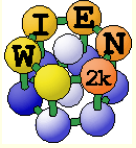


Exercises:



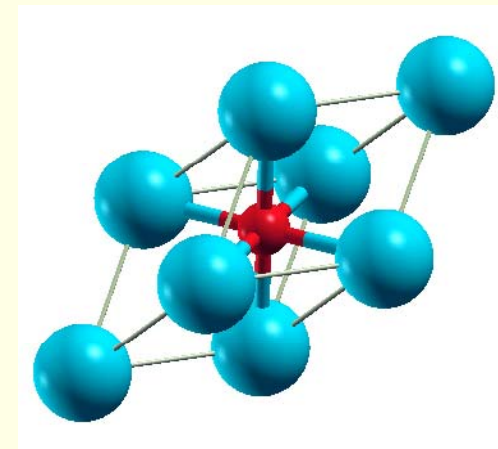
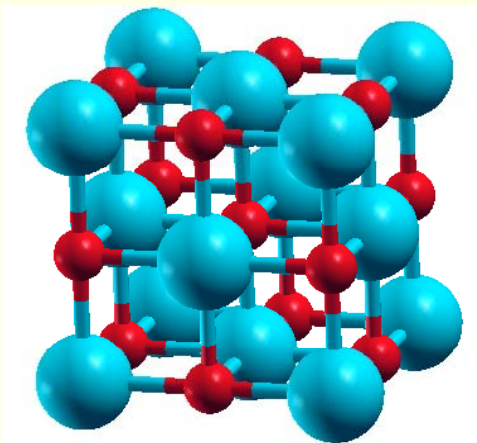
- In the following you find some suggestions for exercises, which teach you various tasks one may perform with WIEN2k.
- Please note, that often “computational parameters” are set to “minimal cpu-time” instead of “fully converged calculations”.
- Do not use such small values for final results and publications without convergence checks !!
- Account information (username/pw): kursXX / wien2k
 - *(for XX put the last 2 numbers shown on your pc)*



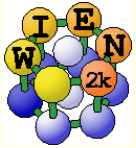
Exercise 1: Getting started:



- i) Open a terminal window.
- ii) Start w2web (accept all defaults, specify account (kursXX/wien2k))
- iii) Connect with firefox to w2web as indicated on the screen of ii)
- iv) Try the "quick-start" example for **TiN** (similar to TiC in the UG)
 - *Generate structure (a=4.235 Ang; reduce RMT by 1%)*
 - *view structure with Xcrysden (switch from primitive to conventional cell)*



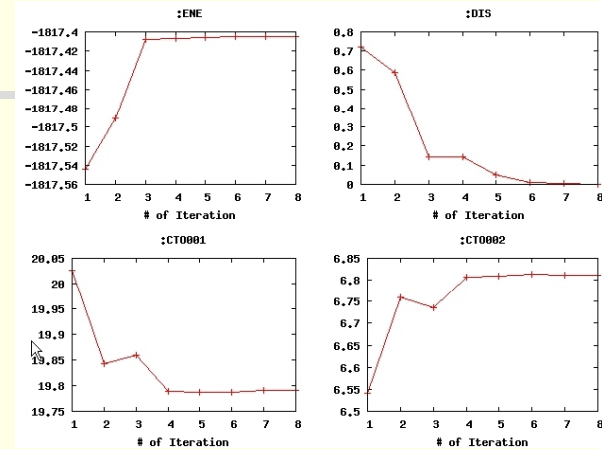
- *init_lapw (use defaults, 1000 k-points)*
- *scf-cycle (run_lapw, use defaults; monitor "STDOUT" and "dayfile")*
 - How many iterations did you need ? How long took a single scf-iteration ?



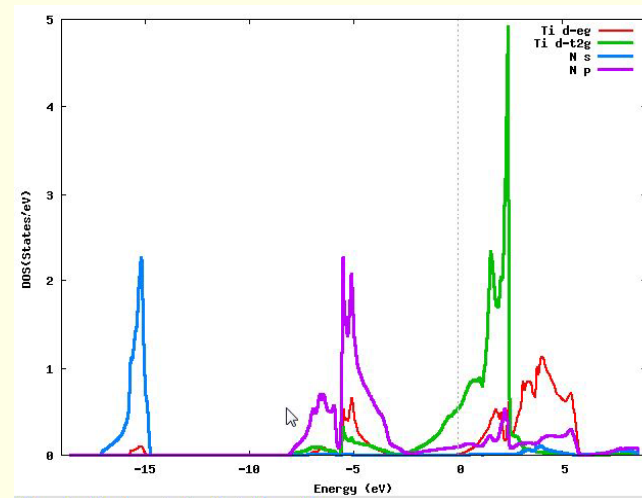
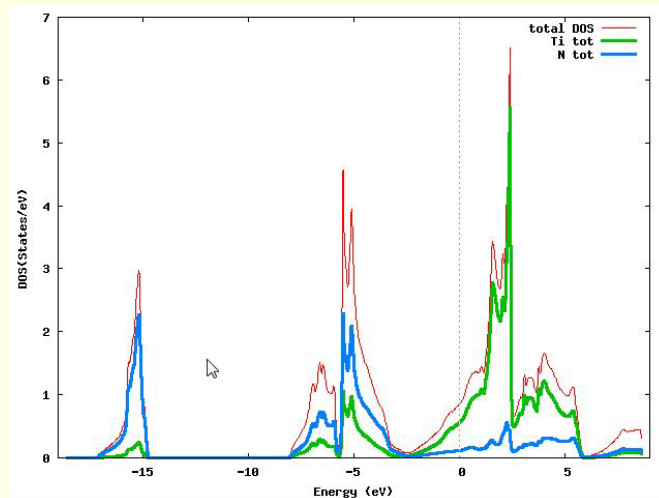
TiN continued



