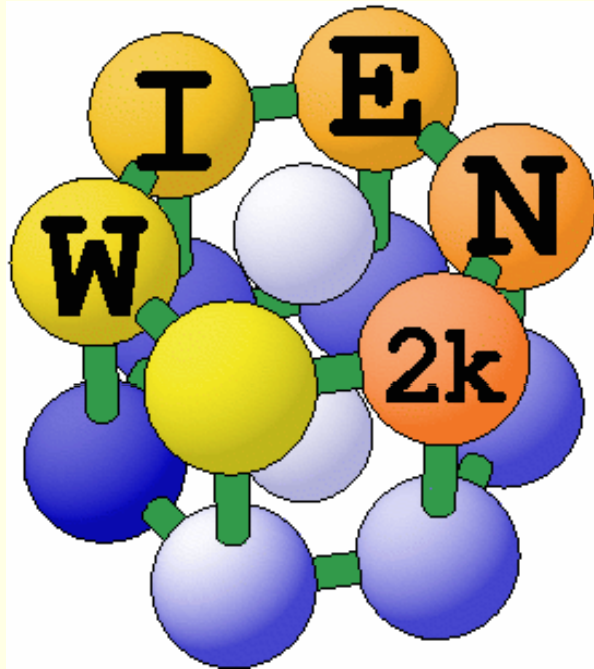




# WIEN2k software package



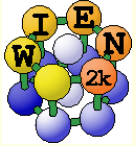
WIEN97: ~500 users  
WIEN2k: ~1250 users

**An Augmented Plane Wave Plus Local  
Orbital  
Program for Calculating Crystal Properties**

**Peter Blaha  
Karlheinz Schwarz  
Georg Madsen  
Dieter Kvasnicka  
Joachim Luitz**

November 2001  
Vienna, AUSTRIA  
Vienna University of Technology

<http://www.wien2k.at>



## General remarks on WIEN2k



- WIEN2k consists of many independent F90 programs, which are linked together via C-shell scripts.
- Each „case“ runs in his own directory `./case`
- The „master input“ is called `case.struct`
- Initialize a calculation: `init_lapw`
- Run scf-cycle: `run_lapw (runsp_lapw)`
- You can run WIEN2k using any www-browser and the w2web interface, but also at the command line in an xterm.
- Input/output/scf files have endings as the corresponding programs:
  - *case.output1...lapw1; case.in2...lapw2; case.scf0...lapw0*
- Inputs are generated using STRUCTGEN(w2web) and `init_lapw`



# w2web: the web-based GUI of WIEN2k



- Based on **www**
  - *WIEN2k can be managed remotely via w2web*
- Important steps:
  - *start w2web on all your hosts*
    - login to the desired host (ssh)
    - w2web (at first startup you will be asked for username/password, port-number, (master-)hostname. creates ~/.w2web directory)
  - *use your browser and connect to the (master) **host:portnumber***
    - firefox <http://fp98.zserv:10000>
  - *create a new session on the desired host (or select an old one)*

Welcome to *w2web*  
the fully web-enabled interface to WIEN2k

**Select stored session:**

- CI2
- Fayalit
- Fccni (<http://fp98.zserv:10000>)
- FeF2
- Forsterit
- H\_atom
- Hg1201
- Hg3AsO4Cl (<http://hal.zserv:10000>)
- HgAsO4Cl (<http://hal.zserv.tuwien.ac.at:10000>)
- I2
- MgCO3
- NdNiSnD (<http://jupiter:10000>)
- NdNiSn\_AF (<http://jupiter:10000>)
- NdNiSn (<http://jupiter:10000>)
- TIC\_evapaph
- TIC\_kla (<http://pauli:10000>)
- TIC
- TIN\_evapaph

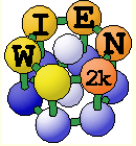
**Create new session:**

Session\_name

on host-node

- master node
- <http://jupiter:10000>
- <http://homer:10000>
- <http://pauli.theochem.tuwien.ac.at:10000>
- <http://fp98.zserv.tuwien.ac.at:10000>
- <http://hal.zserv.tuwien.ac.at:10000>
- <http://venus.theochem.tuwien.ac.at:10000>

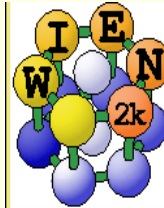
w2web © luitz.at



# w2web GUI (graphical user interface)



- **Structure generator**
  - *spacegroup selection*
  - *import cif file*
- **step by step initialization**
  - *symmetry detection*
  - *automatic input generation*
- **SCF calculations**
  - *Magnetism (spin-polarization)*
  - *Spin-orbit coupling*
  - *Forces (automatic geometry optimization)*
- **Guided Tasks**
  - *Energy band structure*
  - *DOS*
  - *Electron density*
  - *X-ray spectra*
  - *Optics*



#### Execution >>

StructGen™  
initialize calc.  
run SCF  
single prog.  
optimize(V,c/a)  
mini. positions

#### Utils. >>

#### Tasks >>

#### Files >>

struct file(s)  
input files  
output files  
SCF files

#### Session Mgmt. >>

change session  
change dir  
change info

#### Configuration

#### Usersguide

html-Version  
pdf-Version

Idea and realization  
by

Session: TiC

/area51/pblaha/lapw/2005-june/TiC

## StructGen™

You have to click "Save Structure" for changes to take effect!

Save Structure

Title: TiC

Lattice:

Type: F

P  
F  
B  
CXY  
CYZ  
CXZ  
R  
H  
1\_P1

Spacegroups from  
Bilbao Cryst Server

Lattice parameters in Å

a=4.328000038 b=4.328000038 c=4.328000038

$\alpha=90.000000$   $\beta=90.000000$   $\gamma=90.000000$

Inequivalent Atoms: 2

Atom 1: Ti Z=22.0 RMT=2.0000 remove atom

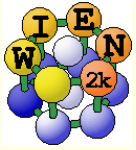
Pos 1: x=0.00000000 y=0.00000000 z=0.00000000 remove

add position

Atom 2: C Z=6.0 RMT=1.9000 remove atom

Pos 1: x=0.50000000 y=0.50000000 z=0.50000000 remove

add position



# Spacegroup $P4_2/mnm$

Structure given by:  
 spacegroup  
 lattice parameter  
 positions of atoms  
 (basis)

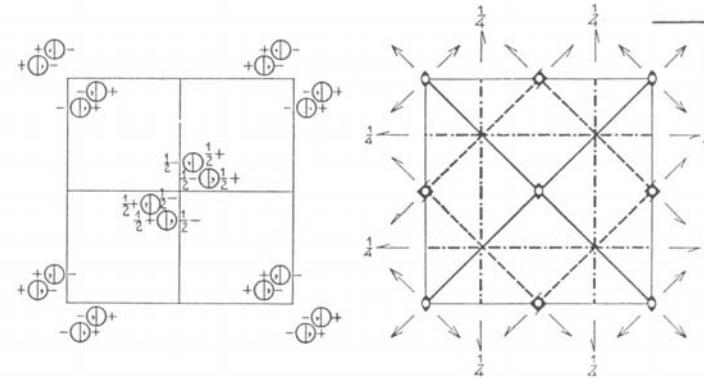
**Rutile  $TiO_2$ :**  
 $P4_2/mnm$  (136)  
 $a=8.68, c=5.59$  bohr  
**Ti: (0,0,0)**  
**O: (0.304,0.304,0)**

$P4_2/mnm$   
 $D_{4h}^{14}$

No. 136

$P4_2/m 2_1/n 2/m$

$4/m m m$  Tetragonal



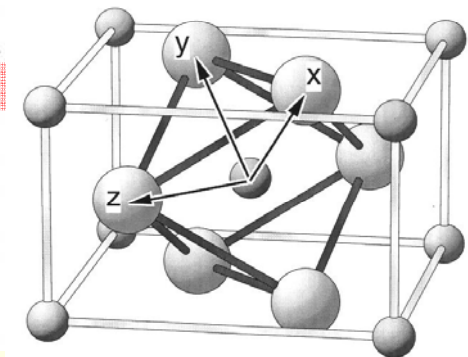
Origin at centre ( $mmm$ )

Number of positions,  
Wyckoff notation,  
and point symmetry

Co-ordinates of equivalent positions

Conditions limiting  
possible reflections

Number of positions, Wyckoff notation, and point symmetry			Co-ordinates of equivalent positions	Conditions limiting possible reflections
16	$k$	1	$x, y, z; \bar{x}, \bar{y}, z; \frac{1}{2} + x, \frac{1}{2} - y, \frac{1}{2} + z; \frac{1}{2} - x, \frac{1}{2} + y, \frac{1}{2} + z;$ $x, y, \bar{z}; \bar{x}, \bar{y}, \bar{z}; \frac{1}{2} + x, \frac{1}{2} - y, \frac{1}{2} - z; \frac{1}{2} - x, \frac{1}{2} + y, \frac{1}{2} - z;$ $y, x, z; \bar{y}, \bar{x}, z; \frac{1}{2} + y, \frac{1}{2} - x, \frac{1}{2} + z; \frac{1}{2} - y, \frac{1}{2} + x, \frac{1}{2} + z;$ $y, x, \bar{z}; \bar{y}, \bar{x}, \bar{z}; \frac{1}{2} + y, \frac{1}{2} - x, \frac{1}{2} - z; \frac{1}{2} - y, \frac{1}{2} + x, \frac{1}{2} - z.$	General: $hkl$ : No conditions $hk0$ : No conditions $0kl$ : $k+l=2n$ $hhl$ : No conditions
8	$j$	$m$	$x, x, z; \bar{x}, \bar{x}, z; \frac{1}{2} + x, \frac{1}{2} - x, \frac{1}{2} + z; \frac{1}{2} - x, \frac{1}{2} + x, \frac{1}{2} + z;$ $x, x, \bar{z}; \bar{x}, \bar{x}, \bar{z}; \frac{1}{2} + x, \frac{1}{2} - x, \frac{1}{2} - z; \frac{1}{2} - x, \frac{1}{2} + x, \frac{1}{2} - z.$	Special: as above, plus } no extra conditions
8	$i$	$m$	$x, y, 0; \bar{x}, \bar{y}, 0; \frac{1}{2} + x, \frac{1}{2} - y, \frac{1}{2}; \frac{1}{2} - x, \frac{1}{2} + y, \frac{1}{2};$ $y, x, 0; \bar{y}, \bar{x}, 0; \frac{1}{2} + y, \frac{1}{2} - x, \frac{1}{2}; \frac{1}{2} - y, \frac{1}{2} + x, \frac{1}{2}.$	
8	$h$	2	$0, \frac{1}{2}, z; 0, \frac{1}{2}, \bar{z}; 0, \frac{1}{2}, \frac{1}{2} + z; 0, \frac{1}{2}, \frac{1}{2} - z;$ $\frac{1}{2}, 0, z; \frac{1}{2}, 0, \bar{z}; \frac{1}{2}, 0, \frac{1}{2} + z; \frac{1}{2}, 0, \frac{1}{2} - z.$	$hkl$ : $h+k=2n; l=2n$
4	$g$	$mm$	$x, \bar{x}, 0; \bar{x}, x, 0; \frac{1}{2} + x, \frac{1}{2} + x, \frac{1}{2}; \frac{1}{2} - x, \frac{1}{2} - x, \frac{1}{2}.$	
4	$f$	$mm$	$x, x, 0; \bar{x}, \bar{x}, 0; \frac{1}{2} + x, \frac{1}{2} - x, \frac{1}{2}; \frac{1}{2} - x, \frac{1}{2} + x, \frac{1}{2}.$	
4	$e$	$mm$	$0, 0, z; 0, 0, \bar{z}; \frac{1}{2}, \frac{1}{2}, \frac{1}{2} + z; \frac{1}{2}, \frac{1}{2}, \frac{1}{2} - z.$	
4	$d$	$\bar{4}$	$0, \frac{1}{2}, \frac{1}{4}; \frac{1}{2}, 0, \frac{1}{4}; 0, \frac{1}{2}, \frac{3}{4}; \frac{1}{2}, 0, \frac{3}{4}.$	
4	$c$	$2/m$	$0, \frac{1}{2}, 0; \frac{1}{2}, 0, 0; 0, \frac{1}{2}, \frac{1}{2}; \frac{1}{2}, 0, \frac{1}{2}.$	
2	$b$	$mmm$	$0, 0, \frac{1}{2}; \frac{1}{2}, \frac{1}{2}, 0.$	
2	$a$	$mmm$	$0, 0, 0; \frac{1}{2}, \frac{1}{2}, \frac{1}{2}.$	



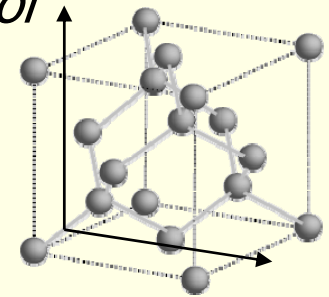


# Structure generator



## ■ Specify:

- Number of *nonequivalent atoms*
- *lattice type* (*P*, *F*, *B*, *H*, *CXY*, *CXZ*, *CYZ*) or *spacegroup symbol*
  - if existing, you must use a **SG-setting** with inversion symmetry:
    - Si:  $\pm(1/8,1/8,1/8)$ , not  $(0,0,0)+(1/4,1/4,1/4)$ !
- *lattice parameters* *a*, *b*, *c* (in Å or bohr)
- *name of atoms* (*Si*) and *fractional coordinates* (*position*)
  - as numbers (0.123); fractions (1/3); simple expressions ( $x-1/2, \dots$ )
  - in fcc (bcc) specify just one atom, not the others in (1/2,1/2,0; ...)



## ■ „save structure “

- updates automatically *Z*, *r0*, *equivalent positions* and generates *case.inst*

## ■ „set RMT and continue“: (specify proper “reduction” of NN-distances)

- *non-overlapping* „as large as possible“ (saves time), but not larger than 3 bohr
- RMT for *sp* (*d*) - elements 10-20 % smaller than for *d* (*f*) elements
- *largest spheres* not more than 50 % larger than *smallest sphere*
- Exception: *H* in C-H or O-H bonds: RMT~0.6 bohr (RKMAX~3-4)
- Do not change RMT in a „series“ of calculations, RMT equal for *same atoms*

## ■ „save structure – save+cleanup“





# scf-cycle



## ■ run\_lapw [options] (for nonmagnetic cases)

■ <i>-ec 0.0001</i>	<i>convergence of total energy (Ry)</i>
■ <i>-cc 0.0001</i>	<i>convergence of charge distance (e)</i>
■ <i>-fc 1.0</i>	<i>convergence of forces (mRy/bohr)</i>
■ <i>-it /-it0</i>	<i>iterative diagonalization (large speedup)</i>
■ <i>-p</i>	<i>parallel calculation (needs .machines file)</i>
■ <i>-so</i>	<i>add spin-orbit (only after „init_so“)</i>
■ <i>Spacegroups without inversion use automatically lapw1c, lapw2c (case.in1c,in2c)</i>	

## ■ case.scf: master output file, contains history of the scf-cycle

- *most information is stored with some „labels“ (grep :label case.scf)*

■ :ENE	:DIS	:FER	:CTO001	:NTO001	:QTL001
■ :FOR002:	2.ATOM	19.470	0.000	0.000	19.470
■ :FGL002:	2.ATOM		13.767	13.767	0.000

- :LAT :VOL :POSxxx





# BZ integration, "FERMI"-methods



- Replace the "integral" of the BZ by a finite summation on a mesh of "k-points"

$$\rho(r) = \sum_n^{E_n < E_F} \int \psi_{k,n}^* \psi_{k,n} d^3k = \sum_{k,n} w_{k,n} \psi_{k,n}^* \psi_{k,n}$$

- weights  $w_{k,n}$  depend on k and bandindex n (occupation)

- for full "bands" the weight is given by "symmetry"

- $w(\Gamma)=1, w(x)=2, w(\Delta)=4, w(k)=8$

➔ shifted "Monkhorst-Pack" mesh

- for partially filled bands (metals) one must find the Fermi-energy (integration up to  $E_F$ ) and determine the weights for each state  $E_{k,n}$

- linear tetrahedron method (TETRA, eval=999)

