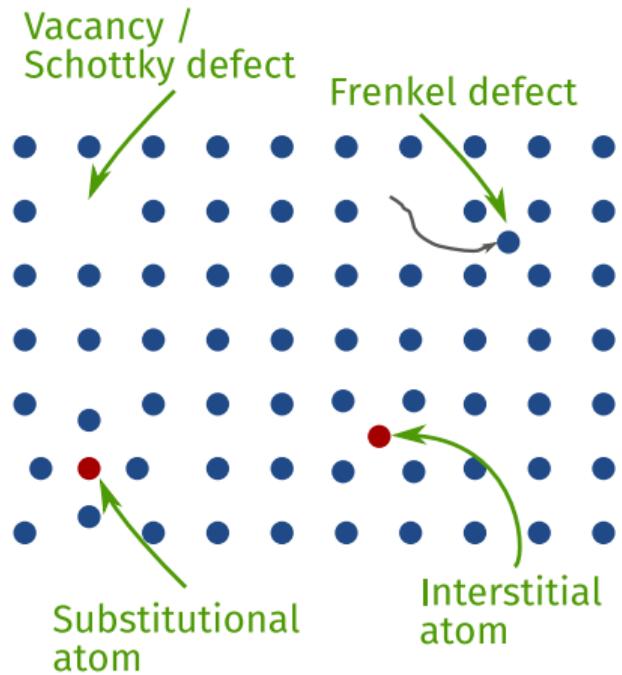
Defects are breakdowns of periodicity. Along each dimension, they can be:

Localized: size \leq a few lattice parameters

Extended: size \gg a few lattice parameters

Name	Localized/extended along	Examples
Point defects	3D/0D	Vacancies Interstitial atoms Substitutions Frenkel defects Small clusters of the above
Line defects	2D/1D	Dislocations Chains of point defects
Planar defects	1D/2D	Grain boundaries Twinning External boundaries
Bulk defects	0D/3D	Pores Inclusions Precipitates

Some point defects



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Equilibrium concentration of defects

Model:

- Crystal with N sites
- Cost in energy by formation of isolated point defect: E_d

Competition between energetic cost and entropy advantage when forming defects:

- Increase in energy due to n defects: nE_d
- $\Omega = \binom{N}{n} = \frac{N!}{(N-n)!n!}$ ways to arrange them
- Configurational entropy $S = k_B \log \Omega$
 $S \approx k_B[N \log N - n \log n - (N - n) \log(N - n)]$

$$V/\text{atom} \approx 10 \text{ \AA} \rightarrow 10^{23} \text{ atoms/cm}^3$$

E_D	n/N	c_D
1.0 eV	$\approx 10^{-5}$	$\approx 10^{-18} \text{ cm}^{-3}$
0.6 eV	$\approx 10^{-3}$	$\approx 10^{-20} \text{ cm}^{-3}$
0.4 eV	$\approx 10^{-2}$	$\approx 10^{-21} \text{ cm}^{-3}$

Defect concentration that minimizes F at constant T :

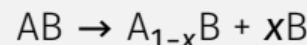
$$\frac{\partial F}{\partial n} = \frac{\partial(nE_d - TS)}{\partial n} = 0 \implies n/N \approx e^{-\frac{E_d}{k_B T}}$$

Defect formation energy

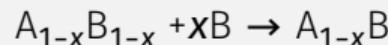
$$E_{D(q)}(\{\mu\}) = E_{f,D(q)} - \sum_{\alpha} n_{\alpha} \Delta \mu_{\alpha} + q \mu_e$$

Example: \square_A in AB.

A rich environment

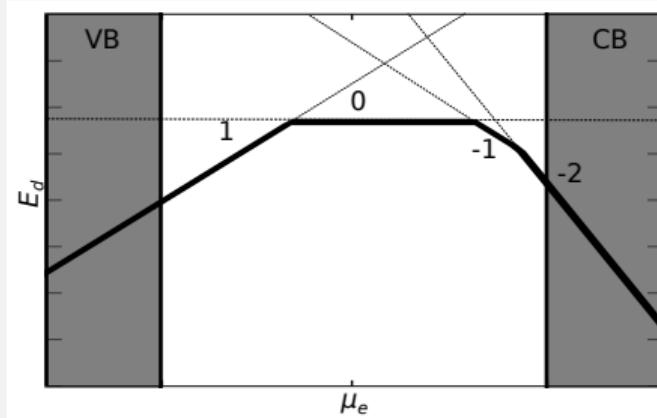


B rich environment



$$2\Delta H_f(AB) \leq \mu_A \leq 0$$

Charge chemical potential

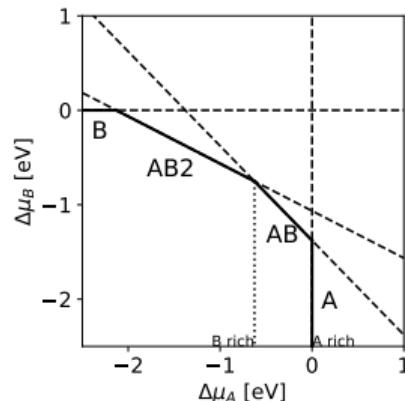
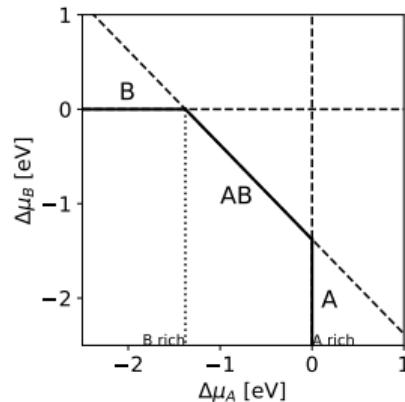