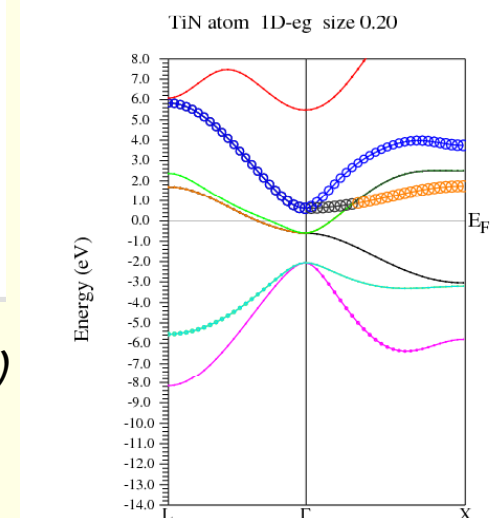
- *utilities: analyse*
 - (:ENE, :DIS, :CTO) graphically

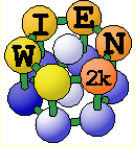


- *utilities: save_lapw (use as save-name: "TiN_exp_pbe_rk7_1000k")*
- *DOS (plot 7 cases: total + Ti-tot + N-tot and Ti-eg + Ti-t2g + N-s + N-p)*



- *bandstructure (along L-Gamma-X with "character plotting")*
(identify "t2g-" and "eg-" bands)

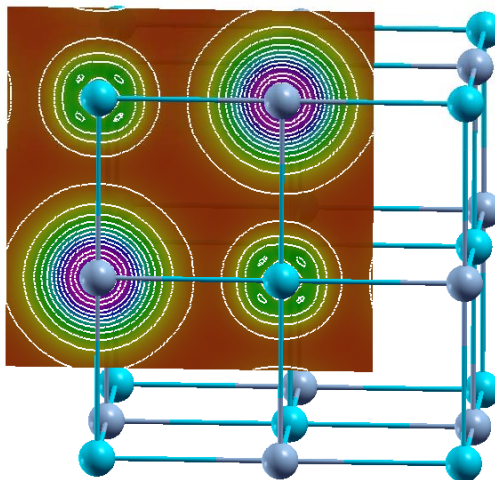




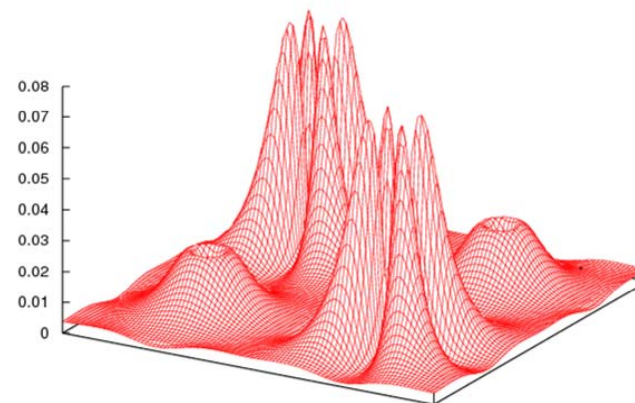
TiN continued ...

- *electron density* (use *xcrysden* to select the (100) plane), view it in *xcrysden* and *rhoplot* to “understand contour and 3D-plots”)
 - valence density (without semicore, check TiN.scf1 to find a EMIN which truncates the Ti-3s,3p states); compare the density around Ti with TiC (UG)
 - difference density (observe “charge transfer” and “t_{2g}-anisotropy” around Ti)
 - densities of the “N-p” and “occupied Ti-d-band” (get the corresponding E-intervals from DOS-plots (in Ry!), check UG how to modify TiN.in2 in order to obtain the density in a certain E-interval (search for “case.in2”); observe the e_g and t_{2g} asymmetry around Ti and the different N-p “weights”, explain the chemical bonding

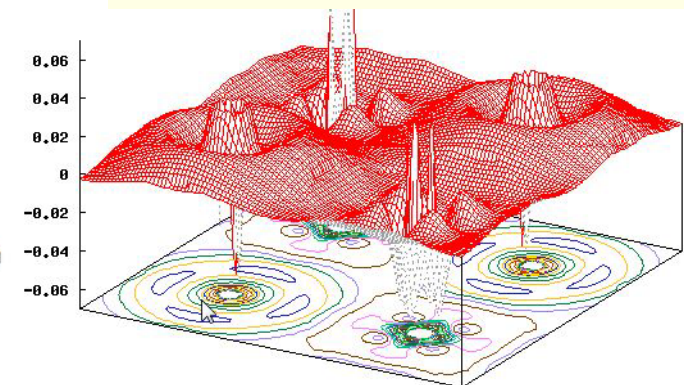
valence ρ

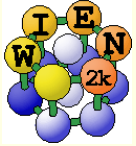


Ti-d band



difference density





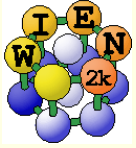
Exercises 2: Testing accuracy: RKmax and k-points