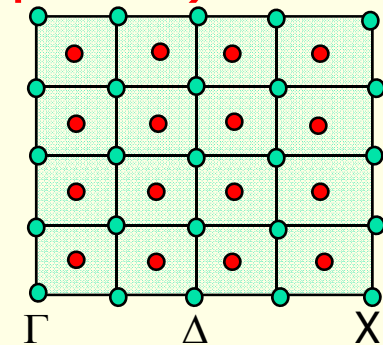
- linear tetrahedron method + "Bloechl" corrections (TETRA)

- "broadening methods"

- gauss-broadening (GAUSS 0.005)

- temperature broadening (TEMP 0.005)

- broadening useful to damp scf oscillations, but dangerous (magnetic moment)





# k-mesh generation



- **x kgen** (generates k-mesh and reduces to irreducible wedge using symmetry)
  - *always "add inversion" except in magnetic spin-orbit calculations*
    - time inversion holds and  $E(k) = E(-k)$
  - *always "shift" the mesh for scf-cycle*
    - gaps often at  $\Gamma$  ! (might not be in your mesh)
  - *small unit cells and metals require large k-mesh (1000-100000)*
  - *large unit cells and insulators need only 1-10 k-points*
  - *use at first a fairly coarse mesh for scf*
  - *continue later with finer mesh*
    - mesh was good if nothing changes and scf terminates after few (3) iterations
  - *use an even finer meshes for DOS, spectra, optics,...*



## Program execution:



- All programs are executed via the „master“ shell-script „x“:

```
x lapw2 -up -c
```

- This generates a „def“ file: `lapw2.def`

```
5, 'tin.in2c',      'old',      'formatted'  
6, 'tin.output2up', 'unknown', 'formatted'  
8, 'tin.clmvalup',  'unknown', 'formatted'  
10, './tin.vectorup', 'unknown', 'unformatted'
```

- and executes: `lapw2c lapw2.def`

- All WIEN2k-shell scripts have long and short names:

- *x\_lapw; runsp\_lapw, runfsm\_lapw → x; runsp; runfsm*

- All scripts have a „help“ switch „-h“, which explains flags and options (without actually execution)

```
x -h
```

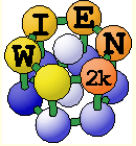
```
x lapw1 -h
```



# Getting help



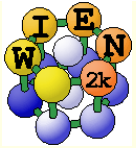
- **\*\_lapw -h** „help switch“ of all WIEN2k-scripts
- **help\_lapw:**
  - *opens usersguide.pdf; Use ^f keyword to search for an item („index“)*
- **html-version of the UG:** ( $\$WIENROOT/SRC\_usersguide/usersguide.html$ )
- **[http://www.wien2k.at/reg\\_user](http://www.wien2k.at/reg_user)**
  - *FAQ page with answers to common questions*
  - *Update information: When you think the program has an error, please check newest version*
  - *Textbook section: DFT and the family of LAPW methods by S.Cottenier*
  - *Mailing-list:*
    - **subscribe** to the list (always use the same email)
    - **full text search** of the „digest“ (your questions may have been answered before)
    - **posting questions: Provide sufficient information**, locate your problem (case.dayfile, \*.error, case.scf, case.outputX).
    - **„My calculation crashed. Please help.“** This will most likely not be answered.



# most common problems



- **scf-cycle diverges (grep :DIS case.scf):**
  - *check structure;*
  - *reduce mixing in case.inm; rm \*.broyd\* case.scf; x dstart*
- **„QTL-B“ value too large - STOP**
  - *identify for which eigenvalue, atom and  $\ell$  it happens, check EF (case.scf2, case.output2)*
  - *identify the corresponding linearization energies in case.scf1*
  - *change the corresponding linearization energy in case.in1*
    - compare and check with :EPL and :EPH lines in case.scf2
    - default E-parameters may need changes for
      - surfaces (EF often negative) or heavy elements (EF often larger than 1.0)
    - use „-in1new 1“ switch
  - *if QTL-B occurs for an atom with large RMT, reduce RMT*
    - this may happen for larger RKMAX („numerical linear dependency“)



# case.in1



## WFFIL

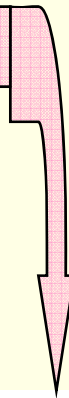
- 7.00 10 4
- 0.30 5 0
- 0 0.30
- 0 -3.72
- 1 -2.07
- 1 0.30
- 2 0.30

## (WFPRI, SUPWF)

(R-MT\*K-MAX; MAX L IN WF, V-NMT

global E-param with N other, napw

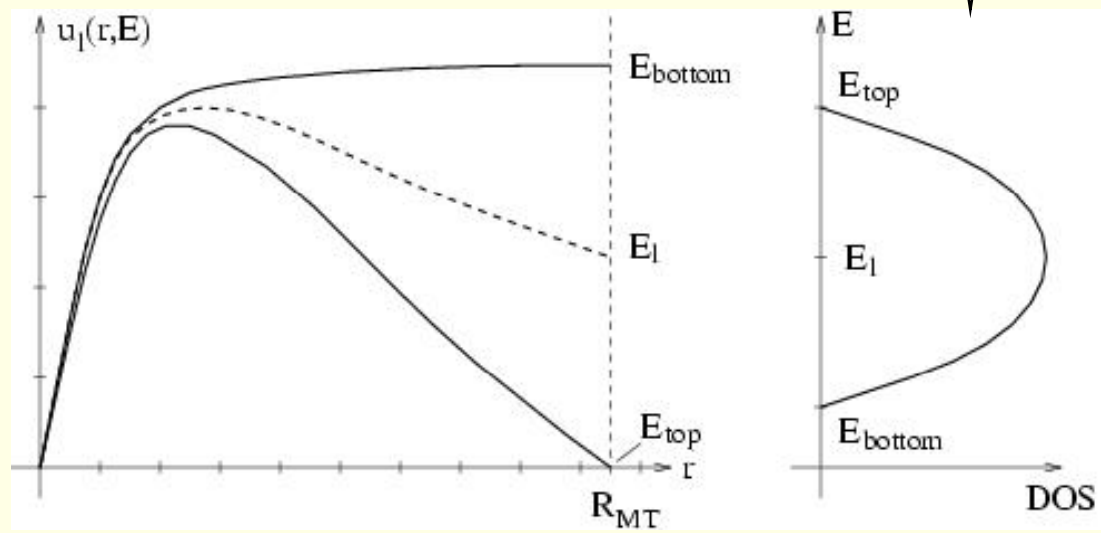
- 0.000 CONT 1 Es
- 0.005 STOP 1 Es-LO with search
- 0.010 CONT 1 Ep with search
- 0.000 CONT 1 Ep-LO
- 0.010 CONT 1 0/1...LAPW/APW+lo



$$\Psi = \sum_{K_n}^{KMAX} c_{K_n} e^{iK_n r}$$

$$\Phi_{K_n} = \sum_l^{lmax} A_{lm} u_l(E_l, r) Y_{lm}$$

$$H_{n,m}^{NS} = \langle \Phi_l | V_{LM}^{NS} | \Phi_l \rangle$$







# run\_lapw -ql 0.05 -in1new 1



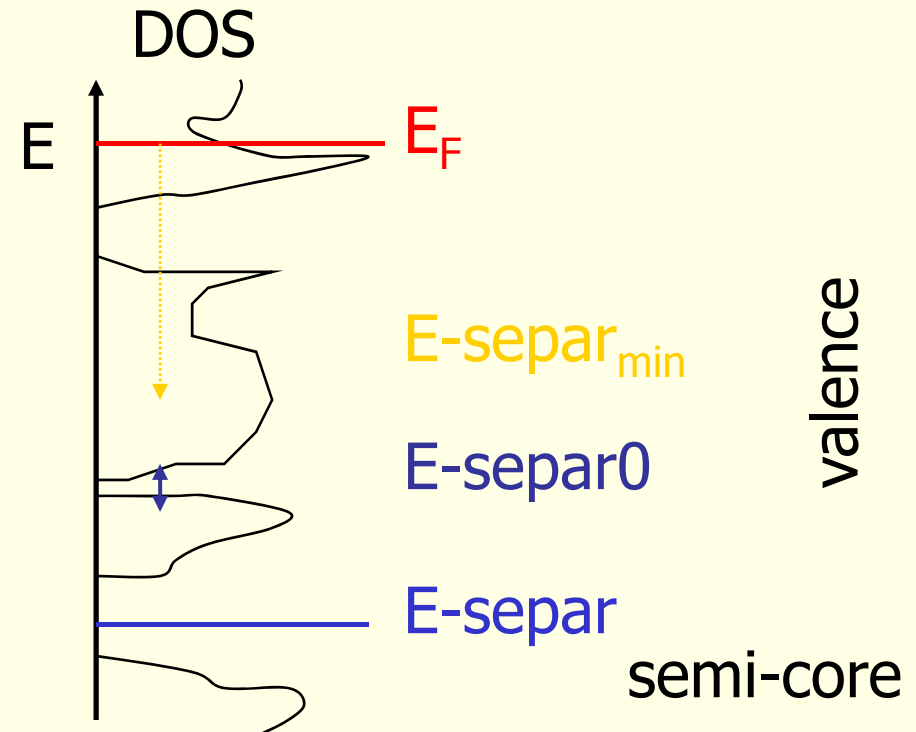
- Alternative case.in1 file produced by write\_in1:  
case.scf2:

- Energy to separate semicore and valence states: 0.34941
- :FER : F E R M I - ENERGY(TETRAH.M.)= 0.79528

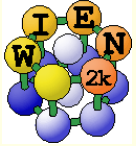
Q-s-low	E-s-low	Q-p-low	E-p-low	Q-d-low	E-d-low
:EPL01:1.9813	-2.6852	5.5892	-1.1099	0.0873	0.0910
Q-s-hi	E-s-hi	Q-p-hi	E-p-hi	Q-d-hi	E-d-hi
:EPH01:0.0668	0.5152	0.1752	0.6174	1.0614	0.6181

## → case.in1:

- | WFFIL    | (WFPRI, SUPWF) |
|----------|----------------|
| 7.00     | 10 4           |
| .49528 6 | 0              |
| 0 0.515  | 0.000 CONT 1   |
| 0 -2.685 | 0.000 CONT 1   |
| 1 0.617  | 0.000 CONT 1   |
| 1 -1.110 | 0.000 CONT 1   |
| 2 0.618  | 0.000 CONT 1   |
| 2 0.091  | 0.000 CONT 1   |
| ...      |                |







# Properties with WIEN2k - I



## ■ Energy bands

- *classification of irreducible representations*
- *'character-plot' (emphasize a certain band-character)*

## ■ Density of states

- *including partial DOS with l and m- character (eg.  $p_x$ ,  $p_y$ ,  $p_z$ )*

## ■ Electron density, potential

- *total-, valence-, difference-, spin-densities,  $\rho$  of selected states*
- *1-D, 2D- and 3D-plots (Xcrysden)*
- *X-ray structure factors*
- *Bader's atom-in-molecule analysis, critical-points, atomic basins and charges*  
(  $\nabla \rho \cdot \vec{n} = 0$  )
- *spin+orbital magnetic moments (spin-orbit / LDA+U)*

## ■ Hyperfine parameters

- *hyperfine fields (contact + dipolar + orbital contribution)*
- *Isomer shift*
- *Electric field gradients*



# DOS



- be sure to have eigenvectors on a dense tetrahedral mesh after a scf calculation

- *eventually:*

- x kgen
- edit case.in1 (larger Emax)
- x lapw1

- case.outputt

- *integrated DOS*

- case.dos1ev (3ev)

- *text-file for plotting*
- *E-zero at  $E_F$*

Session: TiC  
/susi/pblaha/lapw/TiC

## Density of states

x lapw2 -qtl Calculate partial charges  interactively

edit TiC.int Edit input-file for TETRA

x tetra Calculate partial DOS  interactively

edit TiC.outputt Check output of TETRA

dosplot Plot DOS

Session: TiC  
/susi/pblaha/lapw/TiC

## File:

/susi/pblaha/lapw/TiC/TiC.int

continue with DOS

Save

Download this file:

Header from TiC.qtl:

```
ATOM 1 tot,0,1,2,3,xdos(i,j),j=1,i,i=1,1xdos2)
ATOM 2 tot,0,1,2,D-eg,D-t2g,3
```

Title

```
-0.50 0.002 1.500 0.003 EMIN, DE, EMAX, Gauss-broadening(>;de)
3 NUMBER OF DOS-CASES specified below
0 1 total atom, case=column in qtl-header, label
1 2 Atom1-s
2 5 Atom2-eg
```



# Properties with WIEN2k - II



## ■ Total energy and forces

- *optimization of internal coordinates, (MD, BROYDEN)*
- *cell parameter only via  $E_{tot}$  (no stress tensor)*
- *elastic constants for cubic cells*
- *Phonons via supercells*
  - interface to PHONON (K.Parlinski) – bands, DOS, thermodynamics, neutrons
  - interface by G.Madsen to PHON (D.Alfe)
    - <http://www.chem.au.dk/~webuorg/new/groups/gm/gm.html>

## ■ Spectroscopy

- *core level shifts*
- *X-ray emission, absorption, electron-energy-loss (with core holes)*
  - *core-valence/conduction bands including matrix elements and angular dep.*
- *optical properties (dielectric function in RPA approximation, JDOS including momentum matrix elements and Kramers-Kronig)*
- *fermi surface: 2D, 3D (using XcrysDen)*



# X-ray emission and absorption spectra

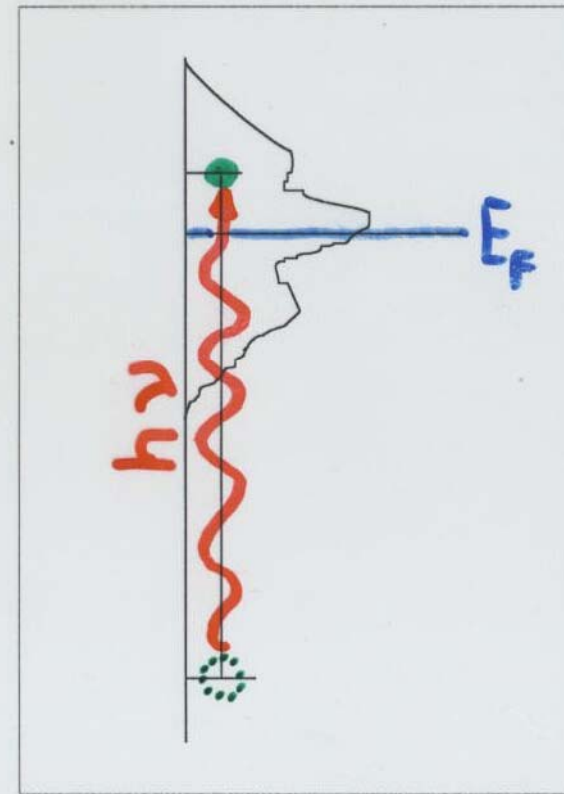
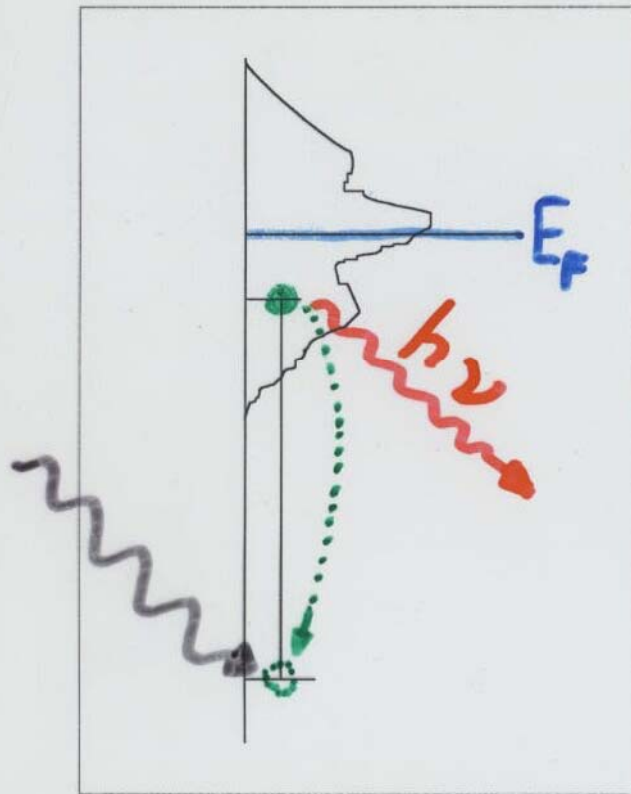