Chemical potentials of the constituents

$$\Delta\mu_A + \Delta\mu_B = 2\Delta H_{f,AB} \quad \Delta\mu_A \leq 0 \quad , \quad \Delta\mu_B \leq 0$$

Competing phase: AB_2

$$\Delta\mu_A + 2\Delta\mu_B = 3\Delta H_{f,AB_2}$$



Supercell approach to $E_{f,d}$

$$E_{f,D(q)} = E_{D(q)} - n_\alpha \mu_{\alpha,ref} + q \epsilon_{VBM} + E_{corr} - E_{bulk}$$

Correction scheme: Point charge at defect position embedded in dielectric constant of crystal (Kumagai, Oba 2014 and refs therein)

$$E_{corr} = E_{PC,q} - q \Delta V_{PC,q}$$

$E_{PC,q}$: Lattice of supercells with point charge at defect position

$\Delta V_{PC,q/b}$: Difference between defect induced potential and the point charge potential averaged over positions “far” from the defect

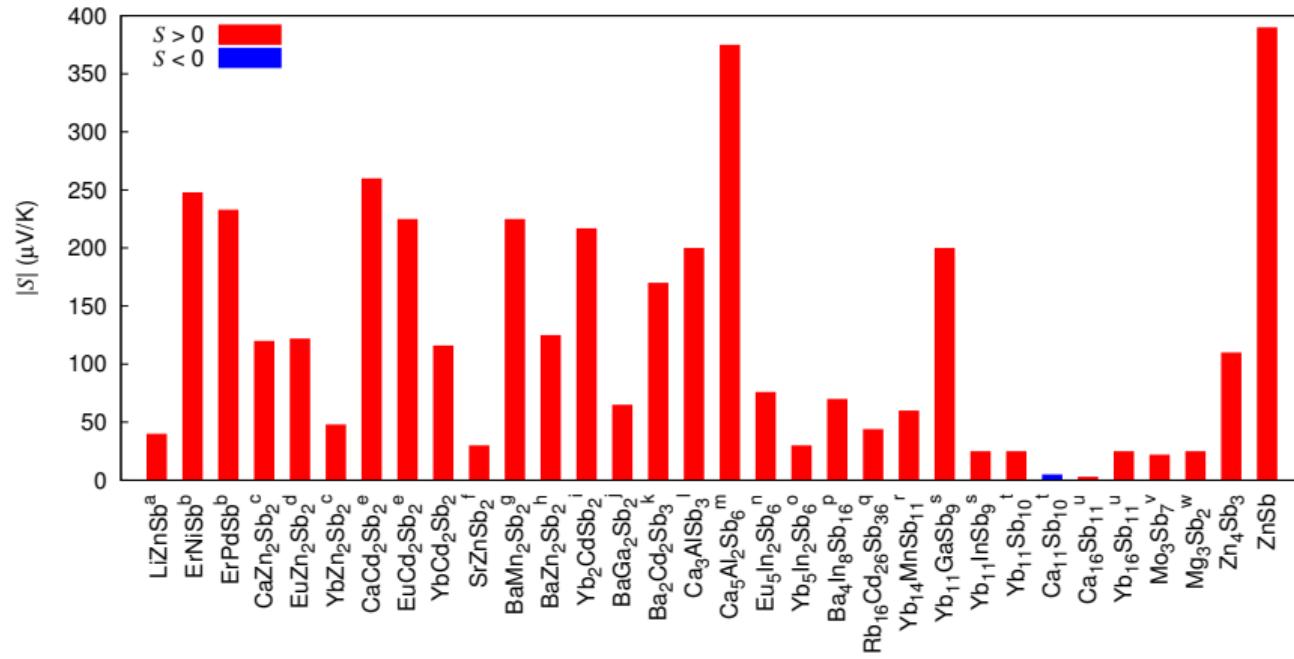
$$\Delta V_{PC,q/b} = V_{D(q)} - V_{bulk} - V_{PC,q}$$

Exercise: $\square_B^{(-3)}$:c-BN

- pip install spinney
- Download Spinney_VacB.tgz from W2K deposit
- Unpack and run script
- Reproduce the results
 - Setup supercells
 - Make defect and break symmetry in pristine cell
 - Initialize calculations. Note the need for a precise calculation of the Coulomb potential.
`lvns, gmax and R2V`
- Compare your `case.struct`, `case.in0` and `case.in2` to the ones from the deposit
- Run the calculation and compare your `case.scf`
- Modify the script to work on your calculation