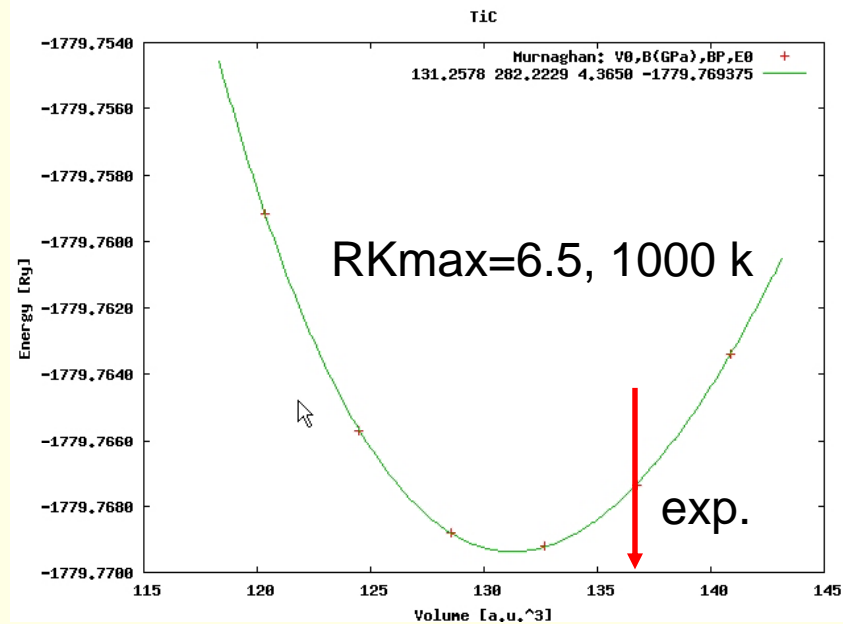
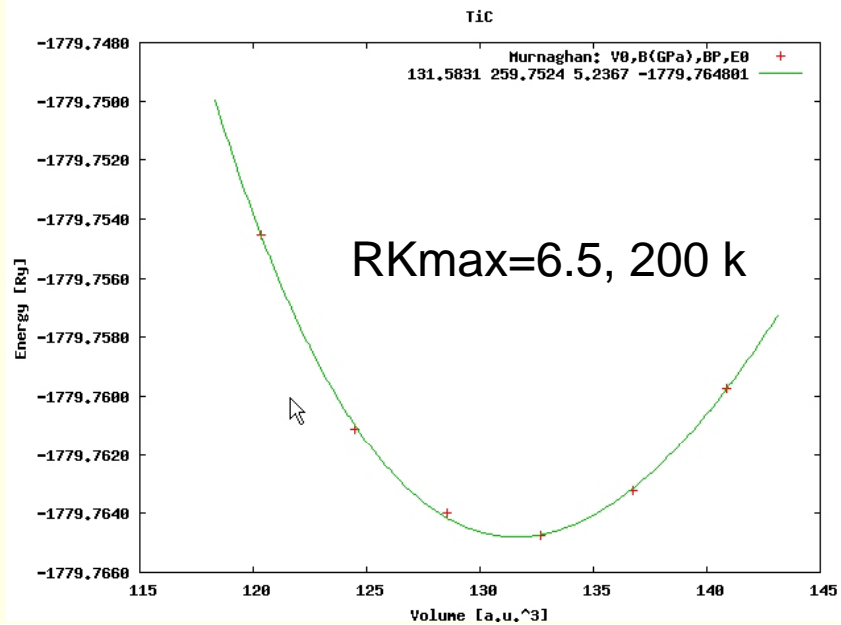
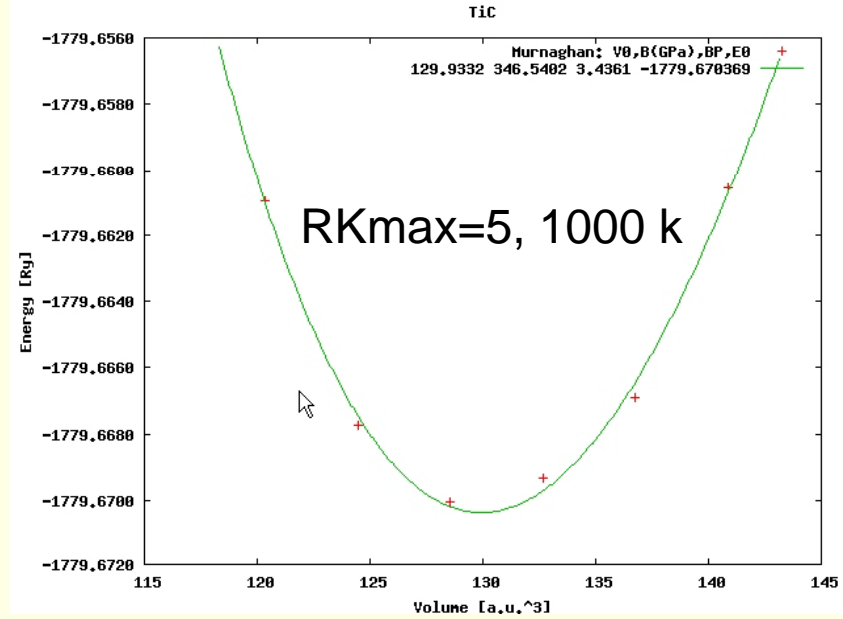
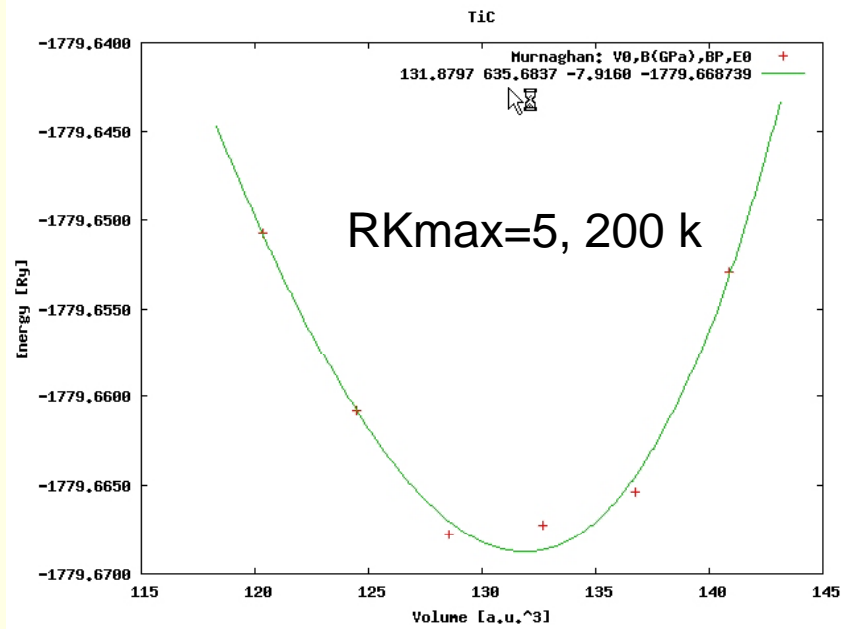
- TiC (fcc, $a=4.328$ Ang, setrmt – 4%, LDA) and calculate the equilibrium volume ("optimize") using:
 - (RKmax=5 200 k-points); (RKmax=6.5 200 k); (RKmax=5 1000 k); (RKmax=6.5 1000 k) and (RKmax=8 5000 k)
- a) run x optimize and generate 6 structures (-12, -9, -6, -3, 0, 3% volume change)
 - (because of LDA we expect 1 2% smaller lattice parameter (3 8% in volume) than experiment)
- b) set RKMAX=5 in TiC.in1 and x kgen with 200 k-points
- c) edit "optimize.job". Modify the "run_lapw" and "save_lapw" commands to:
 - `run_lapw -cc 0.001 -ec 0.0001`
 - `save_lapw $\{i\}$ _rk5_200k`
- d) run optimize.job, plot the results (using *rk5_200k)
- e) set RKMAX=6.5 in TiC.in1
- f) edit "optimize.job". Uncomment the cp line and modify:
 - `cp $\{i\}$ _rk5_200k.clmsum TiC.clmsum # Using previously converged densities saves a lot of CPU time!!`
 - ...
 - `save_lapw $\{i\}$ _rk6.5_200k`
- g) repeat step d) (plot the results for "*_rk6.5_200k")
- h) repeat steps above with proper modifications