## XES

## XAS

◆ Emission

◆ Absorption



conduction bands

valence bands

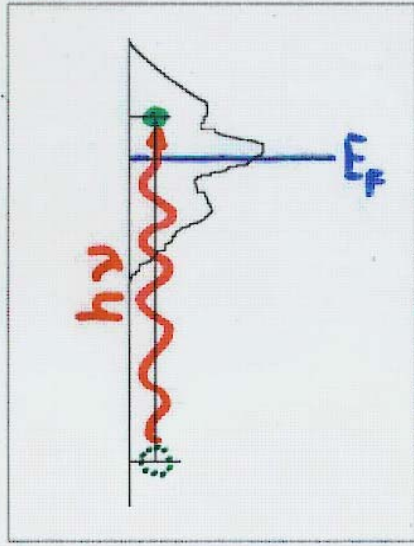
core state



# XAS: X-ray absorption spectra



## ◆ Absorption



$$\Delta l = \pm 1$$

dipole-selection rule

Final state effects:  
 supercell with (partial) core hole  
 on one of the atoms

dipole operator

$$I \propto \nu^3 |\langle \Psi_{val} | r | \Psi_{core} \rangle|^2 \chi_l^A(\epsilon)$$

l-like PDOS (atom A)

$$\frac{I_{n'l'}}{\nu^3} = \sum_l W_{ll'} M_A^2(l, n'l', \epsilon) \chi_l^A(\epsilon) \delta(\epsilon - E_{core}, h\nu)$$

energy conservation

Radial transition probability:

- from a core state with q.n.  $n'l'$  on atom A
- to an l-like conduction band state inside the atomic sphere A
- within the dipole approximation



## “Final state rule”:



“Final state” determines the spectrum:

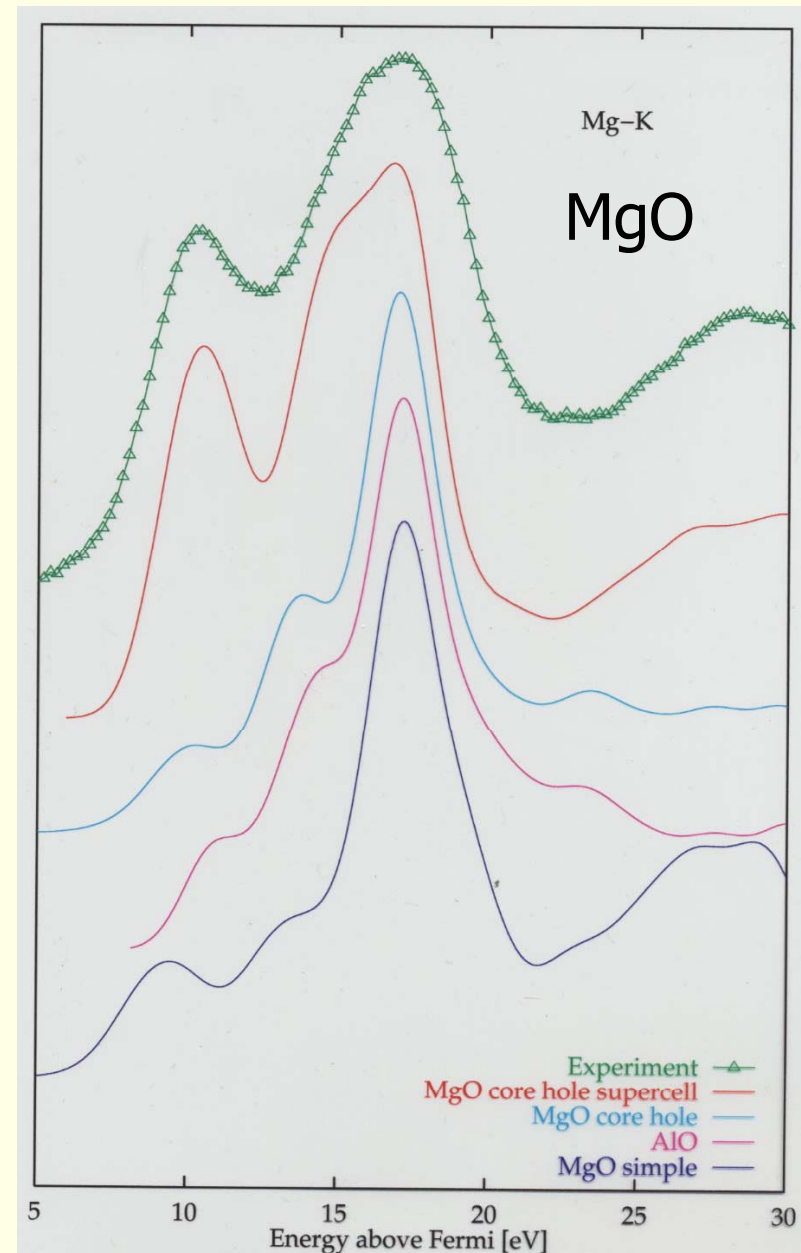
### Emission spectroscopy:

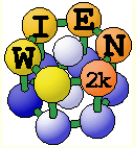
Final state has filled core, but valence hole. This is usually well screened, thus one “sees” the **groundstate**.

### Absorption spectroscopy:

“hole” in core state, but additional  $e^-$  in conduction band. **Core-hole** has large effect on the spectrum (static approximation)

→ **2x2x2 supercell calculation**, with core hole in **one** of the Mg atoms. This allows the conduction state to relax (adjust to the larger **effective** nuclear charge), but also to have screening from the environment.





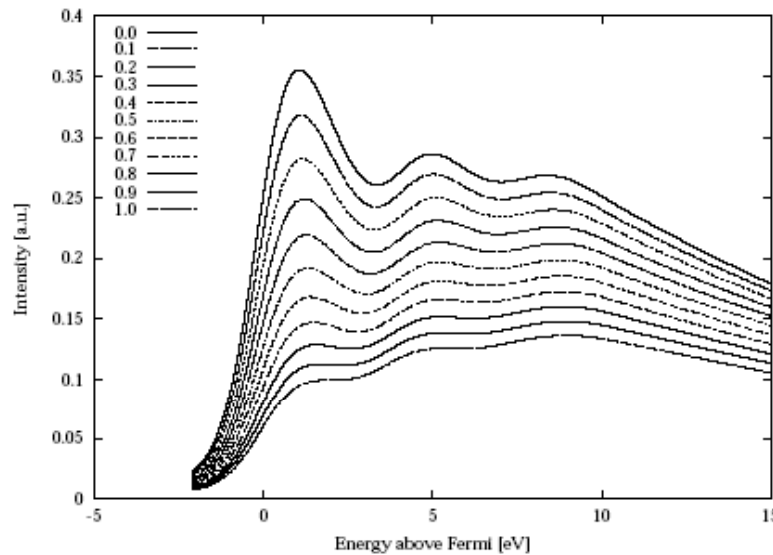
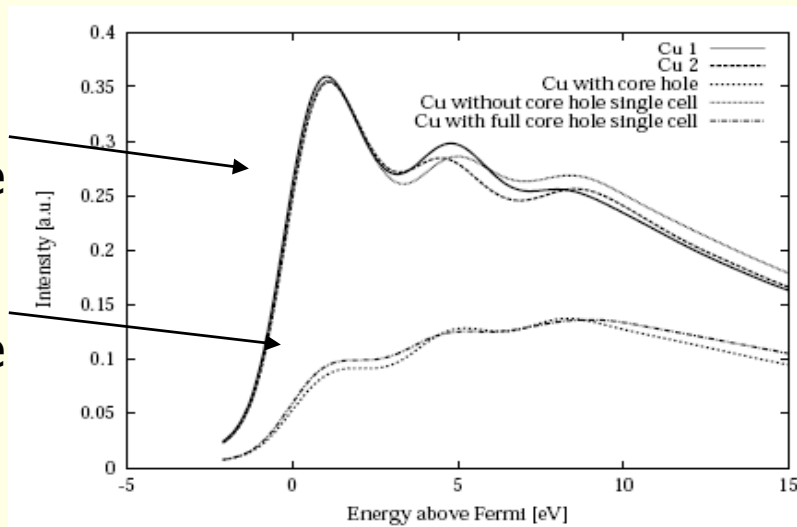
# Partial core hole screening in the Cu L3 edge



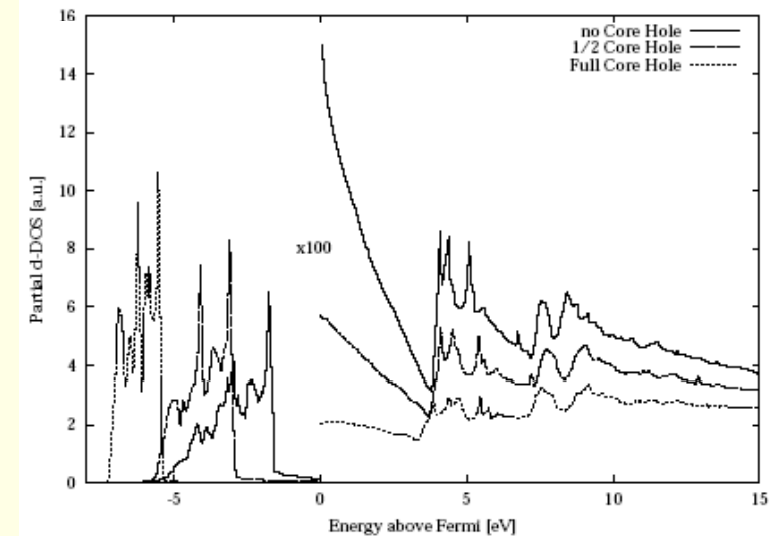
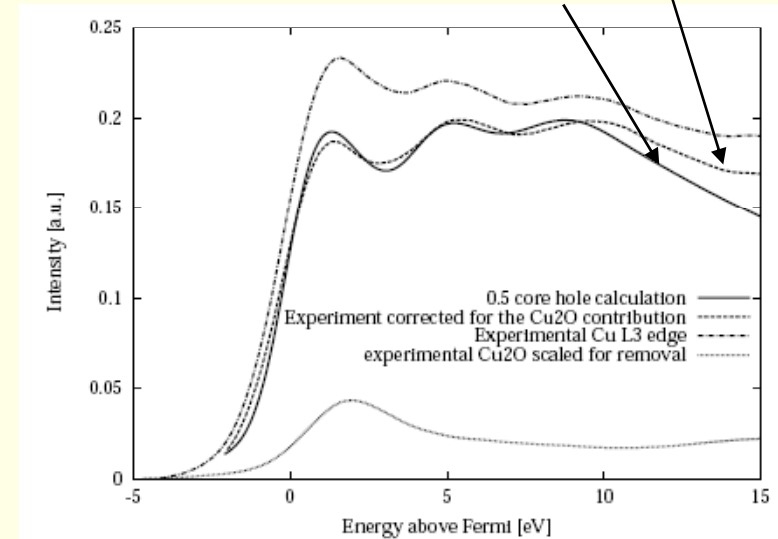
- J. Luitz et al., Eur. Phys. J. B 21, 363{367 (2001)

without  
corehole

with  
corehole



experiment  
0.5e corehole





# Properties with WIEN2k - III



## ■ New developments (in progress)

- *non-collinear magnetism (available on request: [www.wien2k.at](http://www.wien2k.at))*
- *transport properties (Fermi velocities, Seebeck, conductivity, thermoelectrics, ..) (G.Madsen's BotzTrap code)*  
[www.chem.au.dk/~webuorg/new/groups/gm/gm.html](http://www.chem.au.dk/~webuorg/new/groups/gm/gm.html)
- *non-linear optics (available on request)*
- *Bethe-Salpeter equation (Excitons)*
- *Compton profiles*
- *linear response (phonons, E-field) (C.Ambrosch-Draxl)*
- *stress tensor (C.Ambrosch-Draxl)*
- *approximate HartreeFock (+Hybrid functionals)*
- *exact exchange*
- *GW (M.Scheffler, FH Berlin)*
- *grid-computing*



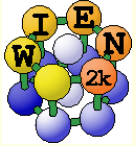


# Cohesive energy



$$E_{A_x B_y}^{cohes.} = E^{crystal} - x E_A^{atom} - y E_B^{atom}$$

- $E^{crystal}$ : scalar-relativistic valence (or approx. SO)
- $E^{atom}$ : LSTART: fully-relativistic → inconsistent description
  - for heavier elements (2<sup>nd</sup> row):  
supercell with one atom in a ~30 bohr FCC box  
(identical RMT, RKmax, 1 k-point, spinpolarized)



# Structural optimizations:



- **Lattice parameters, volume, c/a ratio only via total energies:**
  - *x optimize: creates a series of "struct" files + script "optimize.job"*
    - select volume or c/a, ...
    - select number of cases and desired changes in volume (in % of  $V_0$ )
  - *edit optimize.job*
    - adapt to your need: change / uncomment various lines, eg.:
      - select different convergence parameters, parallelization, more iterations (-i 40)
      - different "save\_lapw" (into a directory with specific names)
      - replace "run\_lapw" by "runsp\_lapw" or `min_lapw -I -j "run_lapw -I -fc 1"`
  - *execute optimize.job*
  - *plot (analyse) the results*
  
- *combinations of volume and c/a are possible:*
  - "x optimize" always uses **case\_initial.struct** (if present)
  - do a "volume" optimization to create case\_vol\_xx.struct files
  - copy the respective case\_vol\_xx.struct file to case\_initial.struct
  - x optimize with "c/a" for this particular volume and proceed as above.



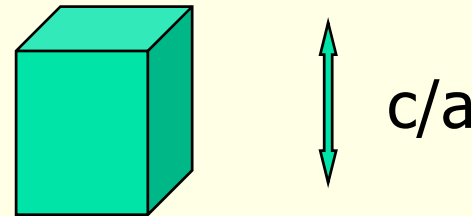
# Symmetry:



## ■ WIEN „preserves“ symmetry:

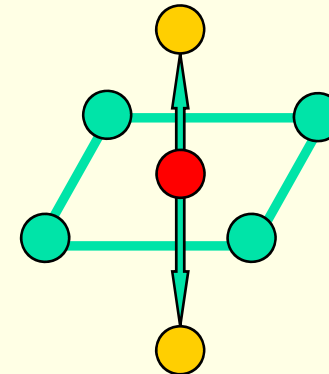
### ■ *c/a optimization of „cubic“ TiC:*

- change c lattice parameter in TiC.struct (tetragonal distortion, #sym.op=0)
- init\_lapw
- change c back to cubic
- x optimize ...



### ■ „Jahn-Teller“ distortion:

- when you start with a perfect octahedra, you will never get any distortion
- → start with slightly distorted positions





# Total energies and atomic forces

(Yu et al.; Kohler et al.)



## Total Energy:

- *Electrostatic energy*
- *Kinetic energy*
- *XC-energy*

$$U[\rho] = \frac{1}{2} \int d^3\vec{r} \rho(\vec{r}) V_{es}(\vec{r}) + \frac{1}{2} \sum_{\alpha} Z_{\alpha} V_{es}^{\alpha}(\vec{r})$$

$$T[\rho] = \sum_i n_i \varepsilon_i - \int d^3\vec{r} \rho(\vec{r}) V_{eff}(\vec{r})$$

$$E_{xc}[\rho] = \int d^3\vec{r} \rho(\vec{r}) \varepsilon_{xc}(\vec{r})$$

## Force on atom $\alpha$ :

$$\vec{F}^{\alpha} = \frac{-dE_{tot}}{d\vec{R}_{\alpha}} = F_{HF}^{\alpha} + F_{core}^{\alpha} + F_{val}^{\alpha}$$

- *Hellmann-Feynman-force*
- *Pulay corrections*

$$F_{HF}^{\alpha} = Z_{\alpha} \sum_{m=-1}^1 \lim_{r_{\alpha} \rightarrow 0} \frac{V_{1m}^{es}(r_{\alpha})}{r_{\alpha}} \nabla_{\alpha} [r_{\alpha} Y_{1m}(\hat{r})]$$

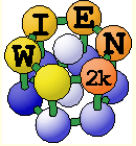
- *Core*
- *Valence*

$$F_{core}^{\alpha} = - \int \rho_{core}(r) \nabla_{\alpha} V_{eff}(r) d\vec{r}$$

- *expensive, contains a summation of matrix elements over all occupied states*

$$F_{val}^{\alpha} = \int_{\alpha} V_{eff}(r) \nabla_{\alpha} \rho_{val}(r) d\vec{r} + \sum_{k,i} n_i \sum_{K,K'} c_i^*(K') c_i(K) \times$$

$$\left[ (K^2 - \varepsilon_i) \oint \phi_{K'}^*(r) \phi_K(r) dS_{\alpha} - i(K - K') \langle \phi_{K'} | H - \varepsilon_i | \phi_K \rangle_{\alpha} \right]$$



## ■ Forces only for "free" structural parameters:

- *NaCl: (0,0,0), (0.5,0.5,0.5) : all positions fixed by symmetry*
- *TiO<sub>2</sub>: Ti (0,0,0), O (u,u,0): one free parameter (u,x,y,z)*

## ■ Forces are only calculated when using "-fc":

- *run\_lapw -fc 1.0 (mRy/bohr)*

- `grep :for002 case.scf`

- 200.