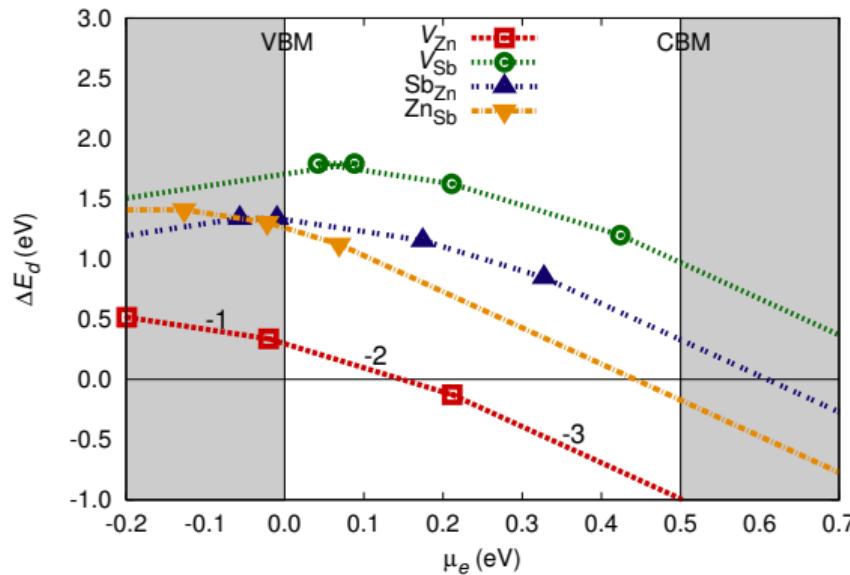
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Zintl Antimonides. Seebeck Coefficient



- Always positive Seebeck coefficient
- Binary and ternary Zintl antimonides always *p*-type

Intrinsic Defects. ZnSb



- Low formation energy of Zn vacancy pins chemical potential of the electrons to lower half of band gap
- Small stability window of ZnSb limits μ_a
- Always positive Seebeck coefficient

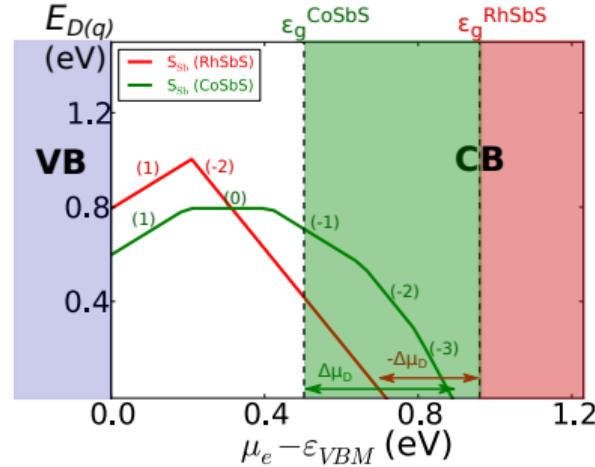
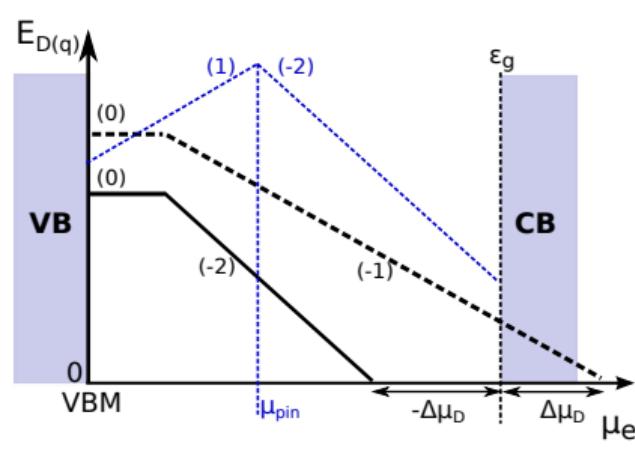
Automated search strategy

- Goal: Low-cost high-performance thermoelectric
- Computationally screen transport properties of innocuous, abundant sulfides
- Calculate bulk energy of competing phases
- Screen intrinsic defects, vacancies, antisites and interstitials, for potential doping limits
- Screen potential extrinsic potential defects, if no doping limits

Find ternary *n*-type sulfide to be used with *p*-type tetrahedrites

Ternary sulfides: Doping limits ?

Likelihood of intrinsic limit to n -doping given by stable hole producing killer defect