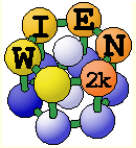
- Find out how RKmax and k-points lead to smooth/non-smooth curves. Estimate good values and compare in particular B and BP (Bulkmodulus and its volume derivative).
- Remember: Depending on the specific property you want to calculate (just a DOS, or Energy-Volume curves, or EFG, or structure optimization with forces,..) and the desired accuracy, the types of atoms, insulator/metal and system size you may need different RKmax and k-point samplings:
 - H: RKmax > 2.5; sp-elements: RKmax > 5; d-elements: RKmax > 6; f-elements: RKmax > 7;
 - 1 atom/cell, metal: 1000-10000 k-points or more
 - 1 atom/cell, insulator: 100-1000 k-points or more
 - For N atoms/cell you can reduce the k-mesh by a factor N

- Remember: Always test your **specific property** for convergence !!



Volume optimization for TiC



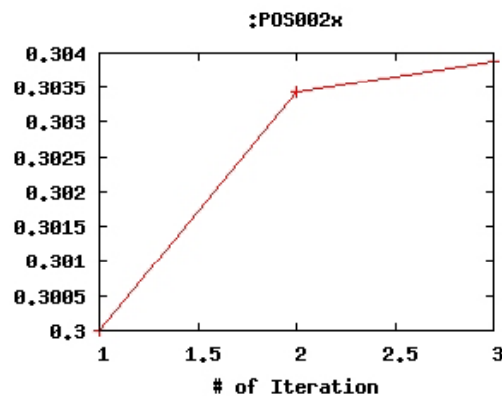
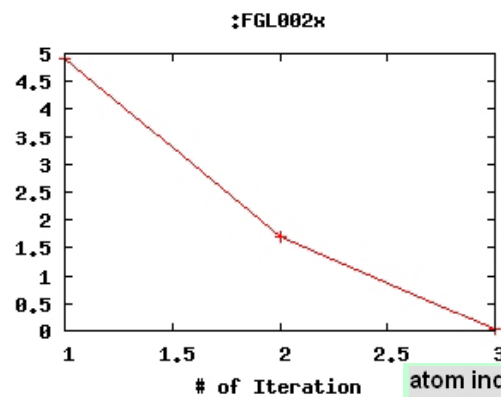
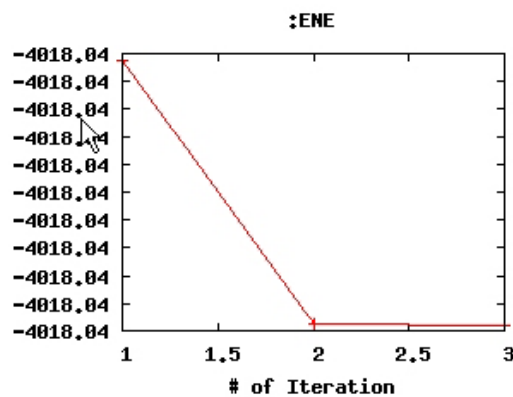