- -130.

- 140.

- 135                    only  $F_{\text{HF}} + F_{\text{core}}$

- 120

- 122                    forces converging

- 121

→ changes "TOT" to "FOR" in case.in2

- -12.3                  $F_{\text{HF}} + F_{\text{core}} + F_{\text{val}}$ , only this last number is correct

## ■ Forces are useful for

- *structural optimization (of internal parameters)*
- *phonons*



# Structural optimization of internal parameters

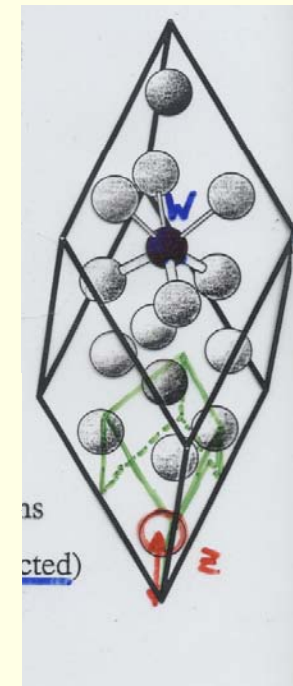
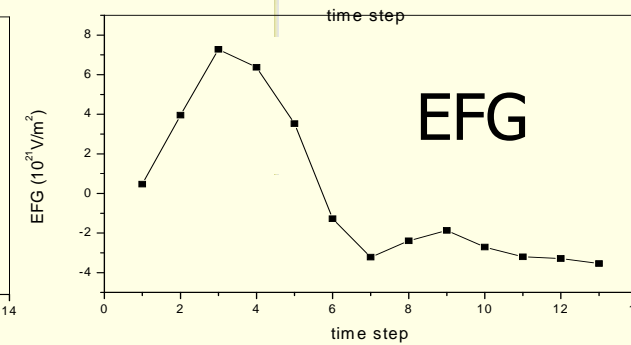
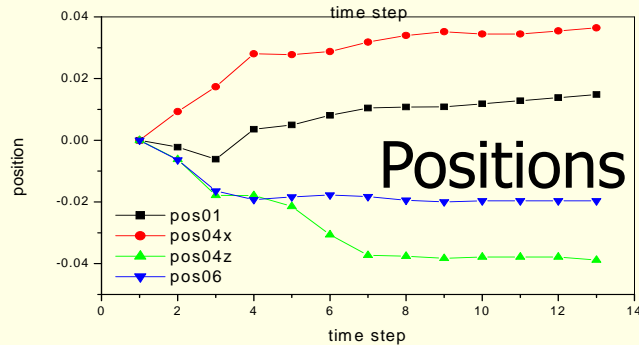
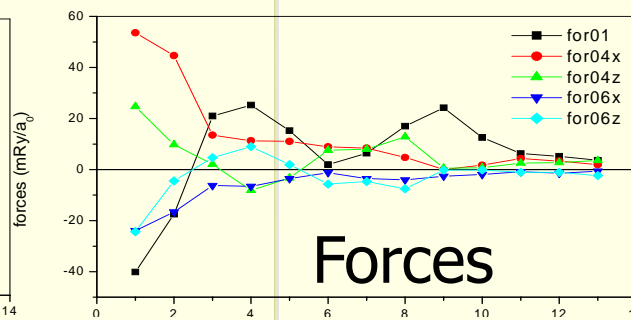
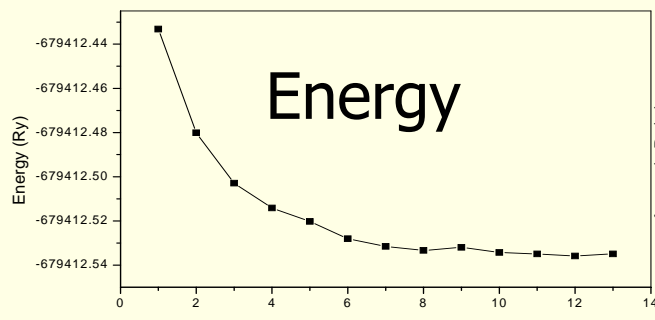


- `/home/pblaha/tio2> min_lapw -h`
  - *OPTIONS:*
    - `-p ->` does a k-point parallel calculation
    - `-it ->` use iterative diagonalization
    - `-sp ->` does a spin-polarized calculation (runsp\_lapw)
    - `-NI ->` without initialization of input-files (continue after a "crash")
    - `-i NUMBER ->` max. NUMBER (50) of structure changes
    - `-j JOB ->` job-file JOB (run\_lapw -I -fc 1. -i 40)
  - *CONTROL FILES:*
    - `.minstop` stop after next structure change
- `/home/pblaha/tiO2> cat tio2.inM`
  - `PORT 2.0`  `#(NEW1, NOSE, MOLD, tolf (a4,f5.2))`
  - `0.0 1.0 1.0 1.0`  `# Atom1 (0 will constrain a coordinate)`
  - `1.0 1.0 1.0 1.0`  `# Atom2 (NEW1: 1,2,3:delta_i, 4:eta (1=MOLD, damping))`
- **monitor minimization in file `case.scf_mini`**
  - *contains last iteration of each geometry step*
  - *each step N is saved as `case_N.scf` (overwritten with next `min_lapw` !)*



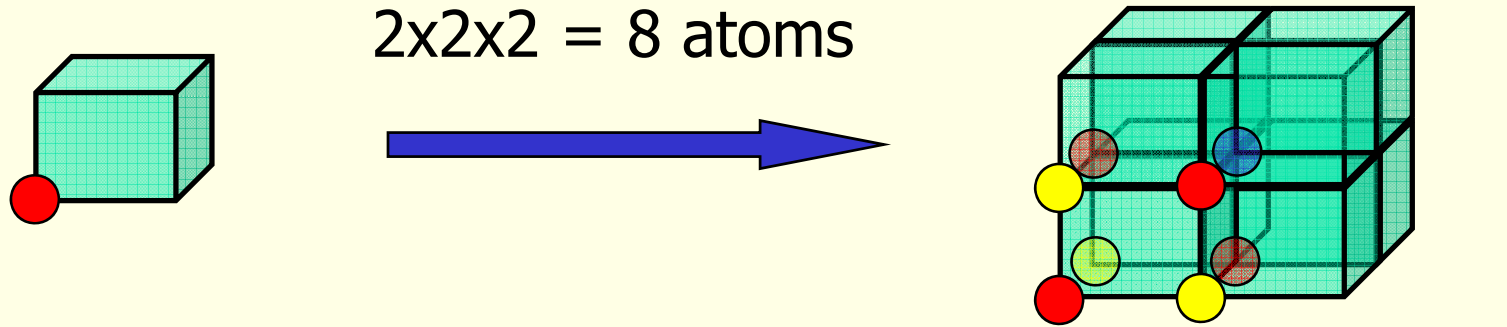
- damped Newton mechanics scheme (NEW1: with variable step)
- **quite efficient quasi-Newton (PORT) scheme**
  - minimizes E (using forces as gradients)
  - If minimization gets stuck or oscillates: (because E and  $F_i$  are inconsistent):
    - touch .minstop; min -nohess (or rm case.tmpM .min\_hess)
    - improve scf-convergence (-ec), Rkmax, k-mesh, ...
    - change to NEW1 scheme

## W impurity in Bi (2x2x2 supercell: Bi<sub>15</sub>W)





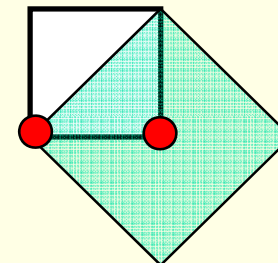
# Supercells



$(0,0,0)$	$P \rightarrow 8 \text{ atoms}$	$(0,0,0)$	$(.5,0,0)$	$(.5,.5,0)$	$(.5,.5,.5)$
			$(0,.5,0)$	$(.5,0,.5)$	
			$(0,0,.5)$	$(0,.5,.5)$	
	$B \rightarrow 4 \text{ atoms}$	yes	yes	no	no
	$F \rightarrow 2 \text{ atoms}$	yes	no	no	yes

4x4x4 supercells: P (64), B (32), F (16) atoms

$\sqrt{2} \times \sqrt{2}$  supercells (1  $\rightarrow$  2 atoms)







# Supercells



- Program „supercell“:
  - *start with „small“ struct file*
  - *specify number of repetitions in  $x, y, z$  (only integers, e.g. 2x2x1)*
  - *specify  $P$ ,  $B$  or  $F$  lattice*
  - *add „vacuum“ for surface slabs (only (001) indexed surfaces)*
  - *shift all atoms in cell*
- You must break symmetry!!!
  - *replace (impurities, vacancies) or displace (phonons) at least 1 atom*
- At present „supercell“ works only along unit-cell axes!!!



## Structeditor (by R.Laskowski)



- requires octave (matlab) and opendx (visualization)
- allows complex operations on struct-files

```
octave
```

```
s=loadstruct("GaN.struct")
```

```
# make an orthorhombic supercell and visualize it
```

```
a=[1 0 0; 1 1 0; 0 0 2]
```

```
sout=makesupercell (s,a);
```

```
showstruct(sout);
```

```
# save it as test.struct
```

```
savestruct (sout,"test.struct");
```

```
# get help on all commands
```

```
helpstruct
```

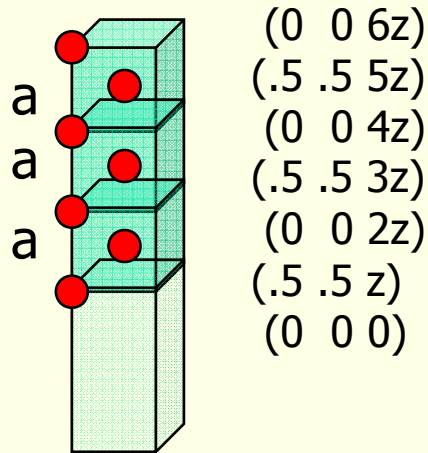


# Surfaces



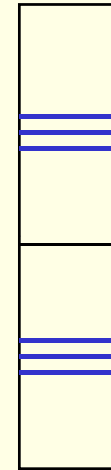
- 2D-slabs with finite number of layers with „vacuum“ in 3<sup>rd</sup> dimension

bcc (001) 7 layers:



(0 0 6z)  
 (.5 .5 5z)  
 (0 0 4z)  
 (.5 .5 3z)  
 (0 0 2z)  
 (.5 .5 z)  
 (0 0 0)

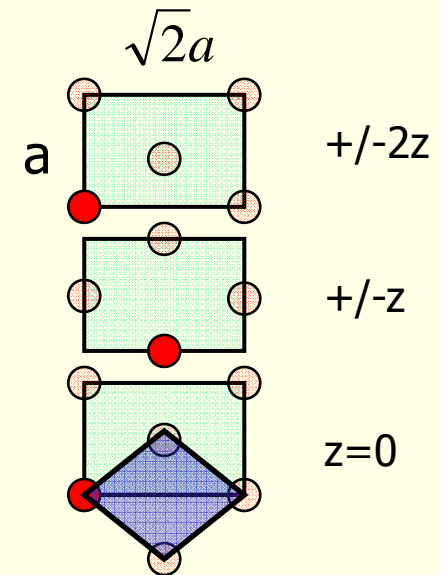
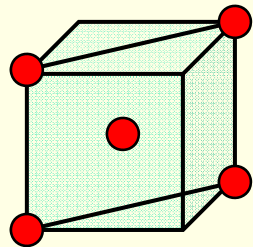
with lattice parameters:  
 $a, a, c = (3a + 15 - 20 \text{ bohr vacuum})$   
 shift to (0 0 0)  $z = a/2c$   
 inversion

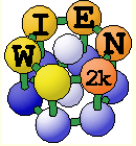


bcc (110):

orthorhombic CXY-lattice:  $a, \sqrt{2}a, c$

(0 0 0)  $z = a/\sqrt{2}a c$   
 (0 .5 +/-z)  
 (0 0 +/-2z)





# Calculations of Phonons: The Direct Method



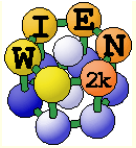
**WIEN2k + Phonon**

*Copyright by K.Parlinski*



<http://wolf.ifj.edu.pl/phonon/>

(alternatively use D.Alfe`s PHON code +W2P-interface from G.Madsen)



## THEORY OF DIRECT METHOD



System energy  $E$  (at  $T = 0$ ) as a function of atomic positions  $\mathbf{R}(\mathbf{n}, \mu)$  is

$$E(\mathbf{R}(\mathbf{n}, \mu), \dots, \mathbf{R}(\mathbf{m}, \nu), \dots) = E_0 + \frac{1}{2} \sum_{\mathbf{n}, \mu, \mathbf{m}, \nu} \Phi(\mathbf{n}, \mu, \mathbf{m}, \nu) \mathbf{U}(\mathbf{n}, \mu) \mathbf{U}(\mathbf{m}, \nu)$$

where the *force constant matrix* are

$$\Phi_{i,j}(\mathbf{n}, \mu, \mathbf{m}, \nu) = \left. \frac{\partial^2 E}{\partial R_i(\mathbf{n}, \mu) \partial R_j(\mathbf{m}, \nu)} \right|_0$$

is defined at  $\left. \frac{\partial E}{\partial R_i(\mathbf{n}, \mu)} \right|_0 = 0$ .

The *dynamical matrix* is defined as

$$\mathbf{D}(\mathbf{k}; \mu, \nu) = \frac{1}{\sqrt{M_\mu M_\nu}} \sum_{\mathbf{m}} \Phi(0, \mu; \mathbf{m}, \nu) \exp\{-2\pi i \mathbf{k} \cdot [\mathbf{R}(0, \mu) - \mathbf{R}(\mathbf{m}, \nu)]\}$$

$\mathbf{m}$  runs over *all* atoms. Diagonalization of the dynamical matrix

$$\omega^2(\mathbf{k}, j) \mathbf{e}(\mathbf{k}, j) = \mathbf{D}(\mathbf{k}) \mathbf{e}(\mathbf{k}, j)$$

gives phonon frequencies  $\omega^2(\mathbf{k}, j)$  and polarization vectors  $\mathbf{e}(\mathbf{k}, j)$ .

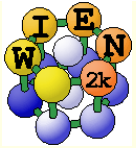
Any *atomic displacement*  $\mathbf{U}(\mathbf{m}, \nu)$  generates forces

$$\mathbf{F}(\mathbf{n}, \mu) = -\partial E / \partial \mathbf{R}(\mathbf{n}, \mu)$$

on all other atoms. Hence

$$F_i(\mathbf{n}, \mu) = -\sum_{\mathbf{m}, \nu, j} \Phi_{i,j}(\mathbf{n}, \mu, \mathbf{m}, \nu) U_j(\mathbf{m}, \nu)$$

Master equation of direct method.