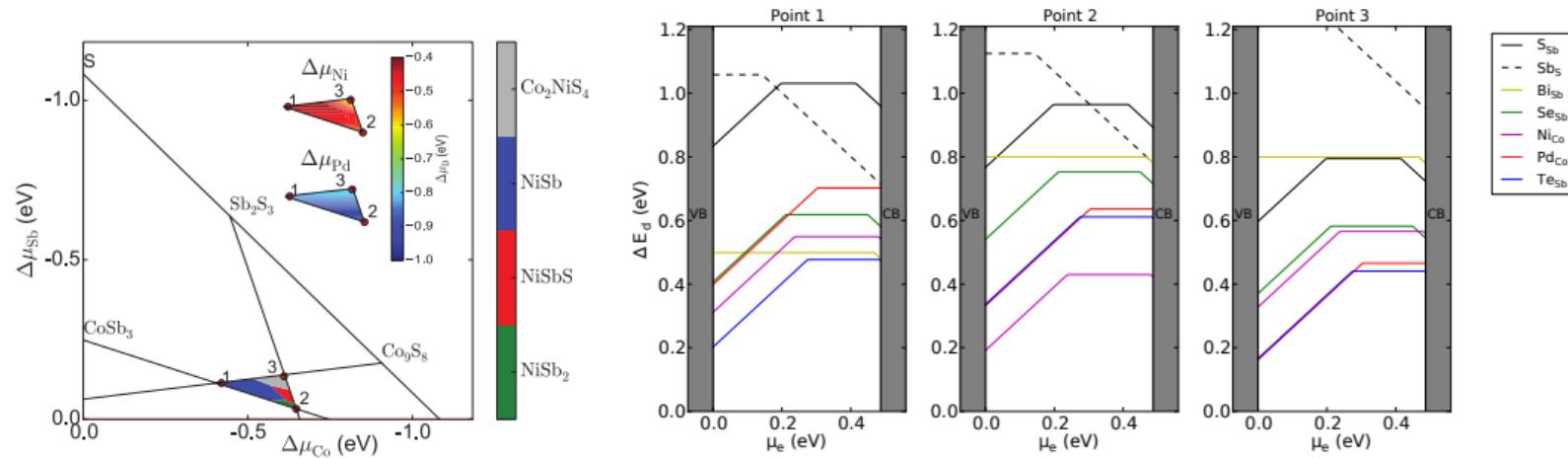
Compound	D	$\Delta\mu_D$ (eV)	ε_g (eV)
$\text{Ca}_2\text{Sb}_2\text{S}_5$	Vac_{Sb}	-0.33	2.11
RhSbS	S_{Sb}	-0.25	0.96
NaSbS_2	NaSb	-0.86	1.58
CuSbS_2	Vac_{Cu}	-0.80	1.05
CoSbS	S_{Sb}	0.38	0.50

CoSbS: Extrinsic doping

hydrogen 1 H 1.0079	Co, Sb, S												helium 2 He 4.0026				
3 Li 6.941	bad dopants (theory)												neon 10 Ne 20.180				
11 Na 22.990	good dopants (theo + exp)												argon 18 Ar 39.948				
19 K 39.098	calcium 20 Ca 40.078	scandium 21 Sc 44.966	titanium 22 Ti 47.867	vanadium 23 V 50.942	chromium 24 Cr 51.996	manganese 25 Mn 54.938	iron 26 Fe 55.845	cobalt 27 Co 58.933	nickel 28 Ni 58.693	copper 29 Cu 63.546	zinc 30 Zn 65.38	gallium 31 Ga 69.723	germanium 32 Ge 72.64	arsenic 33 As 78.96	selenium 34 Se 79.904	bromine 35 Br 83.798	krypton 36 Kr
rubidium 37 Rb 85.468	strontium 38 Sr 87.62	yttrium 39 Y 88.906	zirconium 40 Zr 91.224	niobium 41 Nb 92.906	molybdenum 42 Mo 95.96	technetium 43 Tc [98]	ruthenium 44 Ru 101.07	rhodium 45 Rh 102.91	palladium 46 Pd 106.42	silver 47 Ag 107.87	cadmium 48 Cd 112.41	indium 49 In 114.82	tin 50 Sn 118.71	antimony 51 Sb 121.76	tellurium 52 Te 127.60	iodine 53 I 126.90	xenon 54 Xe 131.29
caesium 55 Cs 132.91	barium 56 Ba 137.33	hafnium 72 Hf 178.49	tantalum 73 Ta 180.95	tungsten 74 W 183.84	rhodium 75 Re 186.21	osmium 76 Os 190.23	iridium 77 Ir 192.22	platinum 78 Pt 195.08	gold 79 Au 196.97	mercury 80 Hg 200.59	thallium 81 Tl 204.38	lead 82 Pb 207.2	bismuth 83 Bi 208.98	polonium 84 Po [209]	astatine 85 At [210]	radon 86 Rn [222]	
franum 87 Fr [223]	radium 88 Ra [226]	rutherfordium 104 Rf [261]	dubnium 105 Db [262]	seaborgium 106 Sg [266]	bohrium 107 Bh [264]	hassium 108 Hs [277]	meitnerium 109 Mt [268]	damarium 110 Ds [271]	roentgenium 111 Rg [272]								