Exercise 3:



■ optimization of positions: rutile TiO_2 :

- $P42/mnm$ (136), $a=b=4.59$ $c=2.96$ Å; $\text{Ti}(0,0,0)$ $\text{O}(0.3,0.3,0)$; RMT-2%
- *init_lapw*: (100k, RKmax=6.5)
- *min_lapw* (use defaults in case.inM)
 - analyze :ENE :FGL002x :POS002x in case.scf_mini (exp.pos:0.3045)



atom independent parameters:

ENE FER DIS NEC-new NEC-old MMTOT

atom dependent parameters:

QTL EFG ETA CHA DTO CTO NTO

atom dependent vector parameters:

FOR FGL POS (x- y- z-coordinate for scfmonitor)

for spin polarized systems:

CUP CDN HFF MMI

other parameter:

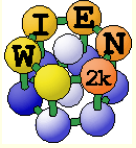
Select atom for atom dependent param. (0 means all atoms, up to 6 atoms possible)

Analysis of: TiO2.scf with 10 lines.

or of alternate scf-files: TiO2.scf_mini with 10 lines.

Analyze scf file

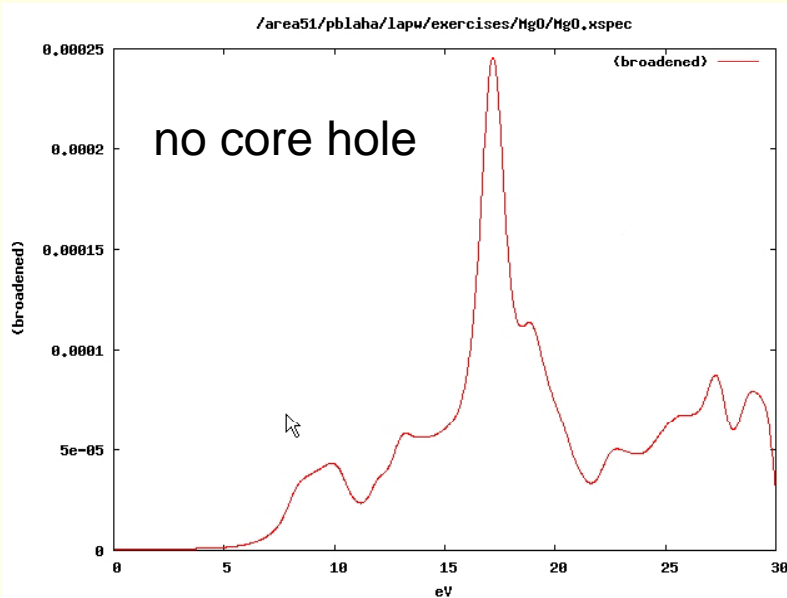
Graphics using scfmonitor (only for single scf file)



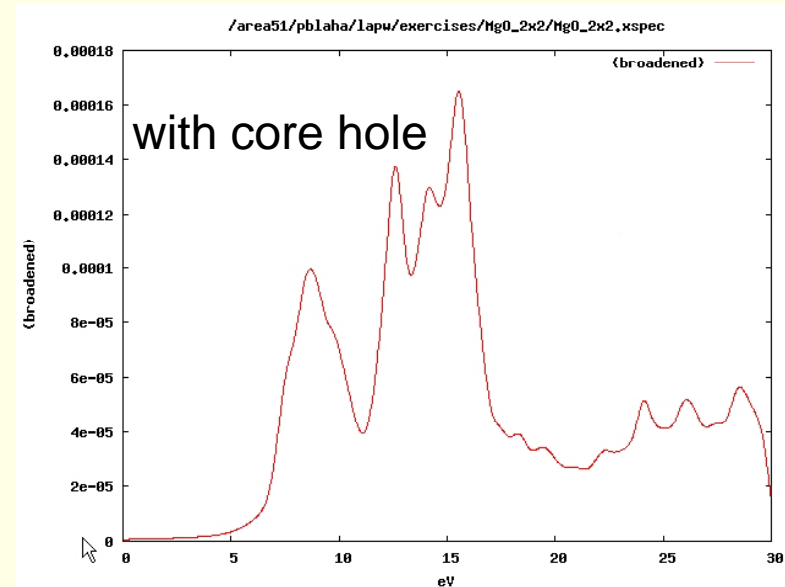
Exercise 4:

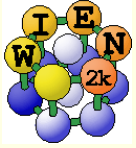


- **X-ray emission/absorption spectroscopy**
 - *MgO* (NaCl structure, $a=7.96$ bohr; default initialization with 1000 k-points; scf-cycle)
 - *XSPEC* task: Mg-K XAS (select larger EMAX in MgO.in1; spectrum from 0-30 eV, vary broadening)
- **Supercells: MgO 2x2x2 FCC-supercell for core-hole simulation**
 - create new "session", copy MgO.struct into new directory
 - x supercell; (specify proper struct-filename, 2x2x2, F-lattice)
 - cp supercell-struct file to correct name "case.struct"; "**label**" 1st atom (Mg \rightarrow Mg1)
 - *init_lapw* (with 200k, RKmax=6.5)
 - edit case.inc (remove a core electron from 1st atom)
 - edit case.in2 (add one valence electron)
 - *run_lapw -it* (use iterative diagonalization, compare timings for 1st and later iterations!)
 - edit case.in2 (remove extra valence electron)
 - *XSPEC* task for Mg-K XAS (see above)



Mg-K XAS



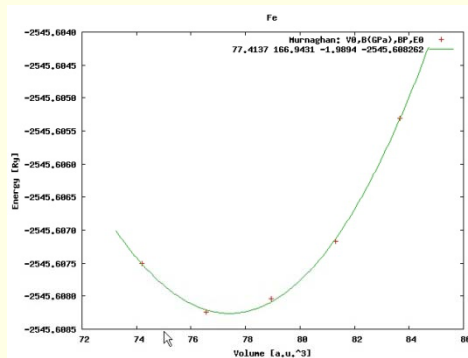


Exercise 5:



■ Magnetism: bcc Fe ($a_0=2.86 \text{ \AA}$)

- *setrmt: -3%; 5000k; spin-polarization:yes, use RKmax=7, then 8*
- *do a volume optimization (-6, -3, 0, 3, 6 %) (activate runsp_lapw instead of run_lapw !)*
 - *check equilibrium volume, :MMTOT as function of volume*



--- MMTOT ----- in 5 files:

Fe_vol__0.0_rk8_5000k.scf::MMTOT: 2.204

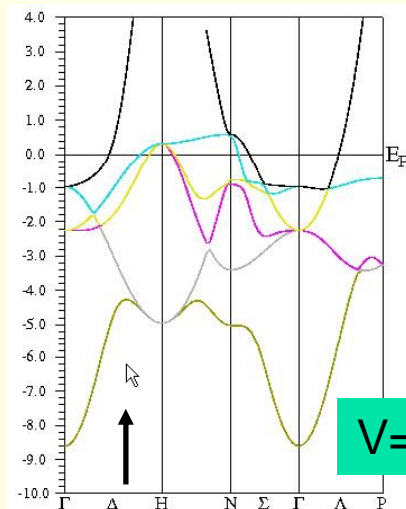
Fe_vol__3.0_rk8_5000k.scf::MMTOT: 2.258

Fe_vol__-3.0_rk8_5000k.scf::MMTOT: 2.159

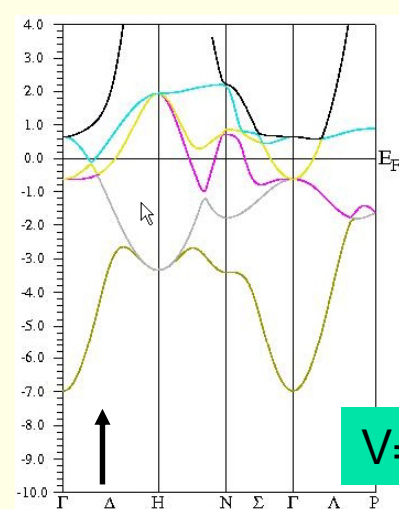
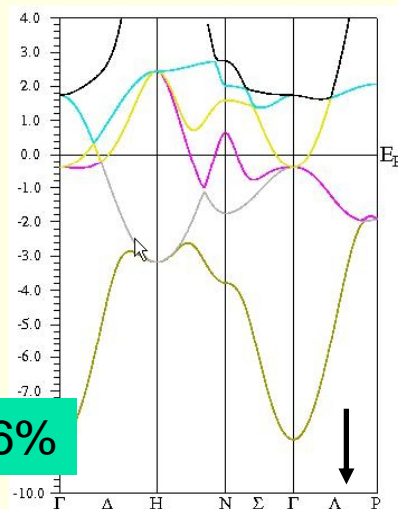
Fe_vol__6.0_rk8_5000k.scf::MMTOT: 2.320

Fe_vol__-6.0_rk8_5000k.scf::MMTOT: 2.114

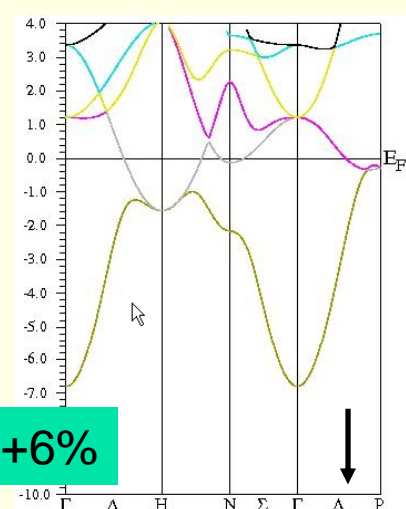
- *compare bandstructure and DOS for large/small volumes (restore_lapw; x lapw0 "recreates" potentials)*