## CUMMULANT FORCE CONSTANTS

Displace an atom by  $\mathbf{U}(\mathbf{m}, \nu)$

$$F_i(\mathbf{n}, \mu) = - \sum_{\mathbf{L}} \Phi_{i,j}(\mathbf{n}, \mu, \mathbf{m} + \mathbf{L}, \nu) U_j(\mathbf{m}, \nu)$$

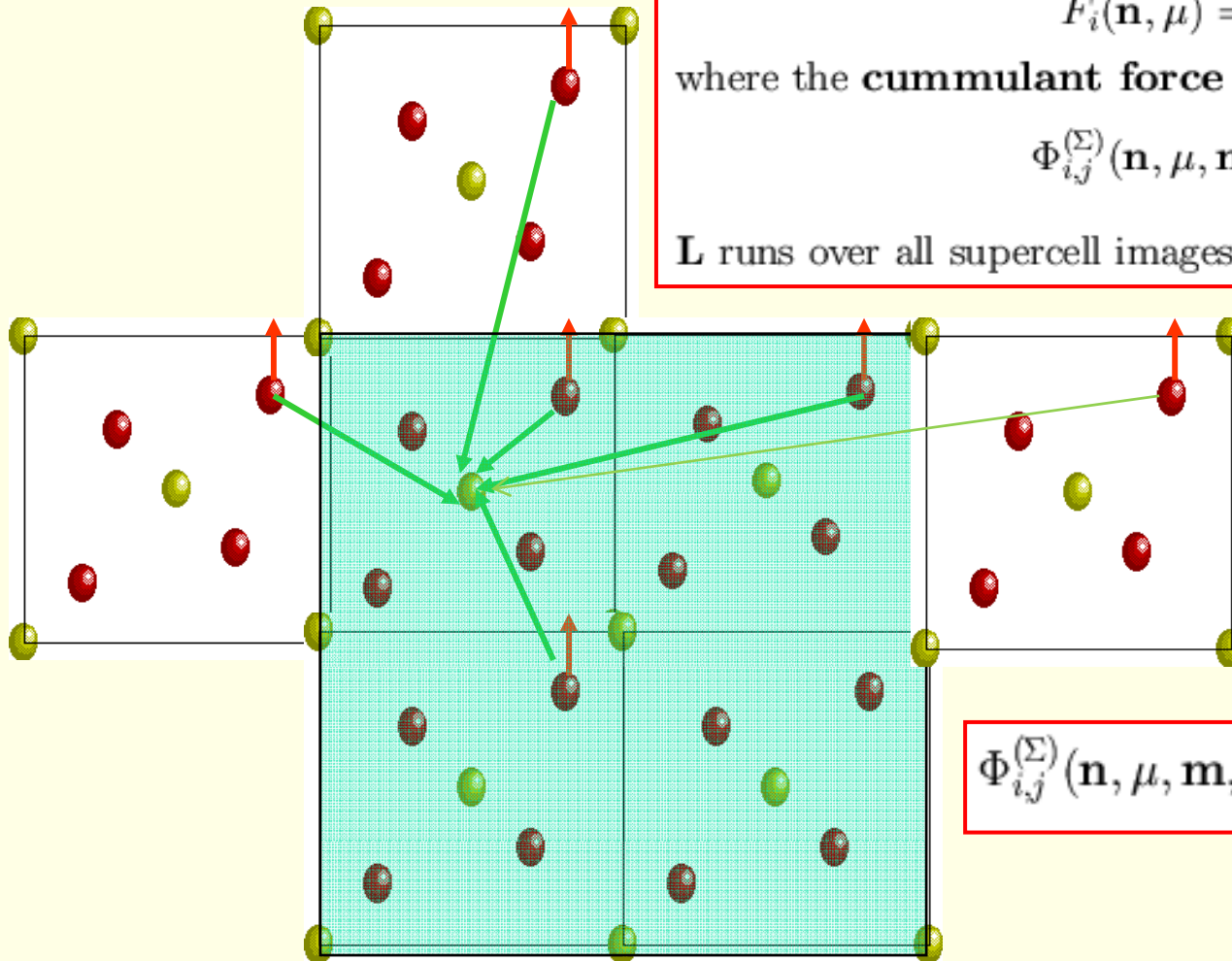
$\mathbf{L} = (L_a, L_b, L_c)$  are the indices of supercell lattice constants.  
or

$$F_i(\mathbf{n}, \mu) = -\Phi_{i,j}^{(\Sigma)}(\mathbf{n}, \mu, \mathbf{m}, \nu) U_j(\mathbf{m}, \nu)$$

where the **cummulant force constant** is

$$\Phi_{i,j}^{(\Sigma)}(\mathbf{n}, \mu, \mathbf{m}, \nu) = \sum_{\mathbf{L}} \Phi_{i,j}(\mathbf{n}, \mu, \mathbf{m} + \mathbf{L}, \nu)$$

$\mathbf{L}$  runs over all supercell images.



$$\Phi_{i,j}^{(\Sigma)}(\mathbf{n}, \mu, \mathbf{m}, \nu) = \sum_{\mathbf{L}} \Phi_{i,j}(\mathbf{n}, \mu, \mathbf{m} + \mathbf{L}, \nu)$$



# Supercell dynamical matrix. Exact wave vectors.



Conventional dynamical matrix:

$$D(\mathbf{k}; \mu, \nu) = \frac{1}{\sqrt{M_\mu M_\nu}} \sum_{\mathbf{m}} \Phi(0, \mu; \mathbf{m}, \nu) \exp\{-2\pi i \mathbf{k} \cdot [\mathbf{R}(0, \mu) - \mathbf{R}(\mathbf{m}, \nu)]\}$$

Supercell dynamical matrix:

$$D^{(SC)}(\mathbf{k}; \mu, \nu) = \frac{1}{\sqrt{M_\mu M_\nu}} \sum_{\mathbf{m} \in SC} \Phi^{(SC)}(0, \mu; \mathbf{m}, \nu) \exp\{-2\pi i \mathbf{k} \cdot [\mathbf{R}(0, \mu) - \mathbf{R}(\mathbf{m}, \nu)]\}$$

These two matrices are equal if

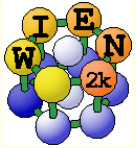
$$D^{(SC)}(\mathbf{k}; \mu, \nu) = D(\mathbf{k}; \mu, \nu)$$

- **interaction range** is confined to **interior** of supercell (supercell is big enough)
- wave vector is **commensurate with the supercell** and fulfils the condition (independent of interaction range):

$$\exp\{-2\pi i \mathbf{k}_s \cdot \mathbf{L}\} = 1$$

At wave vectors  $\mathbf{k}_s$  the phonon frequencies are “exact”, provided the **supercell contains the complete list of neighbors**.

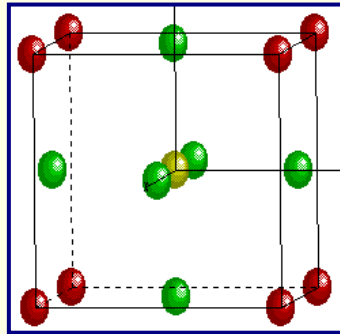
Wave vectors  $\mathbf{k}_s$  are commensurate with the supercell size.



# Exact wave vectors

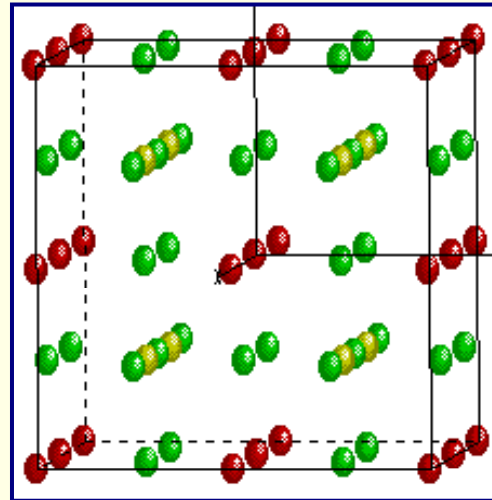


1x1x1



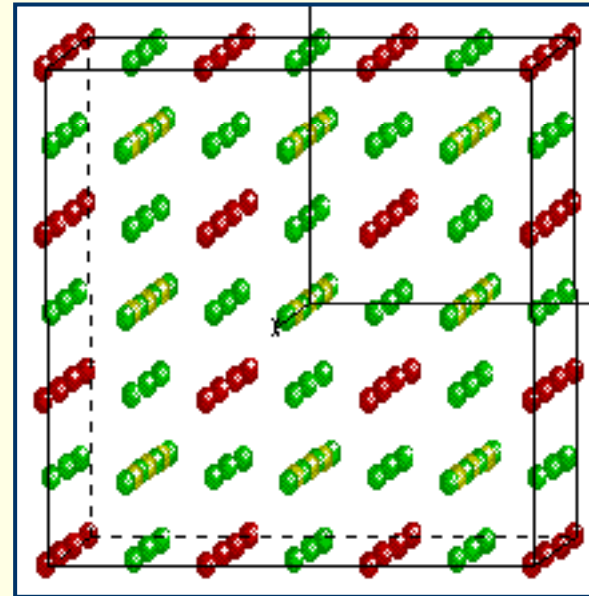
Exact:  $\Gamma$

2x2x2

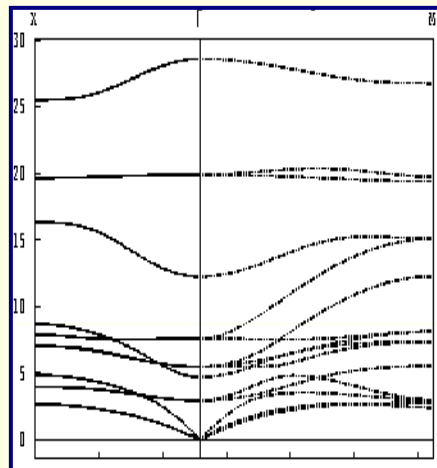


Exact:  $\Gamma, X, M, R$

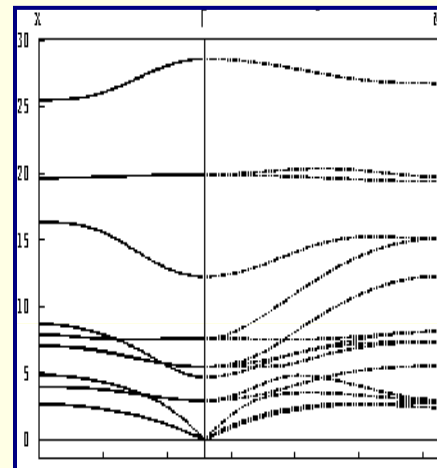
3x3x3



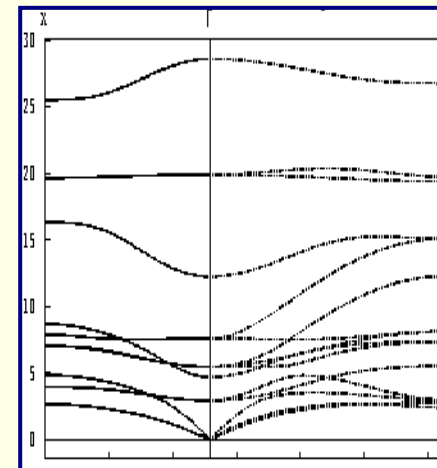
Exact:  $\Gamma$



X  $\Gamma$  M



$\Gamma$



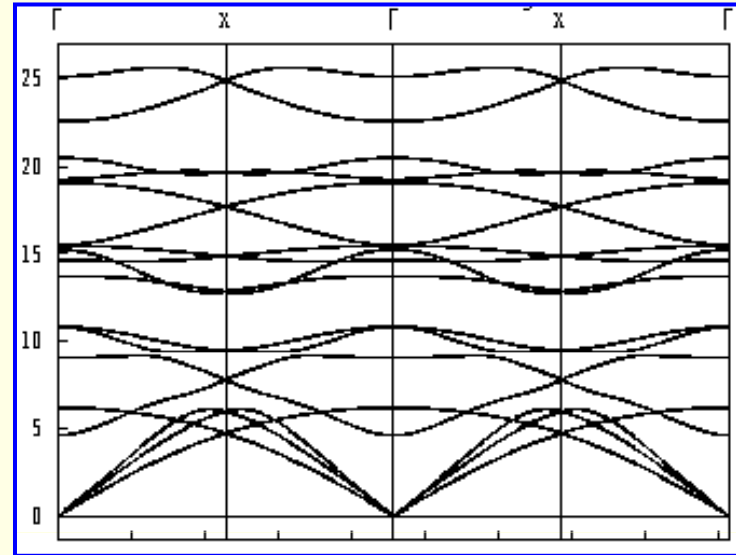




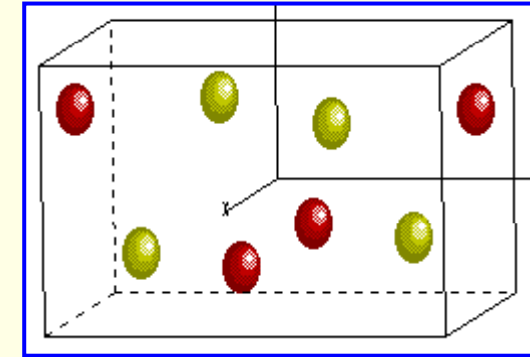
# Phonon dispersions + density of states



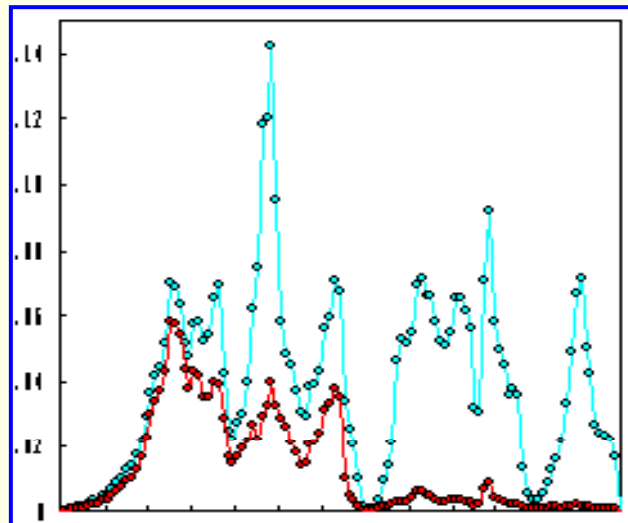
Frequency  
 $\omega$



GeO<sub>2</sub> P4<sub>2</sub>/mm

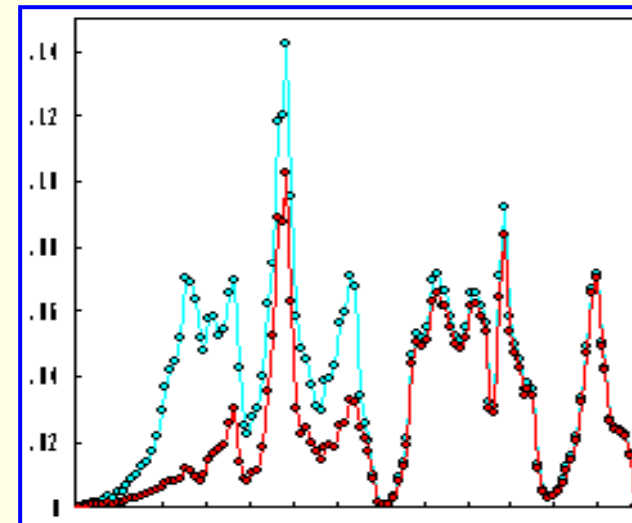


Total + Germanium



$\omega$

Total + Oxygen



$\omega$



# Thermodynamic functions of phonon vibrations



Internal energy:

$$E = \frac{1}{2} r \int_0^\infty d\omega g(\omega) (\hbar\omega) \coth h \left( \frac{\hbar\omega}{2k_B T} \right)$$

Free energy:

$$F = r k_B T \int_0^\infty d\omega g(\omega) \ln \left[ 2 \sinh \left( \frac{\hbar\omega}{2k_B T} \right) \right]$$

Entropy:

$$S = r k_B \int_0^\infty d\omega g(\omega) \left\{ \left( \frac{\hbar\omega}{2k_B T} \right) \left[ \coth \left( \frac{\hbar\omega}{2k_B T} \right) - 1 \right] - \ln \left[ 1 - \exp \left( -\frac{\hbar\omega}{k_B T} \right) \right] \right\}$$

Heat capacity  $C_V$ :

$$C = r k_B \int_0^\infty d\omega g(\omega) \left( \frac{\hbar\omega}{k_B T} \right)^2 \frac{\exp \left( \frac{\hbar\omega}{k_B T} \right)}{\left[ \exp \left( \frac{\hbar\omega}{k_B T} \right) - 1 \right]^2}$$

Thermal displacements:

$$B_{ij}(\mu) = \langle U_i(\mu) U_j(\mu) \rangle$$

$$B_{il}(\mu) = \frac{\hbar r}{2M_\mu} \int_0^\infty d\omega g_{il,\mu}(\omega) \frac{1}{\omega} \coth h \left( \frac{\hbar\omega}{2k_B T} \right)$$