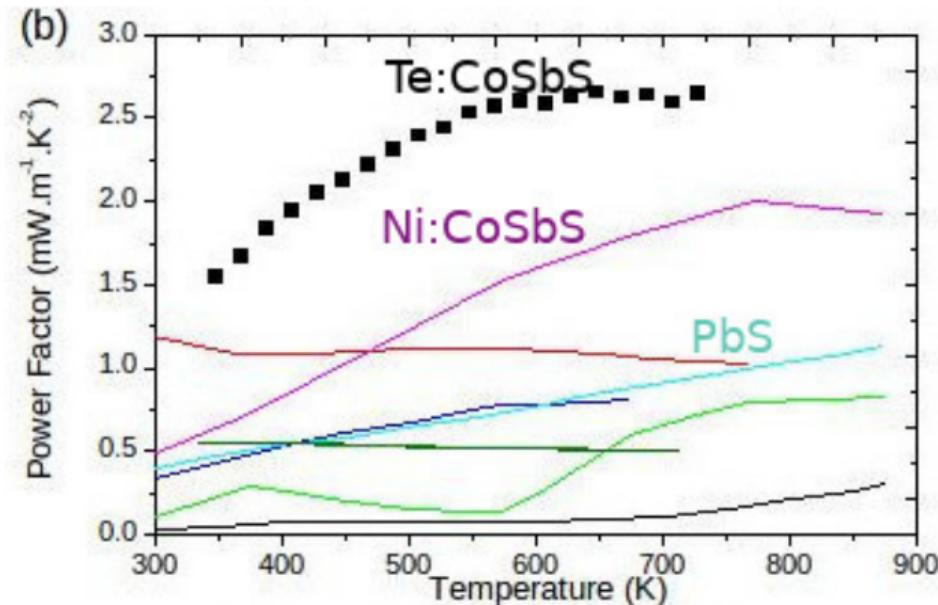
lanthanum 57 La 139.91	cerium 58 Ce 140.12	praseodymium 59 Pr 140.91	neodymium 60 Nd 144.24	promethium 61 Pm [145]	samarium 62 Sm 150.36	euroium 63 Eu 151.96	gadolinium 64 Gd 157.25	terbium 65 Tb 158.93	dysprosium 66 Dy 162.50	holmium 67 Ho 164.93	erbium 68 Er 167.26	thulium 69 Tm 169.93	ytterbium 70 Yb 173.05	lutetium 71 Lu 174.97
actinium 89 Ac [227]	thorium 90 Th 232.04	protactinium 91 Pa 231.04	uranium 92 U 238.03	neptunium 93 Np [237]	plutonium 94 Pu [244]	americium 95 Am [243]	curium 96 Cm [247]	berkelium 97 Bk [247]	californium 98 Cf [251]	einsteinium 99 Es [252]	fermium 100 Fm [257]	mendelevium 101 Md [258]	nobelium 102 No [259]	lawrencium 103 Lr [262]

CoSbS defects



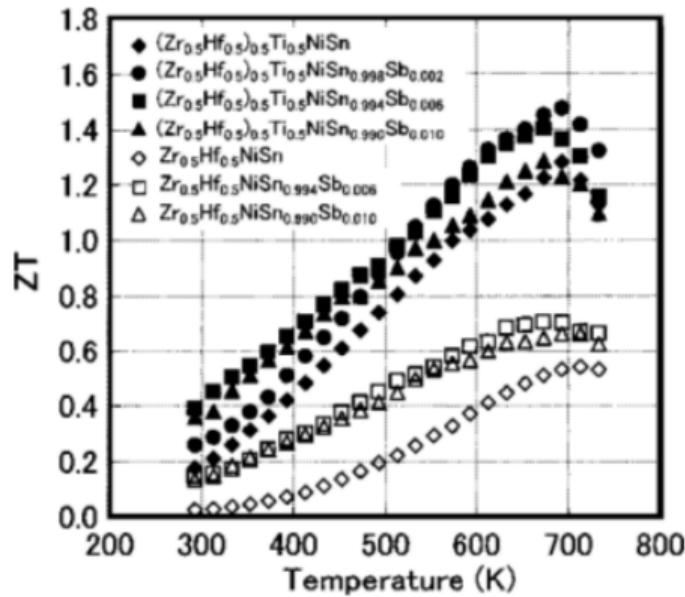
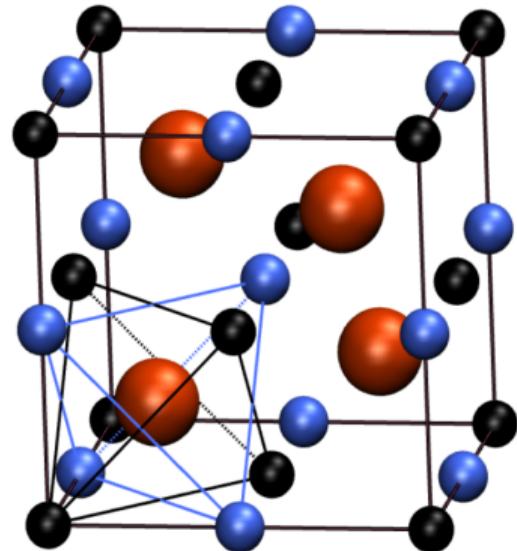
- PdSb₂ limiting competing phase in entire stability region when Pd doping
 - Work in Co AND Sb poor (sulfur rich) environment
- Ni-S phases set strict limits for Ni doping
- Te doping attractive in several limits

CoSbS powerfactor



- 150% increase in PF over best known sulfide
- Te doping: 50% increase in PF over Ni doped CoSbS