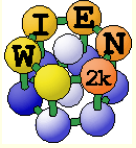


V=-6%



V=+6%

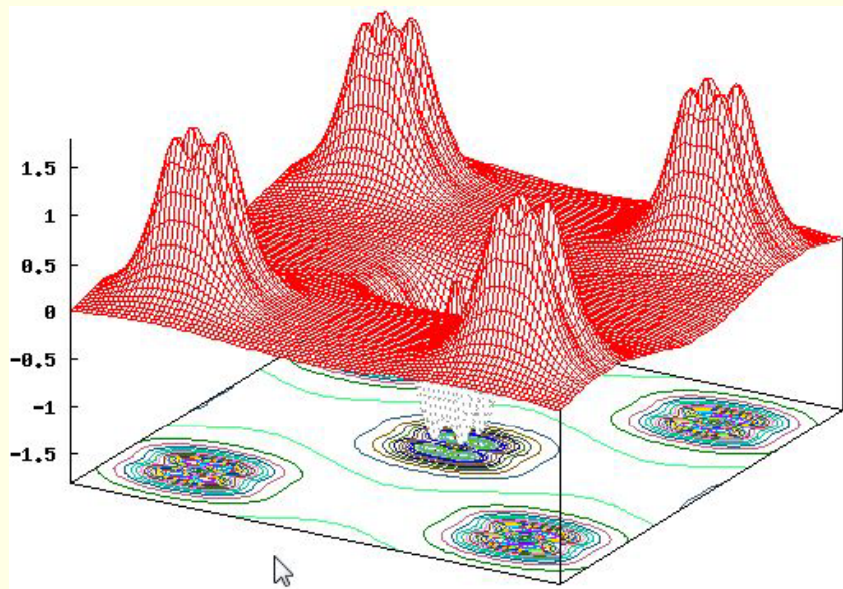


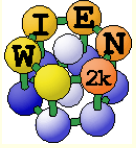


Exercise 5:



- **Antiferromagnetism: bcc Cr ($a_0=2.885 \text{ \AA}$)** (use 5000k, -cc 0.001)
 - *try a ferromagnetic solution (bcc cell with 1 Cr)*
 - *antiferromagnetic calculation (P cell with Cr1 and Cr2 (at 0.5,0.5,0.5))*
 - for afminput your symmetry operation is "identity+(0.5,0.5,0.5)"
 - *is FM or AFM Cr more stable? (:ENE :-2101.769475 vs. -4203.543208 Ry)*
 - *is FM stable at all ? check moments (MMI001: 0.000 vs. $1.116\mu_B$; what "means" 0.0 ???)*
 - *plot spin-densities in the (110) planes (observe t_{2g} -asymmetry)*



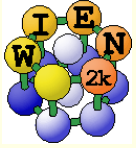


Exercise 6:



■ LDA+U: NiO: NaCl structure, A-type AFM along [111]

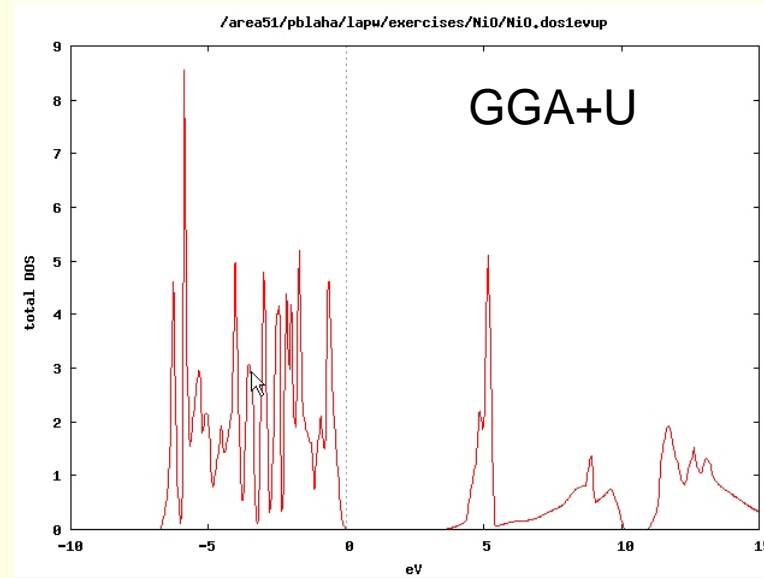
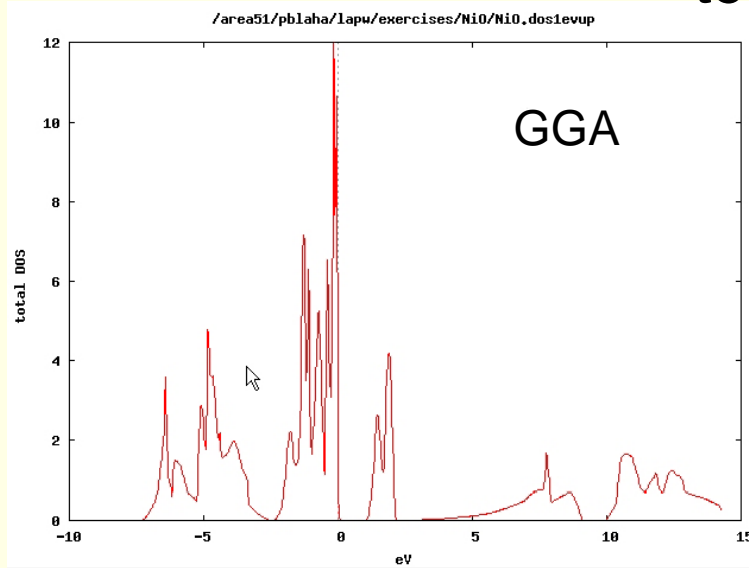
- *R-cell: 5.605236 5.605236 27.459934 bohr*
- *3 non-equivalent atoms: Ni1 (0,0,0), Ni2 (0.5,0.5,0.5), O \pm (.25,.25,.25) ("add 2nd position" **after** first "save_structure"). View and understand the structure (Xcrysden)*
- *case.inst: flip spin for Ni2, make O "non-magnetic"; use 100k-points*
- *GGA calculations (save_lapw NiO_gga)*
- *GGA+U calculations (save_lapw NiO_gga+u)*
 - (use U=7eV, J=0; search the UG to understand case.inorb/indm)
- *GGA+SO calculations (without relativistic LO, Emax=5.0, new k-mesh!)*
 - after scf: x lapwdm -up -so (for :orb001 in NiO.scfdmup)
- *GGA+U+SO calculations (cp NiO.indm NiO.indmc)*
- *compare DOS (total, Ni1, Ni2, O) for GGA and GGA+U*
 - observe the change in gaps (exp: 4eV) and shift of Ni/O weights
 - compare spin moments (GGA: 1.41; GGA+U: 1.76; GGA+U+SO:1.76;GGA+SO: 1.17 μ B)
 - compare orbital moments for SO and SO+U calculations (0.16 and 0.17 μ B)