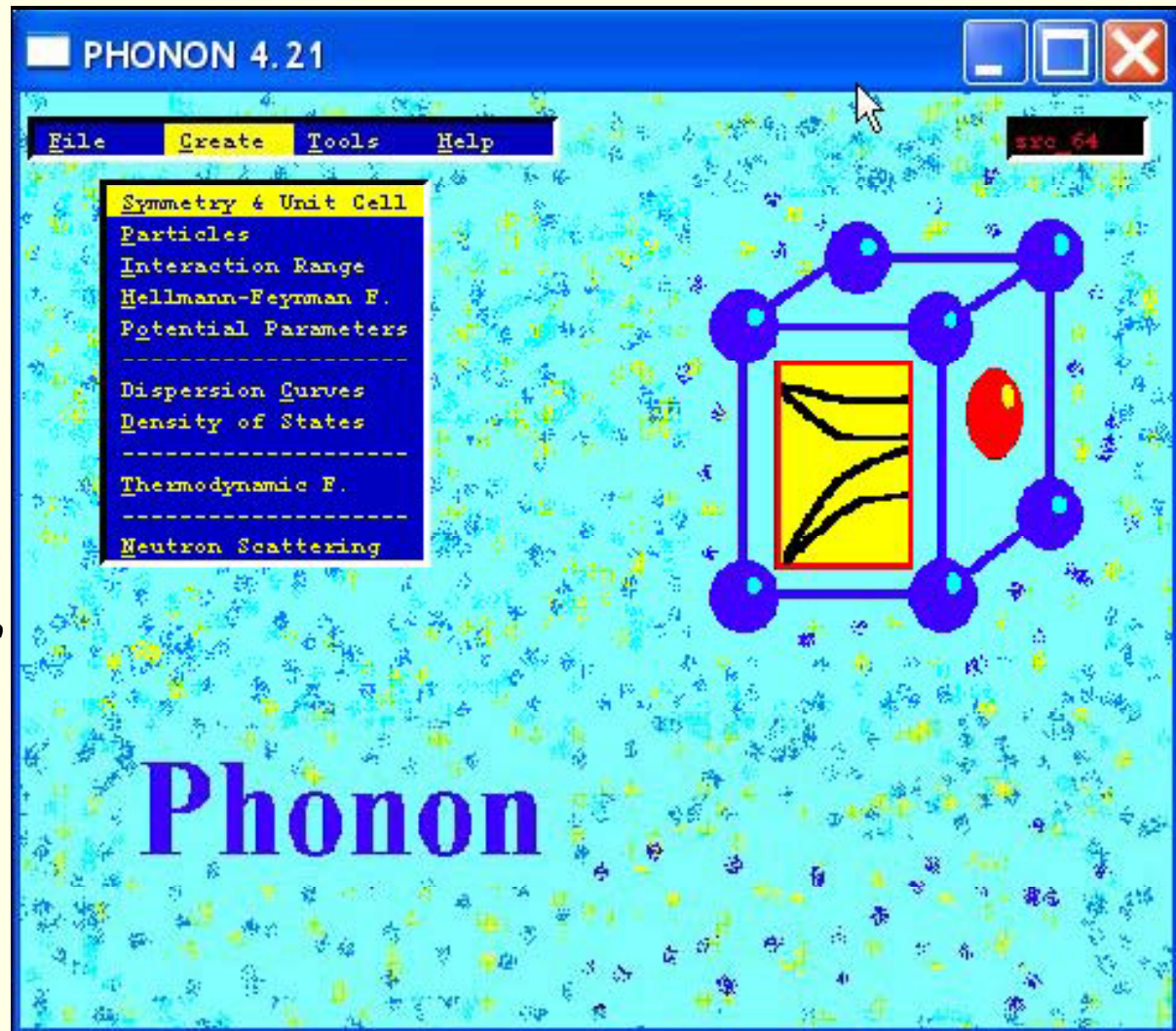
# PHONON-I



## ■ PHONON

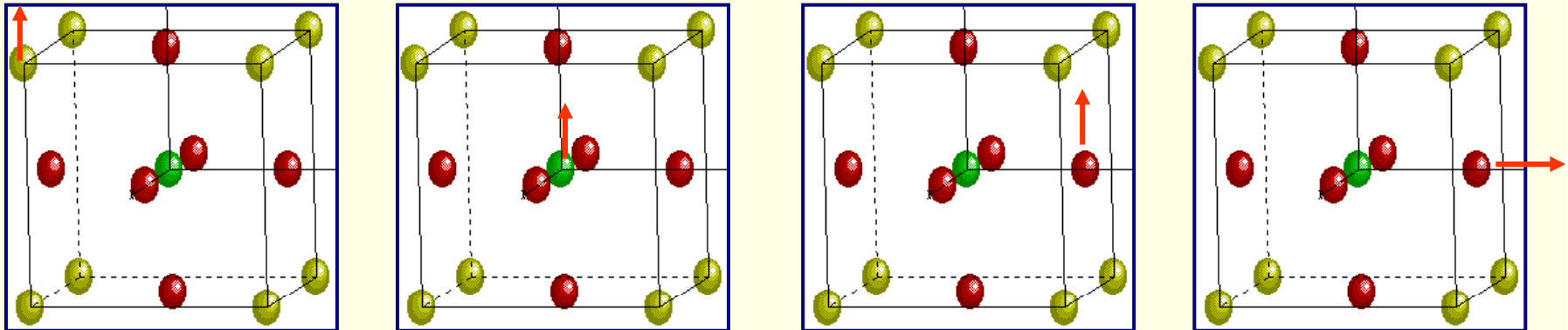
- *by K.Parlinski (Crakow)*
- *runs under MS-windows*
- *uses a „direct“ method to calculate Force-constants with the help of an ab initio program*
- *with these Force-constants phonons at arbitrary k-points can be obtained*

- Define your spacegroup
- Define all atoms



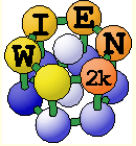
<http://wolf.ifj.edu.pl/phonon/>

- *selects symmetry adapted atomic displacements (4 displacements in cubic perovskites)*



(Displacement pattern for cubic perovskite)

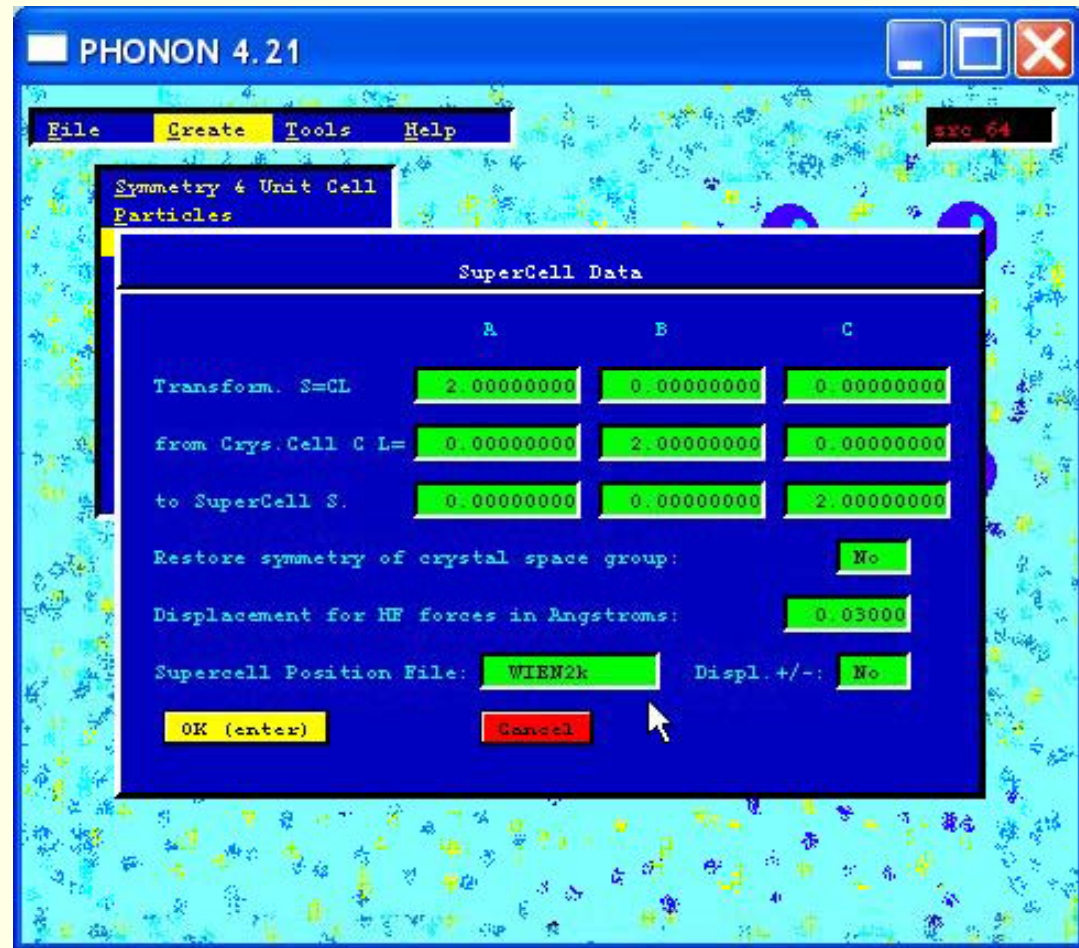
- *select a supercell: (eg. 2x2x2 atom P-type cell)*
- *calculate all forces for these displacements with high accuracy(WIEN2k)*
- *→ force constants between all atoms in the supercell*
- *→ dynamical matrix for arbitrary q-vectors*
- *→ phonon-dispersion ("bandstructure") using PHONON (K.Parlinski)*



# PHONON-II



- Define an interaction range (supercell)
  - *create displacement file*
  - *transfer case.d45 to Unix*
- Calculate forces for all required displacements
  - *init\_phonon\_lapw*
    - for each displacement a *case\_XX.struct* file is generated in an extra directory
    - runs *nn* and lets you define *RMT* values like:
      - 1.85 1-16



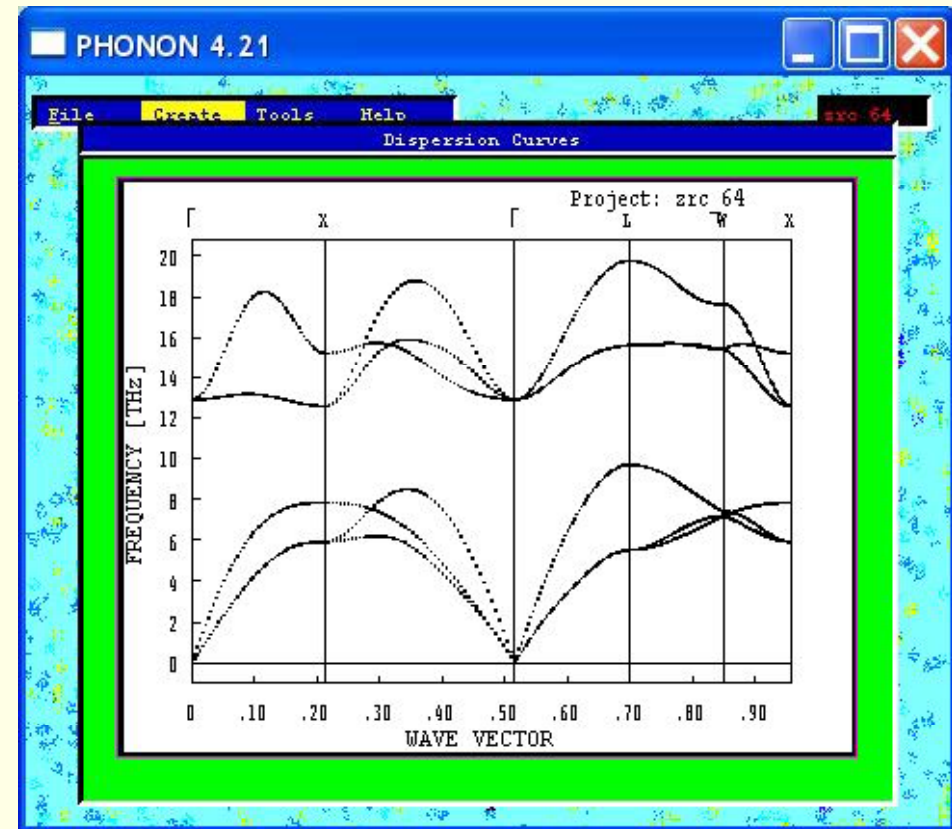
- *init\_lapw*: either *without symmetry* (and then copies this setup to all *case\_XX*) or *with symmetry* (must run *init\_lapw* for all *case\_XX*) (Do **NOT** use *SGROUP*)
- *run\_phonon*: *run\_lapw -fc 0.1 -i 40* for each *case\_XX*



# PHONON-III



- **analyze\_phonon\_lapw**
  - reads the *forces* of the *scf* runs
  - generates „*Hellman-Feynman*“ file *case.dat* and a „*symmetrized HF-file case.dsy* (when you have displacements in both directions)
    - check quality of forces:
    - $\sum F_x$  should be small (0)
    - $\text{abs}(F_x)$  should be similar for +/- displacements
- transfer *case.dat* (*dsy*) to Windows
- Import HF files to PHONON
- Calculate force constants
- Calculate phonons, analyze phonons eigenmodes, thermodynamic functions



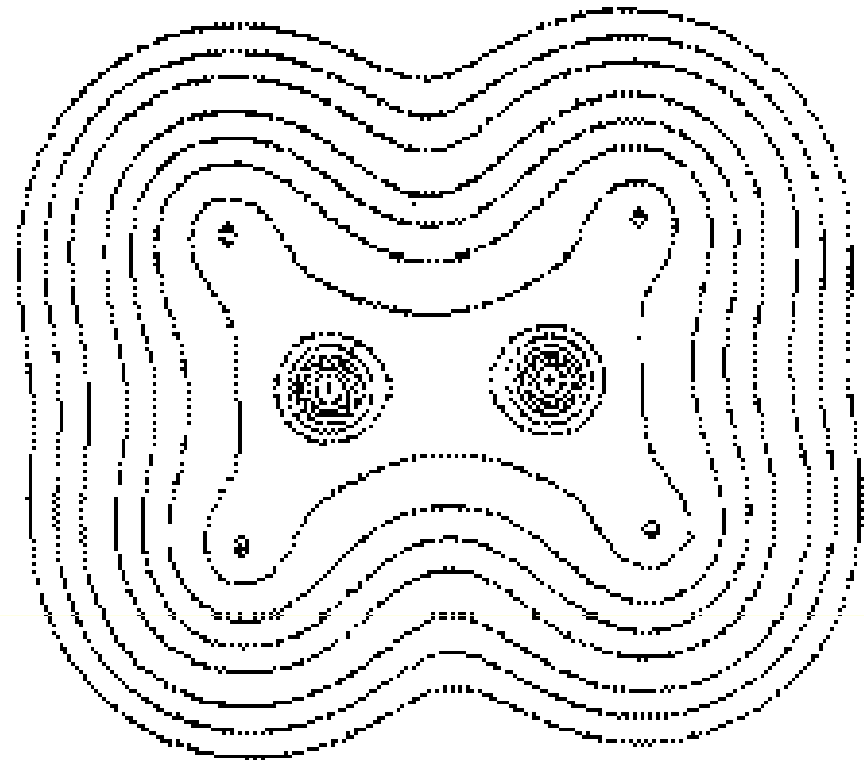
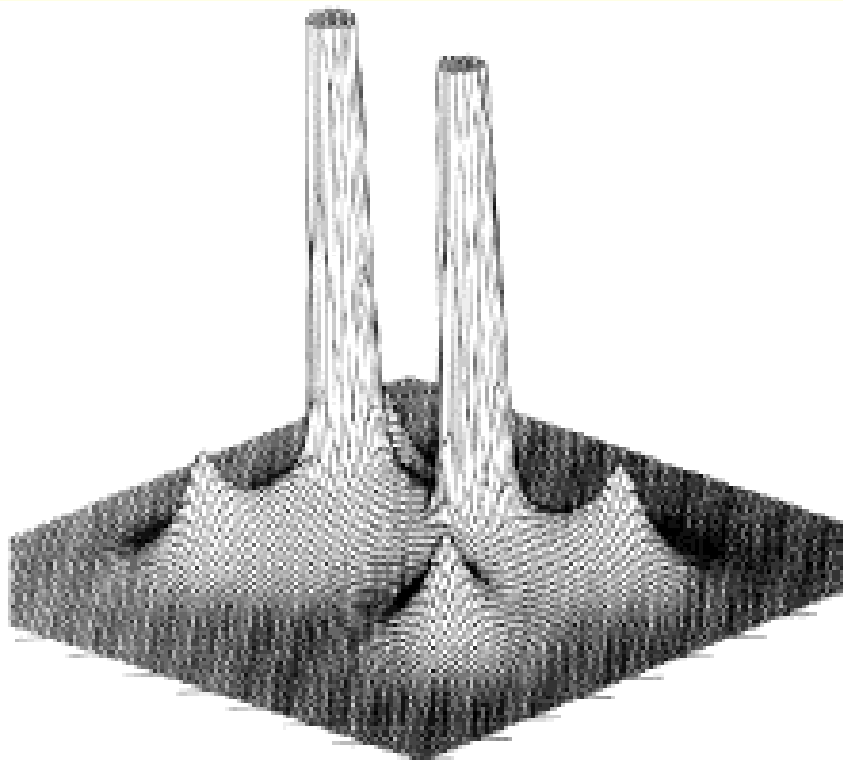