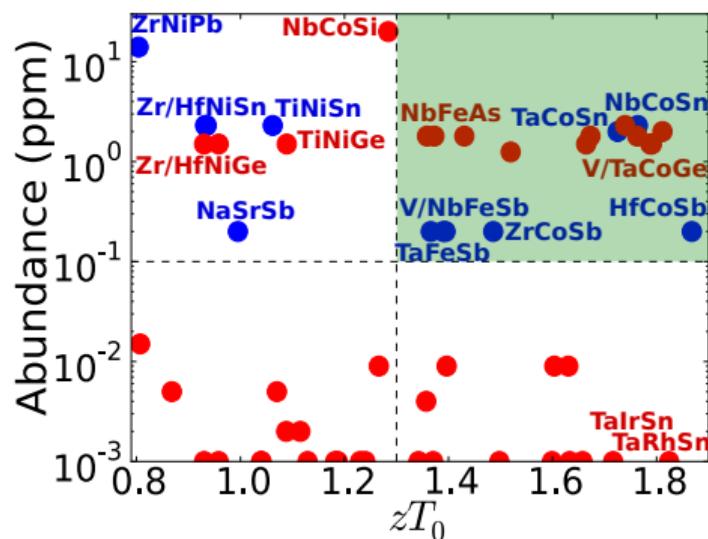
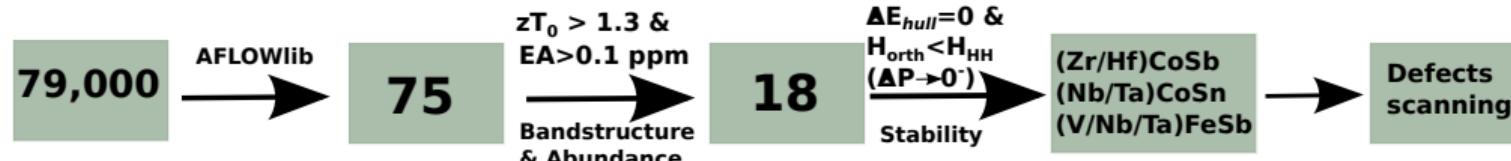
Half-Heusler thermoelectrics



Sakurada, Shutoh *Appl. Phys. Lett.* 86, p082105 (2005)

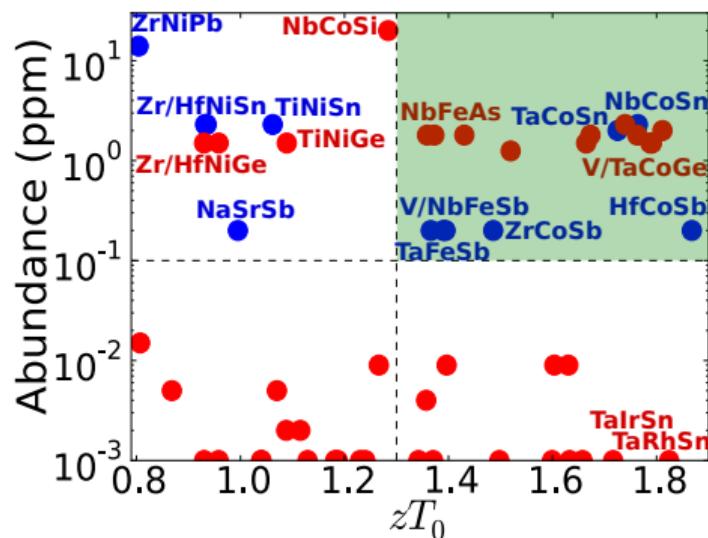
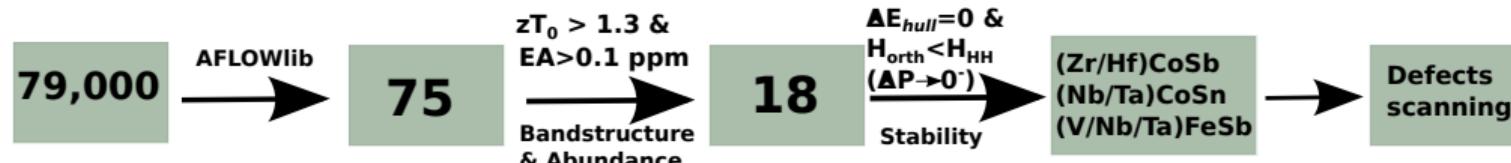
- 18 electron rule
- High intrinsic thermal conductivity
- Large chemical tunability

Screening: Band structure, abundance and stability.