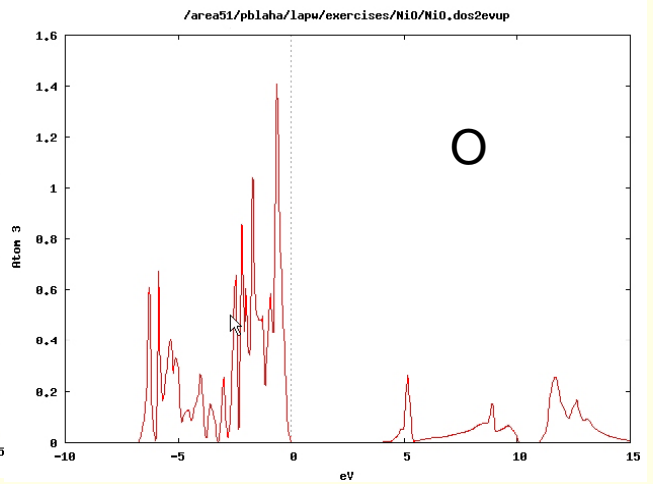
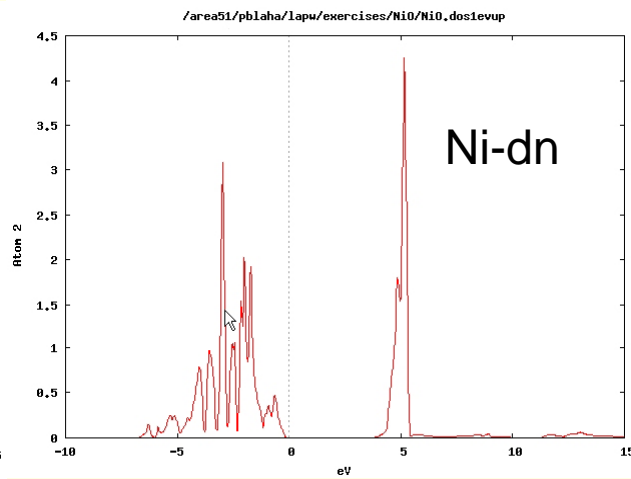
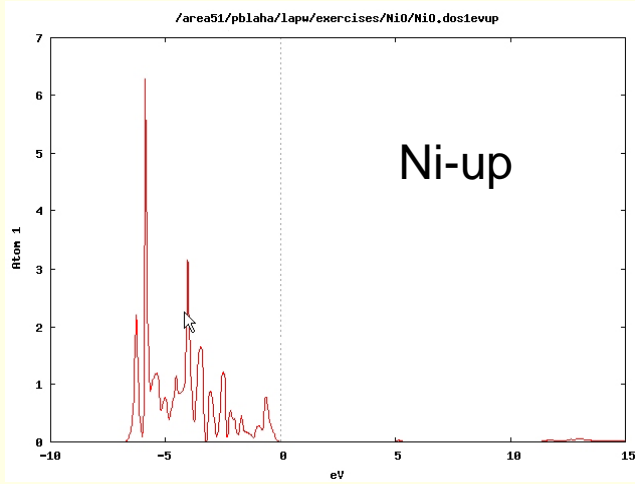
NiO cont...

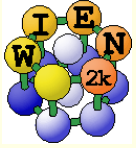


total DOS



GGA+U





Exercise 7:



■ Optical properties: fcc Al

- $a_0 = 4.05 \text{ \AA}$
- *init_lapw* (use 165 IBZ k-points only!)
- *run_lapw*
- *calculate optics* (as described by C. Ambrosch-Draxl, compare with her Fig.)
 - calculate plasma frequency (case.outputjoint) and dielectric function
 - check your results with respect to k-mesh
 - x kgen (check for about 1000 and 4000 IBZ-points)
 - x lapw1
 - x lapw2 -fermi
 - x optic, x joint, x kram

■ Optical properties: fcc Pt

- $a_0 = 3.92 \text{ \AA}$
- *compare optics without / with spin-orbit coupling* (compare with CAD)
 - do scalar-relativistic calc. first including optics
 - *init_so* (put large Emax in case.in1); *run_lapw -so, optics*