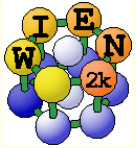
# Atoms in Molecules



- Theory to characterize atoms and chemical bonds from the topology of the electron density, by R.F.Bader  
([http://www.chemistry.mcmaster.ca/faculty/bader/aim/aim\\_0.html](http://www.chemistry.mcmaster.ca/faculty/bader/aim/aim_0.html))

## Electron density of $C_2H_4$

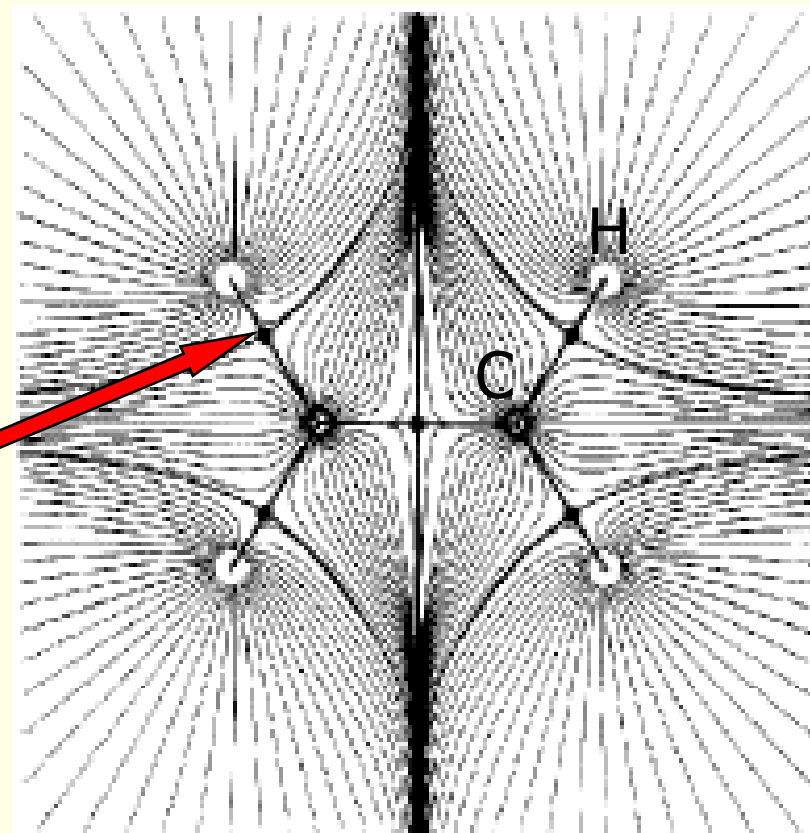




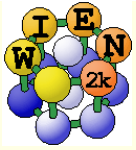
- Bonds are characterized by „critical points“, where  $\nabla\rho = 0$ 
  - density maximum: (3,-3); 3 negative curvatures  $\lambda$ , (at nucleus or non-NM)
  - bond CP: (3,-1): 2 negative, 1 positive  $\lambda$  (saddle point)
    - positive (and large) Laplacian: ionic bond
    - negative Laplacian: covalent bond
  - bridge CP: (3,1)
  - cage CP: (3,3) (minimum)

(3,-1) BCP

trajectories of constant  $\nabla\rho$   
originating at CPs in  $C_2H_4$

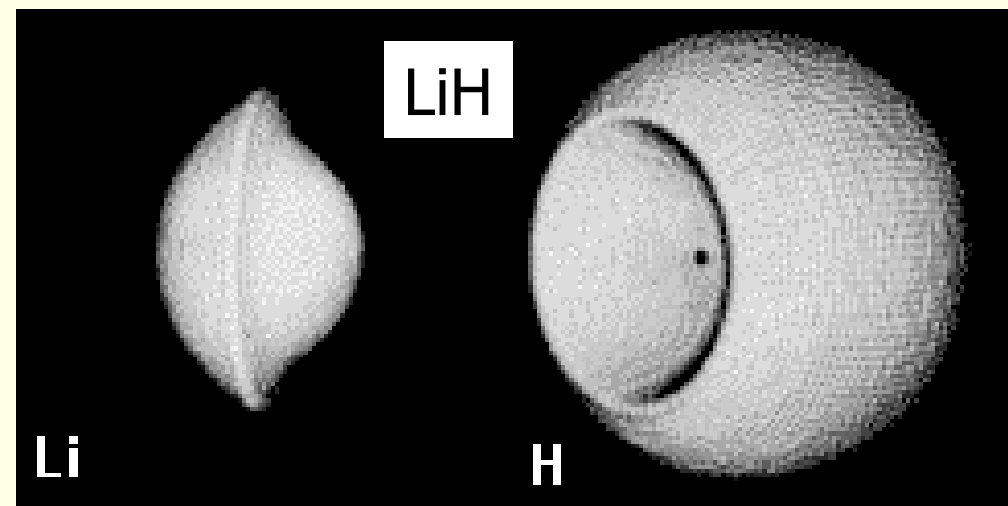
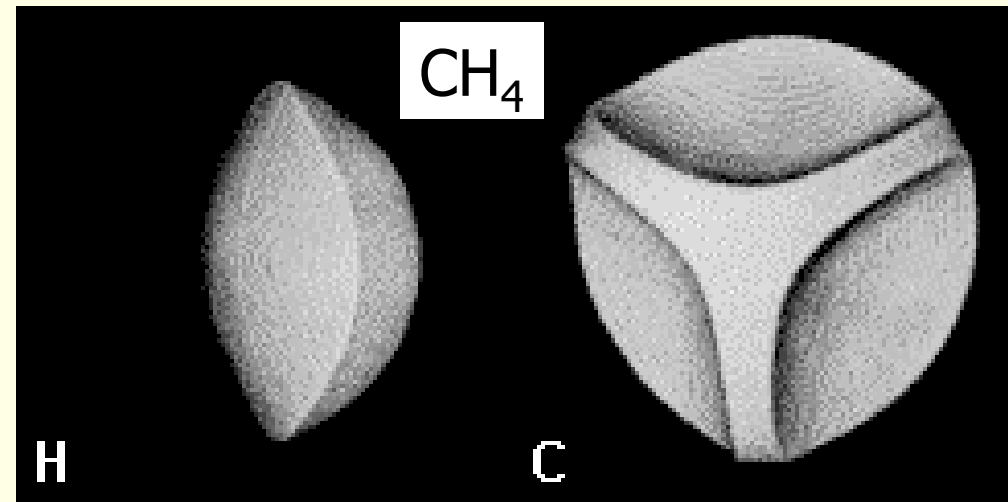
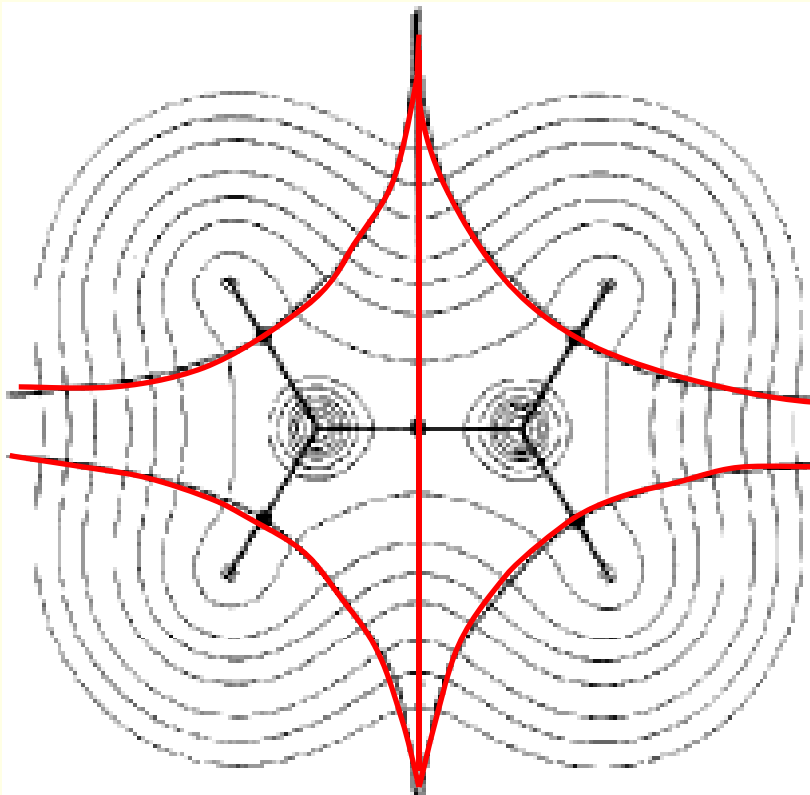






- "Atoms" are regions within a zero-flux surface  $\vec{\nabla}\rho \cdot \vec{n} = 0$

$\rho$  of  $C_2H_4$  with zero-flux lines defining atomic basins





## ■ Bader analysis of some inorganic compounds:

	$\rho(e/\text{\AA}^3)$	$\Delta\rho(e/\text{\AA}^5)$	Q (e)
Cl <sub>2</sub>	1.12	-6.1	-
I <sub>2</sub>	0.48	-0.9	-
TiC	0.51	1.8	1.7
TiN	0.47	3.9	1.7
TiO	0.43	5.8	1.5
KCl	0.08	1.2	0.6

Cl<sub>2</sub> more covalent  
then I<sub>2</sub>

more ionic, but less charge?

less ionic than TiC ?



## x aim [-c]



- You must have a "good" scf-density (case.clmsum)
  - *no core leakage, LMs up to  $L=8-10$  in case.in2*

### SURF

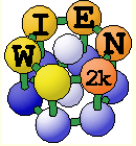
```
1          atom in center of surface (including MULT)
20 0.0 1.570796327  theta, 20 points, from zero to pi/2
20 0.0 0.785398163  phi, from 0 to pi/4 (depends on symmetry!!)
0.07 1.0 4        step along gradient line, rmin (has reached an atom)
1.65 0.1         initial R for search, step (a.u)
3 3 3           nshell
IRHO          "INTEGRATE" rho
WEIT         WEIT (surface weights are available in case.surf)
30          30 radial points outside min(RMIN,RMT)
END
```

-----  

### CRIT

```
1          atom around you search for critical points
ALL        two, three, four, all (dimers,trimers,....all=2+3)
3 3 3     nshell
END
```

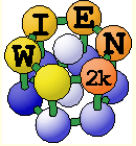
`extractaim_lapw:` → `critical_points_ang` (converted units)  
:PC  $x, y, z, \lambda_1, \lambda_2, \lambda_3, ch, laplacian, rho$



# WIEN2k- hardware/software



- WIEN2k runs on any **Unix/Linux** platform from PCs, workstations, clusters to supercomputers
  - *Intel Core-2 dual-core processors with fast memory bus (2-4 Gb memory, 1Gbit-net, SATA disks), Quad-cores ?*
  - *10 atom cells on 128Mb PC / 100 atom cells require 2 Gb RAM*
  - *installation support for many platforms + compiler*
- **Fortran90** (dynamical allocation, modules)
  - *real/complex version (inversion)*
  - *many individual modules, linked together with C-shell or perl-scripts*
- *web-based GUI – w2web (perl)*
- **f90 compiler (ifort, gfortran), BLAS-library (mkl, gotolib), perl5, ghostscript (+jpg), gnuplot(+png), Tcl/Tk (Xcrysden), pdf-reader, www-browser, octave, opendx**



# Installation of WIEN2k



- Register via <http://www.wien2k.at>
- Create your \$WIENROOT directory (e.g. `./WIEN2k` )
- Download `wien2k_08.tar` and examples (executables)
- Uncompress and expand all files using:
  - `tar -xvf wien2k_08.tar`
  - `gunzip *.gz`
  - `chmod +x ./expand_lapw`
  - `./expand_lapw`
- This leads to the following directories:
  - `./SRC` (*scripts, ug.ps*)
  - `./SRC_aim` (*programs*)
  - ...
  - `SRC_templates` (*example inputs*)
  - ...
  - `SRC_usersguide_html` (*HTML-version of UG*)
  - `example_struct_files` (*examples*)
  - `TiC`



# siteconfig\_lapw



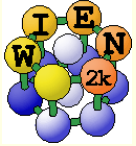
- \*\*\*\*\*
- \* W I E N \*
- \* site configuration \*
- \*\*\*\*\*
- S specify a system
- C specify compiler
- O specify compiler options, BLAS and LAPACK
- P configure Parallel execution
- D Dimension Parameters
- R Compile/Recompile
- U Update a package
- L Perl path (if not in /usr/bin/perl)
- Q Quit

D: define **NMATMAX** (adjust to your hardware/paging!):

**NMATMAX=5000** → 256Mb (real) or 500Mb (complex)

**NMATMAX=10000** → 1Gb (real) → 80-150 atoms/unitcell

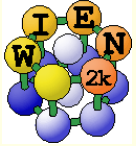
Always use „optimized“ BLAS library (ifort+mkl; gotolib, ATLAS-BLAS)



- **most common: Intel's Fortran compiler (not free for academic)**
  - ***ifort 9.x, 10.0***
    - 8.0 buggy ?, 10.1 buggy ?
    - needs unlimited "stacksize" (limit stacksize=unlimited, otherwise: "segmentation fault")
    - dynamic linking necessary (depends on ifort version, requires system and compiler libraries at runtime, may need \$LD\_LIBRARY\_PATH)
  - ***ifc (7.1), gfortran***
    - -static linking possible
    - for ifc: patch for 2 externals, see faq-web page or mailing-list digest
  - ***mkl 9.x; mkl 10.0 (buggy ?); gotolib, acml-lib, ATLAS-BLAS***
    - -L/opt/intel/mkl/lib -lmkl\_em64t → libmkl\_em64t.so
  - ***compiler/linker options depend on compiler version + Linux-version !!***
    - -FR (free format)      -lguide -lpthread -pthread