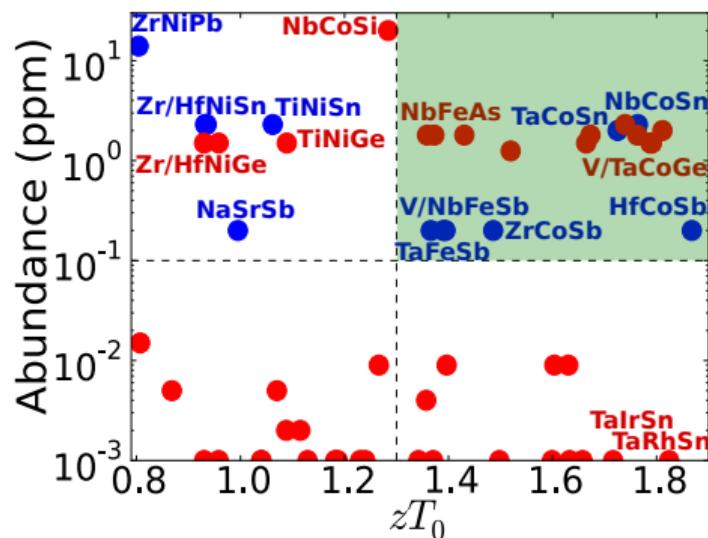
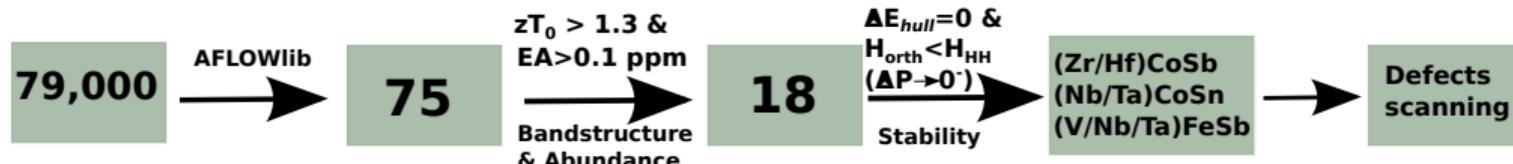
Compound	ΔE_{hull} (meV/atom)	orth phase
TaFeAs	33.03	yes
TaFeSb	0.00	no
NbFeAs	125.81	yes
NbFeSb	0.00	no
VFeSb	0.00	no
ZrCoAs	0.00	yes
ZrCoSb	0.00	no
WFeGe	64.34	no
NbCoGe	0.00	yes
HfCoAs	0.00	yes
TaCoSn	0.00	no
VCoSn	90.77	no
TiCoAs	0.00	yes
NbCoSn	0.00	no
TaCoGe	0.00	yes
VCoGe	0.00	yes
TaCoSi	91.47	yes
HfCoSb	0.00	no

Screening: Band structure, abundance and stability.



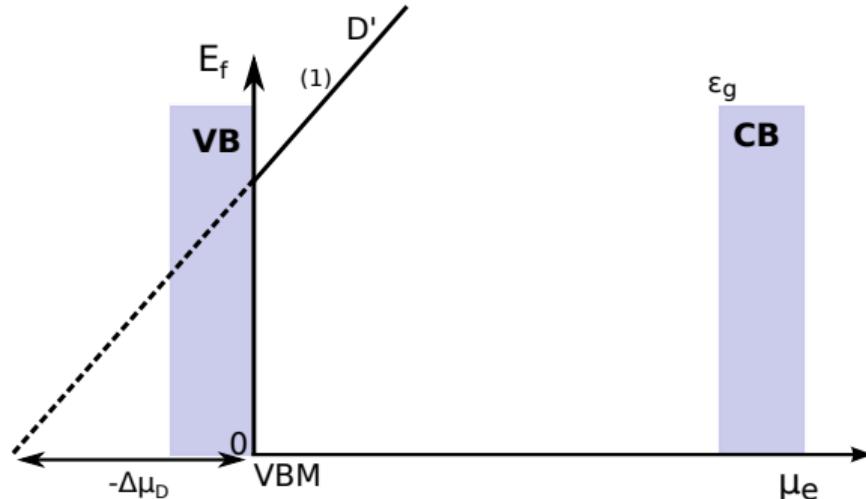
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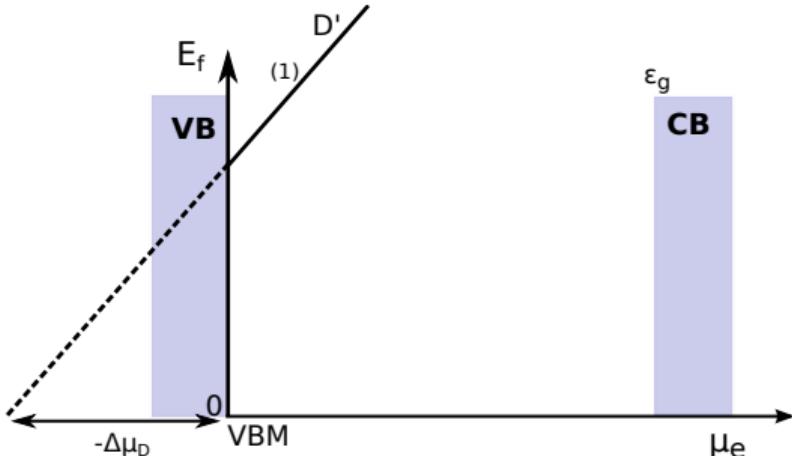
Intrinsic defects



	$\Delta\mu_D'$ (eV)	$D'^{(q)}$
NbCoSn	-0.27	$Co_{Int}^{(3)}$
VFeSb	-0.18	$Fe_{Int}^{(2)}$
ZrCoSb	-0.51	$Sb_{Zr}^{(1)}$
TaCoSn	-0.26	$Co_{Int}^{(2)}$
NbFeSb	-0.60	$Fe_{Int}^{(2)}$
HfCoSb	-0.37	$Co_{Int}^{(2)}$