# k-point Parallelization



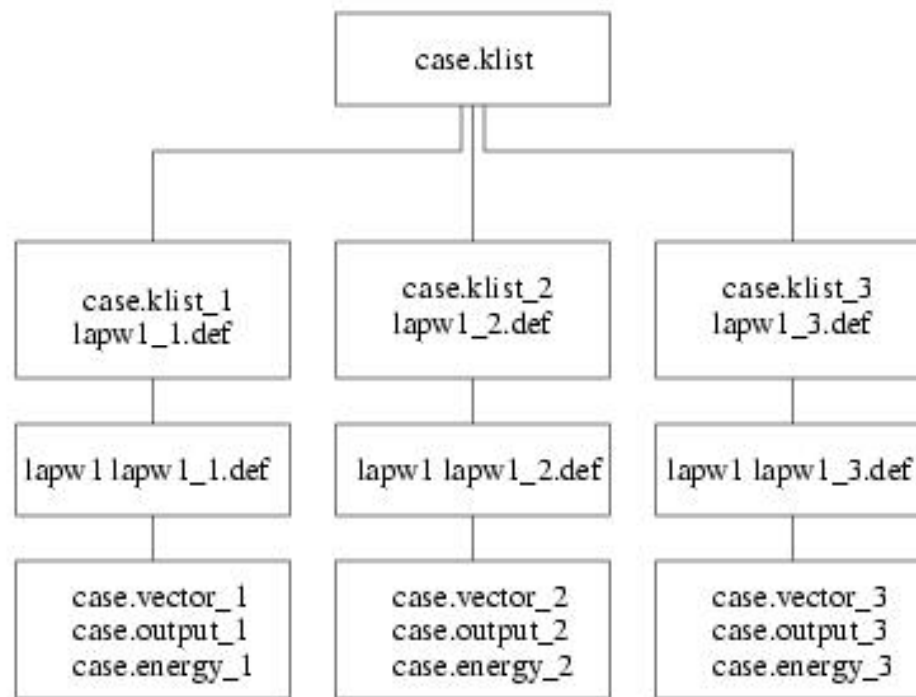
- **very efficient parallelization** even on loosely coupled PCs (**slow network**): lapw1+lapw2
  - *common NFS filesystem (files must be accessible with the same path on all machines; use **/host1** as data-directory on **host1**)*
  - *rsh/ssh without password (.rhosts; private/public keys)*
    - ssh-keygen -t rsa
    - append **.ssh/authorized\_keys** on remote host with **id\_rsa.pub** of local host
    - **.machines** file:
      - 1:host1 (speed:hostname)
      - 2:host2
      - granularity:1 (1:10k+10k; 3: 3+3+3+3+3+3+rest → load balancing, not with \$SCRATCH, -it)
      - extrafine:1 (rest in junks of 1 k)
    - **testpara** (tests distribution); run\_lapw -p
  - *case must fit into memory of one PC !*
  - *high NFS load: use local \$SCRATCH directory (only with commensurate k-points/hosts)*



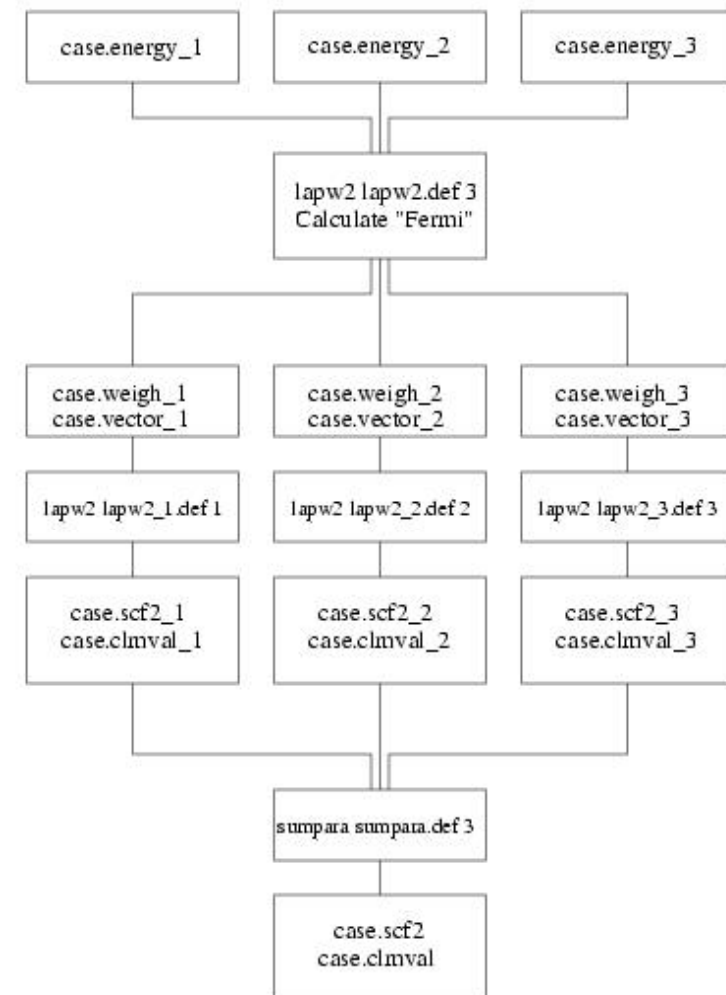
# Flow of parallel execution



## lapw1para



## lapw2para



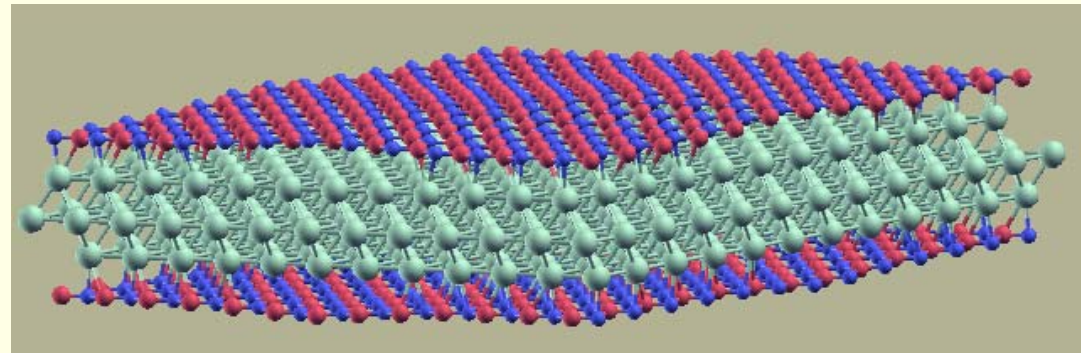


# fine-grain mpi-parallelization



- for bigger cases ( $> 30$  atoms)
- **faster** network (Gbit, Myrinet, Infiniband, shared memory machines)
- **mpi + scalapack**
  - .machines file:
    - 1:host1:4 host2:4                      8 mpi-parallel jobs on host1 and host2
    - lapw0:host1:4 host2:4                    8 parallel jobs; atom-loops only!!!
- **simultaneous k-point and mpi-parallelization possible**

- *BN/Rh(111) nanomesh:*



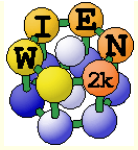
- unitcell with 1150 atoms (556 non-equiv.)
- NMAT=45000-60000; 64 cpus, 2h / iteration



# WIEN2k\_08.1



- **new version with largely improved performance**
  - *lapw0*: speed-up (non-spinpolarized)
  - *lapw1 + lapw2*: large speedup,
    - new iterative diagonalization (5x faster, run\_lapw -it)
  - **largely improved *mpi-versions***
    - perfect scaling with number of cpus for lapw0\_mpi (atoms), lapw1\_mpi (PW, memory, HAMILT+HNS "scales"; diagonalization not "perfect") and lapw2\_mpi (atoms + PW)
    - Gbit network "enough"
    - unitcell with 1150 atoms (556 non-equiv.): BN/Rh(111)
      - NMAT=45000-60000; 64 cpus, 2h / iteration
  - *mixer* (L.Marks): 30-50% less scf-cycles, very stable, "real" convergence (no "pseudo-convergence" due to small mixing)
  - *min\_lapw*: charge extrapolation for new positions (30-70% less iterations)

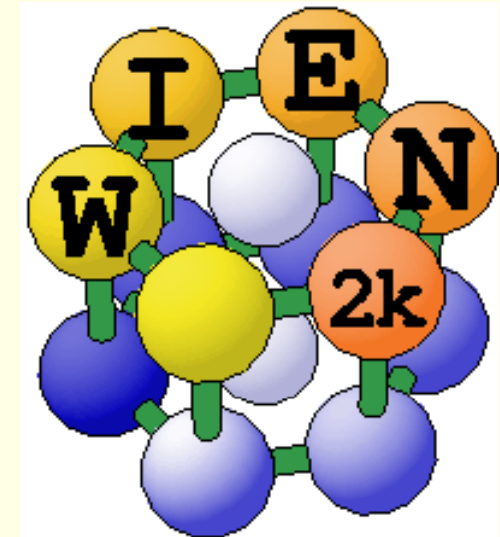


# Development of W2GRID

**DI Johannes Schweifer**

Project 8

9<sup>th</sup> Aurora plenary meeting (Strobl)





# A computational task within W2GRID



user



computational task (run\_lapw)



GridClient

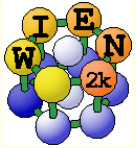


- 1) analysis of job (memory, time, nodes)
- 2) checking all hosts (load)
- 3) selection of suitable hosts
- 4) automatic distribution of tasks to „best“ hosts
- 5) checking of running jobs, control files
- 6) returns output data

platform-dependent execution



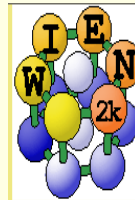
GridServer on local workstations, clusters or supercomputers



# Task for electron density plot



- A task consists of
  - a series of steps
  - that must be executed
  - to generate a plot
- For electron density plot
  - select states by E-window in case.in2 (e.g. valence  $e$ : Ti-3d,4s, C-2s,2p)
  - for difference densities make sure you calculate the same states for the free atoms
  - select plane for plot (do not put an atom at the corner or edges)
  - generate 3D or contour plot with gnuplot or Xcrysden (Tone.Kokalj@ijs.si)
  - reset EMIN in case.in2



## Execution >>

StructGen™  
initialize calc.  
run SCF  
single prog.  
optimize(V,c/a)  
mini. positions

## Utils. >>

## << Tasks

El. Dens.  
DOS  
XSPEC  
TELNES.2  
OPTIC  
Bandstructure

## Files >>

struct file(s)  
input files  
output files  
SCF files

## Session Mgmt. >>

change session  
change dir  
change info

## Configuration

Usersguide  
html-Version

Session: TiC

/area51/pblaha/lapw/2005-june/TiC

23:05:35 idle  
refresh | no refresh

## Electron density plots

You must have a valid TiC.vector file (from an scf calculation).  
If you don't have it, you must run "x lapw1" with an appropriate input.

change EMIN to truncate semicore

Calculate clmval

### For difference densities only !

#### default valence states:

#### non-default valence states:

Calculate atomic valence densities

Calculate atomic valence densities as defined above

put P for all your states

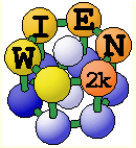
Edit input-file

Calculate density

Plot Density

reset EMIN

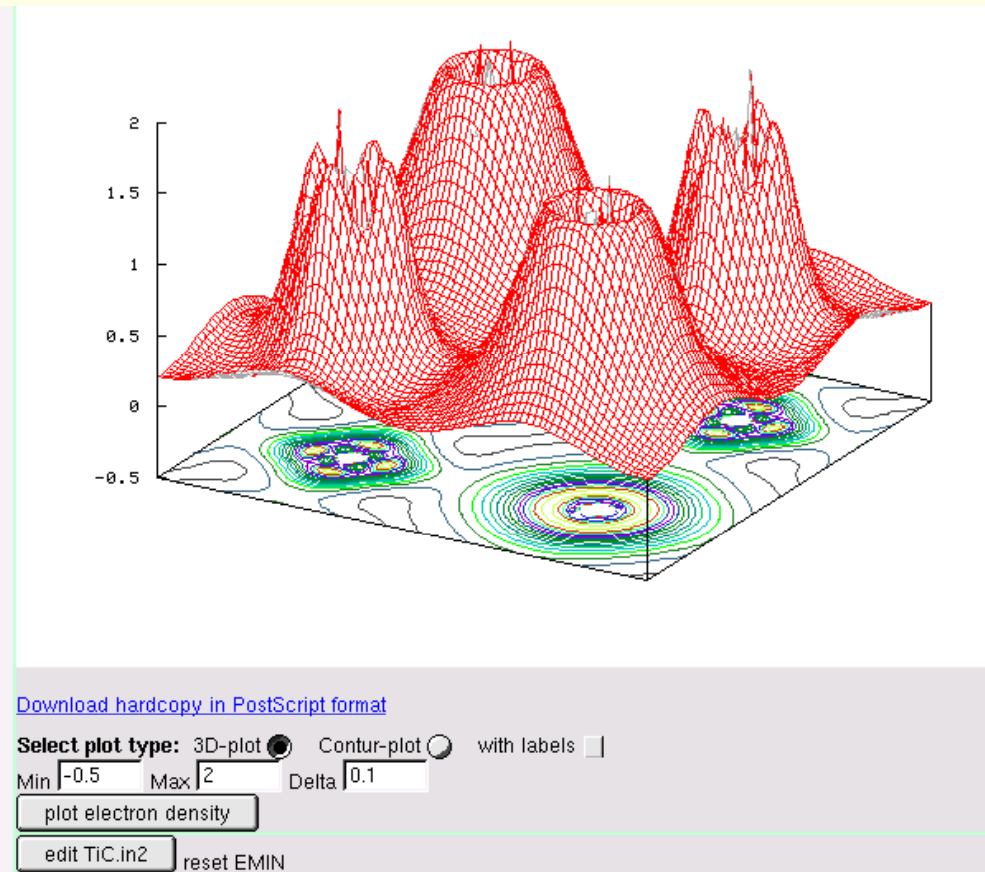
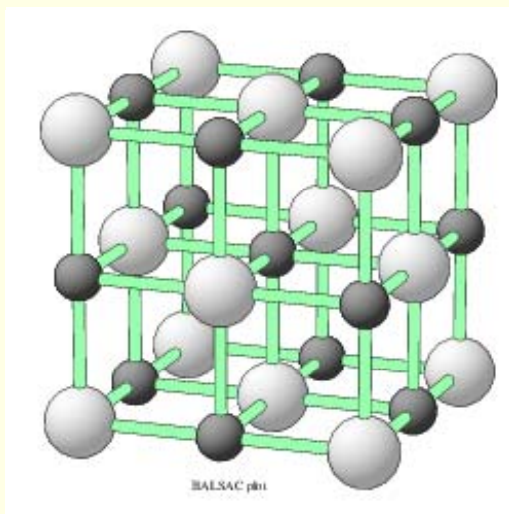


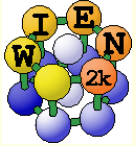


# TiC electron density



- NaCl structure (100) plane
- Valence electrons only
- plot in 2 dimensions
- Shows
  - *charge distribution*
  - *covalent bonding*
    - between the Ti-3d and C-2p electrons
  - $e_g/t_{2g}$  symmetry





# band structure



Session: TiC

/susi/pblaha/lapw/TiC

## Band structure

(save klist as xcrysden.klist)

fcc

Calculate Eigenvalues  interactively

**needed only for continuous lines in the plot (not for non-symmorphic spacegroups)!**

Calculate irreducible representations  interactively

**for band character plots only!**

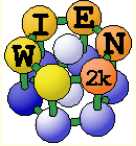
Calculate partial charges ("qtl"-file)  interactively

Insert correct EF

Calculate bandstructure

Plot bandstructure

- Note: spaghetti may fail if you have an old (incompatible) case.irrep file



# case.insp



Session: TiC

/susi/pblaha/lapw/TiC

## File:

/susi/pblaha/lapw/TiC/TiC.insp

continue with bandstructure

Save

Download this file:

Header from TiC.qtl and possible FERMI energies:

```
ATOM 1 tot,0,1,2,3,xdos(i,j),j=1,i),i=1,lxdos2)
ATOM 2 tot,0,1,2,D-eg,D-t2g,3
```

```
/susi/pblaha/lapw/TiC/TiC.scf: EF (TETRAH.M.)= 0.74210
/susi/pblaha/lapw/TiC/TiC_save.scf: EF (TETRAH.M.)= 0.74250
/susi/pblaha/lapw/TiC/TiC_vol_-10.0.scf: EF (TETRAH.M.)= 0.84297
/susi/pblaha/lapw/TiC/TiC_vol__-5.0.scf: EF (TETRAH.M.)= 0.79064
/susi/pblaha/lapw/TiC/TiC_vol__10.0.scf: EF (TETRAH.M.)= 0.65771
/susi/pblaha/lapw/TiC/TiC_vol___0.0.scf: EF (TETRAH.M.)= 0.74249
/susi/pblaha/lapw/TiC/TiC_vol___5.0.scf: EF (TETRAH.M.)= 0.69841
```

```
### Figure configuration
5.0 3.0 # paper offset of plot
10.0 15.0 # xsize,ysize [cm]
1.0 4 # major ticks, minor ticks
1.0 1 # character height, font switch
1.1 2 4 # line width, line switch, color swit

### Data configuration
-1.0 1.0 1 # energy range, energy switch (1:Ry, 2
1 0.74250 # Fermi switch, Fermi-level (in Ry u
1 999 # number of bands for heavier plotting
2 2 0.8 # jatom, jtype, size of heavier plott
```