No intrinsic doping limits

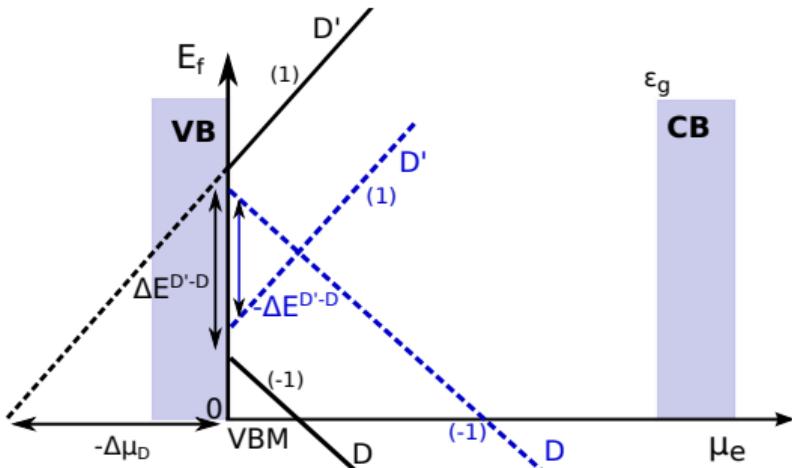
Extrinsic doping



	$E_f^D(\mu_{VB})$ (eV)	$D(q)$	$\Delta E^{D'-D}$ (eV)	$D'(q)$
NbFeSb	-0.01	Hf ⁽⁻¹⁾	1.20	Fe ⁽²⁾ Int
	0.09	Ti ⁽⁻¹⁾ Nb ⁽⁻¹⁾	1.10	Fe ⁽²⁾ Int
	0.21	Mn ⁽⁻¹⁾ Fe ⁽⁻¹⁾	0.72	Vac ⁽²⁾ Fe ⁽²⁾
	0.41	Zr ⁽⁻¹⁾ Nb ⁽⁻¹⁾	0.78	Fe ⁽²⁾ Int
	0.55	Sn ⁽⁻¹⁾ Sb ⁽⁻¹⁾	0.70	Fe ⁽²⁾ Int
ZrCoSb	0.17	Sc ⁽⁻¹⁾ Zr ⁽⁻¹⁾	0.60	Sb ⁽¹⁾ Zr ⁽¹⁾
	0.61	Sn ⁽⁻¹⁾ Sb ⁽⁻¹⁾	0.35	Vac ⁽¹⁾ Co ⁽¹⁾
	0.61	Fe ⁽⁻¹⁾ Co ⁽⁻¹⁾	-0.48	Vac ⁽¹⁾ Co ⁽¹⁾
NbCoSn	0.32	Hf ⁽⁻¹⁾ Nb ⁽⁻¹⁾	0.50	Co ⁽³⁾ Int
	0.62	Fe ⁽⁻¹⁾ Co ⁽⁻¹⁾	-0.15	Fe ⁽²⁾ Int
	0.68	Ti ⁽⁻¹⁾ Nb ⁽⁻¹⁾	0.14	Co ⁽³⁾ Int
	0.72	Zr ⁽⁻¹⁾ Nb ⁽⁻¹⁾	0.10	Co ⁽³⁾ Int
TaCoSn	0.44	Hf ⁽⁻¹⁾ Ta ⁽⁻¹⁾	0.33	Co ⁽²⁾ Int
	0.87	Fe ⁽⁻¹⁾ Co ⁽⁻¹⁾	-0.10	Fe ⁽²⁾ Int

- Known carrier inducing defects reproduced in NbFeSb and ZrCoSb
- A new system with favorable extrinsic dopants identified

Extrinsic doping



	$E_f^D(\mu_{VB})$ (eV)	$D(q)$	$\Delta E^{D'-D}$ (eV)	$D'(q)$
NbFeSb	-0.01	Hf ⁽⁻¹⁾	1.20	Fe ⁽²⁾ Int
	0.09	Ti ⁽⁻¹⁾ Nb ⁽⁻¹⁾	1.10	Fe ⁽²⁾ Int
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ZrCoSb	0.17	Sc ⁽⁻¹⁾ Zr ⁽⁻¹⁾	0.60	Sb ⁽¹⁾ Zr ⁽¹⁾
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	0.62	Fe ⁽⁻¹⁾ Co ⁽⁻¹⁾	-0.15	Fe ⁽²⁾ Int
	0.68	Ti ⁽⁻¹⁾ Nb ⁽⁻¹⁾	0.14	Co ⁽³⁾ Int
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TaCoSn	0.44	Hf ⁽⁻¹⁾ Ta ⁽⁻¹⁾	0.33	Co ⁽²⁾ Int
	0.87	Fe ⁽⁻¹⁾ Co ⁽⁻¹⁾	-0.10	Fe ⁽²⁾ Int

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- A new system with favorable extrinsic dopants identified